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## The stability and unexpected chemistry of oxide clusters†

Xiaohu Yu,<sup>id</sup>\*<sup>ab</sup> Artem R. Oganov,<sup>acd</sup> Qiang Zhu,<sup>e</sup> Fei Qi<sup>id</sup><sup>af</sup> and Guangrui Qian<sup>e</sup>

Using evolutionary structure prediction and *ab initio* thermodynamics, we determine stable compositions and structures of small  $Ce_mO_n$  and  $Fe_mO_n$  clusters at realistic temperatures and oxygen pressures. We use second energy differences as the criterion determining clusters of particular stability ("magic" clusters), whereas HOMO–LUMO gaps are used to gauge chemical inertness – *i.e.* the ability of a cluster to survive in a complex chemical environment. We find that, similar to atomic nuclei (which are clusters made of neutrons and protons), compositional space of two-component clusters also has ridges and islands of stability, surrounded by sea of instability. Long ridges of stability correspond to stoichiometric compositions – *e.g.*,  $(CeO_2)_k$ ,  $(Ce_2O_3)_k$ ,  $(FeO)_k$ ,  $(Fe_2O_3)_k$  and  $(Fe_3O_4)_k$  series of clusters, while "islands of stability" can have very unexpected compositions. For example, at room temperature and ambient atmosphere, superoxidized  $Fe_4O_8$  clusters will be dominant among the  $Fe_4O_n$  clusters. We emphasize that stability is dictated not only by closed geometric and electronic shells, but also by magnetism.

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Nanoclusters possess a wide variety of physical and chemical properties, and peculiar electronic and geometric structures. However, their small size hampers full structural characterization by experimental techniques.<sup>1</sup> In fact, most of the structural models are generally inferred indirectly, often with the aid of theoretical approaches.<sup>2</sup> It was found that nanoclusters of certain sizes and shapes, known as 'magic clusters', possess enhanced stability. Such magic clusters can serve as building blocks to assemble bulk materials.<sup>3</sup> Their stability can be either structural (closed atomic shell – when atoms form a complete symmetric shell, *e.g.* icosahedral<sup>4</sup>) or electronic (due to closed electronic shell, precise electron count, or aromaticity). Stability due to a closed electronic shell is found, *e.g.*, in alkali metal clusters, which are well described by the jellium model<sup>5,6</sup> and behave as "superatoms" with collective electronic orbitals that form

electronic shells. In addition, new models of aromaticity have been put forward to explain enhanced stability of certain metal clusters, such as  $Al_n$  and  $Cr_n$ ,<sup>7</sup> and antiferromagnetism was also found to play an important role in stabilizing such clusters as  $Ti_8O_{12}$ .<sup>8</sup> Recently, it was reported that for the cage  $Fe_{12}O_{12}$  cluster, the antiferromagnetic state is much more stable (by  $\sim 1.25$  eV) than ferromagnetic.<sup>9</sup>

Full understanding of the intrinsic factors, which determine the stability of oxide clusters, is still lacking. The first step to this goal is to understand the relationship between stability of such clusters and chemical variables, such as composition and chemical environment. In this paper, we tackle this question by a systematic study of stable structures of  $Ce_mO_n$  and  $Fe_mO_n$  clusters under realistic conditions using evolutionary structure prediction. We identify magic  $Ce_mO_n$  and  $Fe_mO_n$  clusters using thermodynamic criteria and correlate stability with their electronic structure. Conclusions drawn from such study are applicable to other oxide clusters.

Cerium and iron oxide clusters play an important role in catalysis and environmental remediation. There have been a number of studies dedicated to investigating the geometric and electronic structures of transition metal oxide nanoclusters: neutral  $Ce_mO_{2m-x}$  ( $m \leq 4$ ,  $0 \leq x \leq m$ )<sup>10,11</sup> and  $Ce_mO_{2m}$  ( $m = 1-6$ )<sup>12,13</sup> clusters. Li *et al.*<sup>10</sup> found that Hubbard's *U* parameter strongly affects the electronic structure of oxygen-deficient  $Ce_nO_{2n-1}$  clusters. Structures of  $Fe_mO_n$ ,<sup>14-19</sup>  $(FeO)_k$ ,<sup>20</sup> and  $(Fe_2O_3)_k$ <sup>21,22</sup> clusters have also been studied. Ding *et al.*,<sup>21</sup> studying cage and noncage  $(Fe_2O_3)_k$  ( $k = 2-6, 10$ ) cluster, found that global minima have noncage structures in most cases. Recently, it was reported that the precise magnetic state of clusters has small

<sup>a</sup> Moscow Institute of Physics and Technology, 9 Institutskiy Lane, Dolgoprudny, Moscow Region, 141700, Russia. E-mail: yuxiaohu950203@126.com

<sup>b</sup> Institute of Theoretical and Computational Chemistry, Shaanxi Key Laboratory of Catalysis, School of Chemical & Environment Sciences, Shaanxi University of Technology, Hanzhong 723000, China

<sup>c</sup> Skolkovo Institute of Science and Technology, 3 Nobel St., 143026 Moscow, Russia

<sup>d</sup> International Center for Materials Discovery, Northwestern Polytechnical University, Xi'an 710072, China

<sup>e</sup> Department of Geosciences, Stony Brook University, Stony Brook, New York 11794, USA

<sup>f</sup> School of Electronic Engineering, Xidian University, Xi'an, Shaanxi, 710071, China

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influence on the geometric structure from investigating neutral  $(\text{Fe}_2\text{O}_3)_k$  ( $k = 1-5$ ) clusters,<sup>22</sup> and evolution of the structure of  $\text{Fe}_2\text{O}_3$  clusters from the smallest size to nanometer-sized  $\text{Fe}_2\text{O}_3$  particles was studied.<sup>23</sup> López *et al.*<sup>24</sup> studied the structure and electronic properties of iron oxide clusters using the GGA+*U* method. Palotás *et al.*<sup>25</sup> demonstrated the adequacy of GGA+*U* in their study of iron oxide clusters.

Structure predictions were performed using the *ab initio* evolutionary algorithm USPEX,<sup>26</sup> which has been successfully applied to various bulk materials (*e.g.*, ref. 27), 2D structures,<sup>28</sup> nanoclusters<sup>29,30,41</sup> and polymers.<sup>31</sup> In our nanoparticle structure predictions, an initial population of structures is produced using random symmetric generator.<sup>27</sup> All structures were relaxed and then ranked by energy, with lower-energy 60% of structures selected as parents for the new generation (by specially designed variation operators, such as heredity, permutation and soft-mutation). According to our tests, spin states play an important role in determining the stability of  $\text{Fe}_m\text{O}_n$  clusters and rather little role for  $\text{Ce}_m\text{O}_n$  clusters – therefore, a spin mutation operator was used for  $\text{Fe}_m\text{O}_n$  clusters and the initial magnetic moments were randomly assigned to iron atoms at the stage of producing the first generation of structures. Calculations proceeded until the lowest-energy structure remained unchanged for a sufficiently large number of generations, as was done in the previously published papers.<sup>8,9</sup> During global optimization, all structures were relaxed using the all-electron-projected-augmented wave method as implemented in the Vienna *ab initio* simulation package (VASP).<sup>32,33</sup> In these calculations, clusters were surrounded by a 12 Å thick vacuum. In order to decrease the computational cost, we used the Perdew–Burke–Ernzerhof (PBE) functional, with a Hubbard-like *U* correction (DFT+*U*), in the form developed by Dudarev *et al.*,<sup>34</sup> to account for the self-interaction error inherent to semilocal functionals such as PBE. We used an effective  $U_{\text{eff}} = U - J$  value of 5 eV<sup>10,11,35,36</sup> for the Ce atom and 4 eV for the Fe atom,<sup>25,37–39</sup> which have been shown to be a good choice for an accurate description of both the atomic and electronic structure of ceria and iron oxide. The energy cutoff for the plane-wave expansion was set to 400 eV. Structure relaxations were performed until the Hellmann–Feynman force on each atom became less than 0.001 eV Å<sup>-1</sup>. Vibrational frequency calculations were performed to check that the predicted structures have no imaginary frequencies. The hybrid functional (HSE06)<sup>40–42</sup> was also used to test the magnetic moments and HOMO–LUMO gaps of  $\text{Ce}_m\text{O}_n$  and  $\text{Fe}_m\text{O}_n$  clusters. It was found that the effect of different methods on the magnetic moments of  $\text{Ce}_m\text{O}_n$  and  $\text{Fe}_m\text{O}_n$  clusters is small, while the effect of different methods on the HOMO–LUMO gaps of  $\text{Ce}_m\text{O}_n$  and  $\text{Fe}_m\text{O}_n$  clusters is large, and the detailed test results are listed in Tables S1, S2, and Fig. S1, S2 in the ESI.† However, the results of HOMO–LUMO gaps agree well with previous results: hybrid functionals usually predict a larger gap than the DFT methods.<sup>43</sup> One can easily see that stoichiometric  $\text{Fe}_m\text{O}_n$  and  $\text{Ce}_m\text{O}_n$  clusters have larger HOMO–LUMO gaps than other clusters for both different methods from Fig. S1 and S2 (ESI†). Note that there are no experimental HOMO–LUMO gaps for nonstoichiometric  $\text{Ce}_m\text{O}_n$  and  $\text{Fe}_m\text{O}_n$  clusters to compare with our DFT+*U* gaps.

We considered a wide range of  $\text{Ce}_m\text{O}_n$  and  $\text{Fe}_m\text{O}_n$  cluster sizes:  $m$  from 1 to 10 and  $n$  in the  $n = 0-14$  and  $n = 0-20$  ranges for small ( $m = 1-6$ ) and larger ( $m > 6$ ) clusters, respectively. For each stoichiometry, the energy was minimized with respect to both geometry and spin state.

We evaluated the Gibbs free energy<sup>44–47</sup> as  $\Delta G_{\text{f}}(T, p_{\text{O}_2}) = F_{\text{M}_m\text{O}_n}(T) - F_{\text{M}_m}(T) - n\mu_{\text{O}}(T, p_{\text{O}_2})$ , where  $F_{\text{M}_m\text{O}_n}(T)$  and  $F_{\text{M}_m}(T)$  are the Helmholtz free energy of the  $\text{M}_m\text{O}_n$  ( $\text{Ce}_m\text{O}_n$  and  $\text{Fe}_m\text{O}_n$ ) and the pristine  $\text{M}_m$  ( $\text{Ce}_m$  and  $\text{Fe}_m$ ) clusters (at their ground state with respect to geometry and spin), respectively, and  $\mu_{\text{O}}(T, p_{\text{O}_2})$  is the chemical potential of oxygen. In equilibrium with molecular  $\text{O}_2$  gas,  $\mu_{\text{O}}(T, p_{\text{O}_2})$  is expressed as  $\mu_{\text{O}}(T, p_{\text{O}_2}) = 1/2[E_{\text{O}_2} + \mu_{\text{O}_2}(T, p^0) + k_{\text{B}}T \ln(p/p^0)]$ , where  $p^0$ ,  $k_{\text{B}}$ , and  $p$  are the standard atmospheric pressure, Boltzmann constant, and oxygen partial pressure, respectively.  $E_{\text{O}_2}$  was obtained from a spin-polarized calculation. The  $\mu_{\text{O}_2}(T, p^0)$  term includes contributions from spin flips, vibrations and rotations of  $\text{O}_2$  molecules at atmospheric pressure and was taken from thermodynamic tables.<sup>48</sup>  $k_{\text{B}}T \ln(p/p^0)$  is the contribution of temperature and  $\text{O}_2$  partial pressure to the oxygen chemical potential. The phase diagram for each  $m$  was constructed by identifying the lowest-free-energy structures at each  $(T, p_{\text{O}_2})$ .

Fig. 1 shows the computed phase diagrams of  $\text{Ce}_4\text{O}_n$  and  $\text{Fe}_4\text{O}_n$ . Ambient conditions and lower temperature or higher oxygen pressure favor nonstoichiometric oxygen-rich clusters  $\text{Ce}_4\text{O}_n$  ( $n > 8$ ) and  $\text{Fe}_4\text{O}_n$  ( $n > 6$ ). In contrast, oxygen-deficient clusters are stabilized by high  $T$  and low  $p_{\text{O}_2}$ . Oxygen desorption

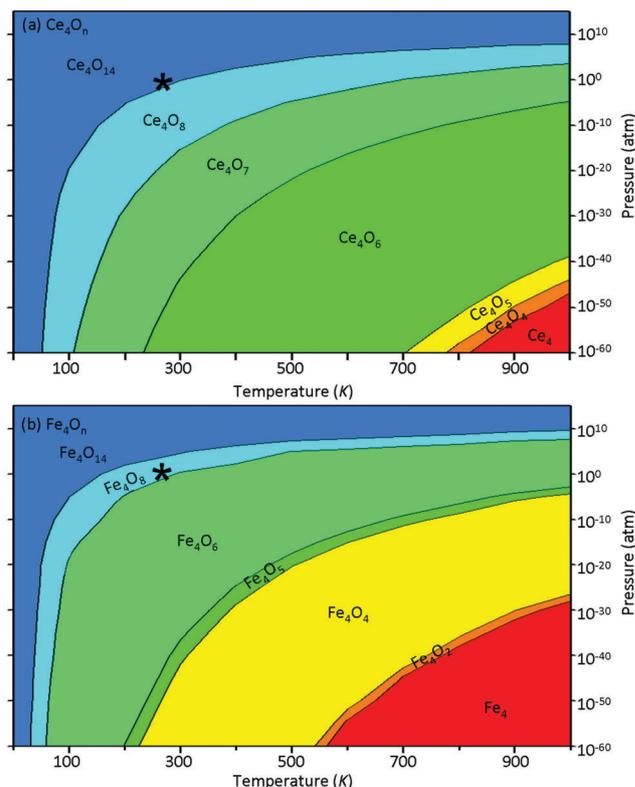


Fig. 1 Phase diagrams of  $\text{Ce}_4\text{O}_n$  (a) and  $\text{Fe}_4\text{O}_n$  (b) in an oxygen atmosphere. The star “\*” shows standard conditions.

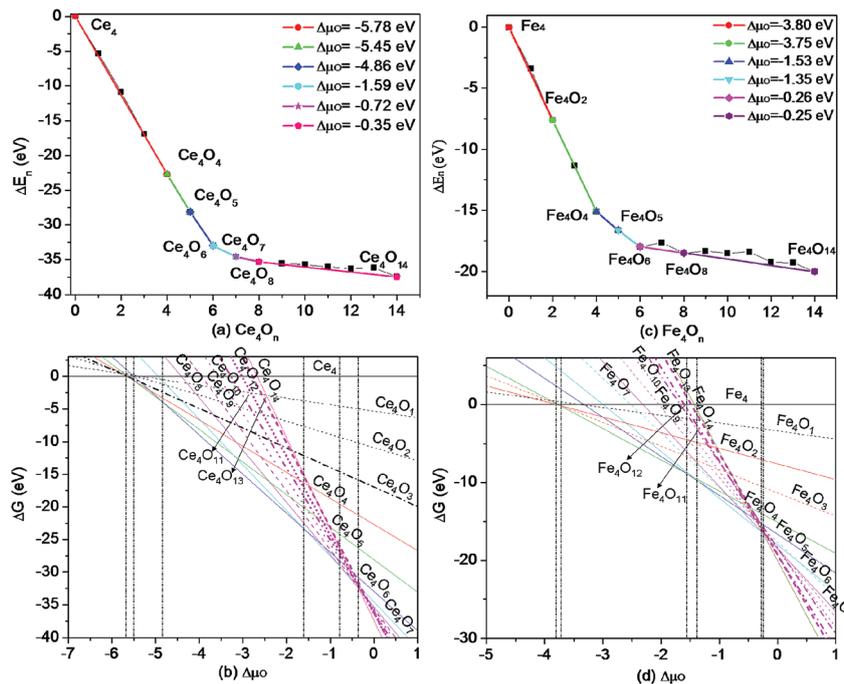


Fig. 2 (a) Convex hull and (b) phase diagram of  $\text{Ce}_4\text{O}_n$  ( $n = 0-14$ , number of oxygen atoms); (c) convex hull and (d) phase diagram of  $\text{Fe}_4\text{O}_n$  ( $n = 0-14$ ) (solid line means stable cluster-refers to the phase diagrams [i.e. (b) and (d)]).

at high temperatures is easy to understand: oxygen gas has very high entropy, and its formation is favorable at sufficiently high temperatures. Existing experimental results for  $(\text{Fe}_2\text{O}_3)_k$  and  $(\text{FeO})_k$  clusters are consistent with our predictions.<sup>49–52</sup> Free energies of formation of the most stable nonstoichiometric (a)  $\text{Ce}_m\text{O}_n$  ( $n \neq 2m$ ) and (b)  $\text{Fe}_m\text{O}_n$  ( $n \neq m$ ) clusters, relative to stoichiometric (a)  $\text{Ce}_m\text{O}_n$  ( $n = 2m$ ) and (b)  $\text{Fe}_m\text{O}_n$  ( $n = m$ ) clusters under several  $(T, p_{\text{O}_2})$  conditions are shown in Fig. S3 (ESI†).

Spin density distribution in  $\text{Ce}_4\text{O}_n$  clusters is shown in Fig. S4 (ESI†). For  $\text{Ce}_4\text{O}_8$  clusters, the ground state is singlet, while in  $\text{Ce}_4\text{O}_7$  and  $\text{Ce}_4\text{O}_6$  clusters there are two and four, respectively, unpaired electrons localized on f-orbitals – just as many as  $\text{Ce}^{3+}$  atoms. We found this to be a general trend: the  $\text{Ce}_n\text{O}_{2n}$  and  $\text{Ce}_n\text{O}_{2n-1}$  clusters are in singlet and triplet states, respectively. By changing  $T$  and partial pressure of oxygen, one can change the stoichiometry and magnetic state of clusters.

On the other hand,  $\text{Fe}_4$  is ferromagnetic,  $\text{Fe}_4\text{O}_1$  is ferrimagnetic, and the other lowest-energy  $\text{Fe}_4\text{O}_n$  clusters are all antiferromagnetic (Fig. S5 and S6, ESI†). Exchange interaction between non-neighboring magnetic ions that have the same oxidation state is known as the superexchange effect,<sup>53</sup> while the exchange interaction between non-neighboring magnetic ions with different oxidation states can be termed the double exchange effect.<sup>54</sup> Thus the ferrimagnetic state of  $\text{Fe}_4\text{O}_n$  clusters is induced by the competition between ferromagnetism (due to direct exchange) and antiferromagnetism (induced by superexchange or double exchange) – the latter wins in oxygen-rich clusters. For example, superexchange effect plays an important role in the antiferromagnetic  $\text{Fe}_4\text{O}_4$  cluster with all four Fe atoms in the +2 oxidation state, while double exchange effect wins in the antiferromagnetic  $\text{Fe}_4\text{O}_5$  cluster with two Fe atoms

in the +2 oxidation state and other two Fe atoms in the +3 oxidation state. The structure of the  $\text{Fe}_4\text{O}_6$  cluster is made of a tetrahedral  $\text{Fe}_4$  core and six O atoms on each of the six edges of the tetrahedron – which agrees well with available experimental and theoretical results.<sup>17</sup> Our calculations show that spin state is a very important factor determining the stability of  $\text{Fe}_m\text{O}_n$  clusters (Fig. S6, ESI†).

Convex hulls and phase diagrams of  $\text{Ce}_4\text{O}_n$  and  $\text{Fe}_4\text{O}_n$  clusters are shown in Fig. 2. In the convex hulls (Fig. 2a and c), each of clusters is represented as a point, and the points corresponding to stable clusters form a convex hull. In the phase diagrams (Fig. 2b and d), each cluster is represented by a straight line which represents its Gibbs free energy as a function of the chemical potential of oxygen, and stable clusters are those that have the lowest Gibbs free energy at a certain range of chemical potentials. There are seven stable clusters appearing on the convex hull in Fig. 2a:  $\text{Ce}_4$ ,  $\text{Ce}_4\text{O}_4$ ,  $\text{Ce}_4\text{O}_5$ ,  $\text{Ce}_4\text{O}_6$ ,  $\text{Ce}_4\text{O}_7$ ,  $\text{Ce}_4\text{O}_8$  and  $\text{Ce}_4\text{O}_{14}$ , consistent with Fig. 1a and 2b. For  $\text{Fe}_4\text{O}_n$  clusters, there are six stable clusters (Fig. 2c and d):  $\text{Fe}_4$ ,  $\text{Fe}_4\text{O}_2$ ,  $\text{Fe}_4\text{O}_4$ ,  $\text{Fe}_4\text{O}_5$ ,  $\text{Fe}_4\text{O}_6$ ,  $\text{Fe}_4\text{O}_8$  and  $\text{Fe}_4\text{O}_{14}$ , consistent with Fig. 1b.

Another interesting feature is that nonstoichiometric  $\text{Ce}_m\text{O}_n$  ( $n > 2m$ ) and  $\text{Fe}_m\text{O}_n$  ( $2n > 3m$ ) clusters have peroxo- and superoxo-groups (e.g., in  $\text{Fe}_4\text{O}_9$  and  $\text{Fe}_4\text{O}_{12}$  – Fig. S7, ESI†), with peroxo-groups attached side-on  $[\text{Fe}(\text{O})_2]$  and superoxo-groups attached end-on (FeOO).<sup>55</sup> Recently, the same phenomenon was reported for  $\text{Si}_m\text{O}_n$  ( $n > 2m$ ) clusters.<sup>56</sup> The thermodynamically favored excess of oxygen in nanoclusters is in sharp contrast with the bulk crystal, where the stoichiometric composition is strongly favored. In contrast to the nonstoichiometric clusters, stoichiometric clusters  $(\text{CeO}_2)_k$  are always singlet (non-magnetic), while both the nonstoichiometric and stoichiometric  $\text{Fe}_m\text{O}_n$  clusters are antiferromagnetic or ferrimagnetic.

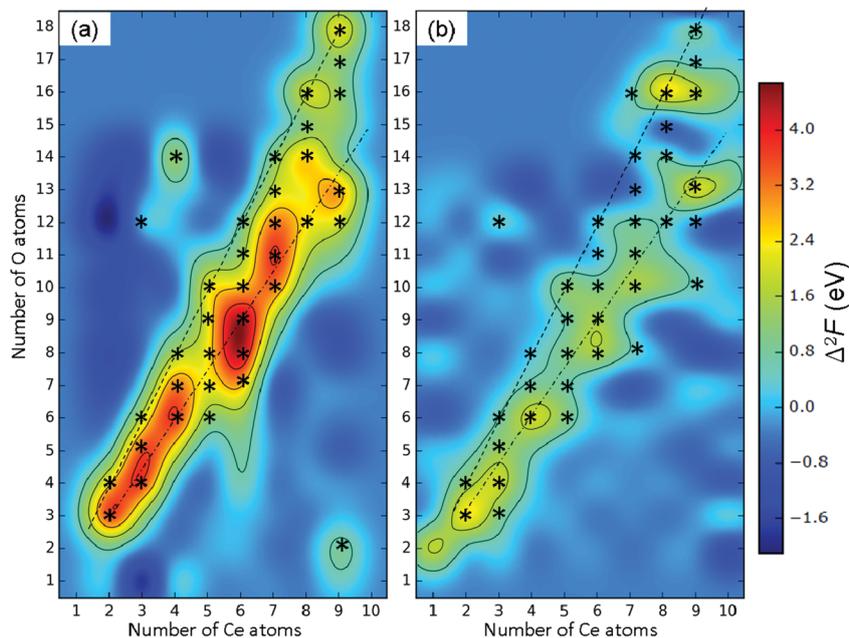


Fig. 3 Stability of  $\text{Ce}_m\text{O}_n$  clusters (shown by stars) determined by second energy differences (a)  $\Delta^2F_m = F(\text{Ce}_{m-1}\text{O}_n) + F(\text{Ce}_{m+1}\text{O}_n) - 2F(\text{Ce}_m\text{O}_n)$  and (b)  $\Delta^2F_n = F(\text{Ce}_m\text{O}_{n-1}) + F(\text{Ce}_m\text{O}_{n+1}) - 2F(\text{Ce}_m\text{O}_n)$ . Clearly,  $(\text{CeO}_2)_k$  and  $(\text{Ce}_2\text{O}_3)_k$  clusters are doubly magic. This map shows ridges and islands of stability, surrounded by the “sea of instability”, and allows one to find feasible and infeasible chemical transformations. For example,  $\text{Ce}_3\text{O}_{12}$  is a doubly magic cluster, but its synthesis requires simultaneous addition of six oxygen atoms to the nearest doubly magic cluster  $\text{Ce}_3\text{O}_6$ , which will be a rare process.

To evaluate the stability of clusters as a function of chemical composition, we considered the second energy difference defined from the energy of formation:

$$\Delta^2F_m = F(\text{M}_{m-1}\text{O}_n) + F(\text{M}_{m+1}\text{O}_n) - 2F(\text{M}_m\text{O}_n) \quad (1)$$

$$\Delta^2F_n = F(\text{M}_m\text{O}_{n-1}) + F(\text{M}_m\text{O}_{n+1}) - 2F(\text{M}_m\text{O}_n) \quad (2)$$

where the energy of formation is defined as:

$$F(\text{M}_m\text{O}_n) = E(\text{M}_m\text{O}_n) - mE(\text{M}) - 1/2nE(\text{O}_2) \quad (3)$$

$E(\text{M}_m\text{O}_n)$  is the energy of the  $\text{M}_m\text{O}_n$  cluster and  $E(\text{M})$  is the energy of a single metal atom. A positive second energy difference indicates a magic  $\text{M}_m\text{O}_n$  cluster: in this case, a homogeneous ensemble of identical clusters is stable against disproportionation by transfer of an atom from one cluster to another. Positiveness of  $\Delta^2F$  with respect to both types of atoms indicates that the cluster is doubly magic – suggesting that it should be abundant, less reactive than neighboring clusters, and well defined in terms of its geometric and electronic structure. This language

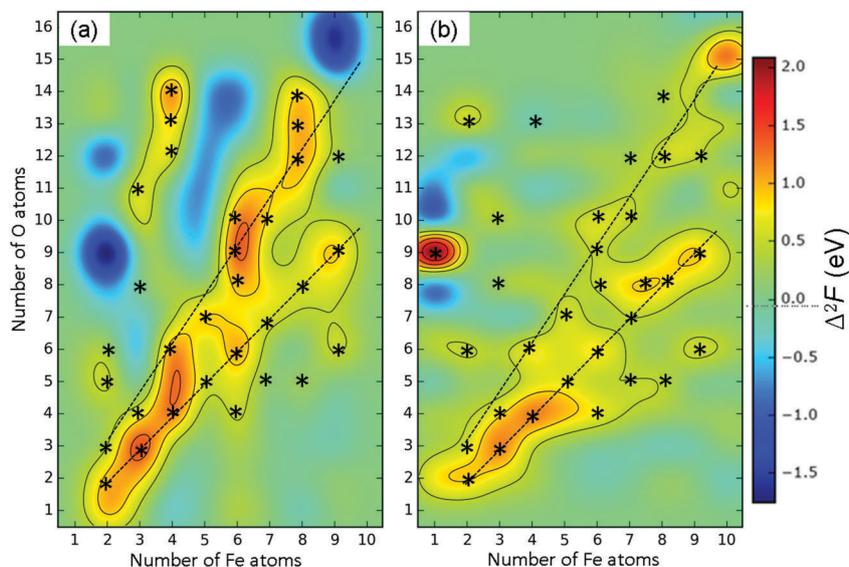


Fig. 4 Stability of  $\text{Fe}_m\text{O}_n$  clusters (shown by stars) as determined by second energy differences (a)  $\Delta^2F_m = F(\text{Fe}_{m-1}\text{O}_n) + F(\text{Fe}_{m+1}\text{O}_n) - 2F(\text{Fe}_m\text{O}_n)$  and (b)  $\Delta^2F_n = F(\text{Fe}_m\text{O}_{n-1}) + F(\text{Fe}_m\text{O}_{n+1}) - 2F(\text{Fe}_m\text{O}_n)$ . Clearly,  $(\text{FeO})_k$ ,  $(\text{Fe}_2\text{O}_3)_k$  and  $(\text{Fe}_3\text{O}_4)_k$  clusters are doubly magic.

(singly magic and doubly magic clusters) is similar to the terminology used for describing stable atomic nuclei – singly magic have particular stability with respect to the number of either protons or neutrons, and doubly magic are stable with respect to both. In this language, our above consideration of stable  $Ce_4O_n$  and  $Fe_4O_n$  clusters with fixed numbers of metal atoms corresponds to finding singly magic clusters (with respect to the number of oxygen atoms). Adding stability with respect to the number of metal atoms indicates only  $Ce_4O_6$ ,  $Ce_4O_7$  and  $Ce_4O_8$  clusters as doubly magic among all  $Ce_4O_n$  clusters – and only  $Fe_4O_4$ ,  $Fe_4O_6$  and  $Fe_4O_{13}$  among the  $Fe_4O_n$  clusters. Doubly magic clusters feature most prominently on the phase diagrams (Fig. 1).

Fig. 3 shows the second energy difference for  $Ce_mO_n$  clusters. Clusters with large positive second energy differences are particularly stable. We can clearly see that clusters in  $(CeO_2)_k$  and  $(Ce_2O_3)_k$  series are doubly magic, due to their electronic structure. It is well known that clusters with large gaps between the highest occupied molecular orbitals (HOMOs) and lowest unoccupied molecular orbitals (LUMOs) exhibit lower reactivity, because such clusters are more reluctant to either accept or donate an electron. Consequently, the HOMO–LUMO gap is a simple marker of less reactive, and consequently, chemically more stable clusters that have larger chances of surviving in complex chemical environments. As is well known, Ce atoms have two oxidation states,  $Ce^{3+}$  and  $Ce^{4+}$ , and clusters possessing only these states are expected to have large HOMO–LUMO gaps. In order to analyze the stability of  $(CeO_2)_k$  and  $(Ce_2O_3)_k$  clusters, we calculated the HOMO–LUMO gaps of stable clusters (Fig. 5a). The HOMO–LUMO gaps of  $(CeO_2)_k$  clusters calculated using the GGA+ $U$  method are all between 1.50–2.30 eV, except  $CeO_2$  (0.3 eV). The HOMO–LUMO gaps of  $(Ce_2O_3)_k$  series are in the range 1.2–1.7 eV. Due to the stable  $Ce^{3+}$  state,  $Ce_nO_{2n-1}$  clusters also have large second energy differences and large HOMO–LUMO gaps (1.3–1.8 eV).

In one type of geometrically magic clusters, we find denser clusters similar to bulk crystal structures: for example, doubly magic clusters  $Ce_3O_4$ ,  $Ce_6O_8$ ,  $Ce_7O_{10}$ ,  $Ce_8O_{14}$ ,  $Ce_9O_{16}$ , and so on (Fig. S8, ESI†). In particular, the  $Ce_6O_8$  cluster is the smallest octahedral cluster with exposed (111) facets and possesses unique d-orbital spherical aromaticity.<sup>57</sup> In experiments, transmission electron microscopy revealed nanoparticles of octahedral and cuboctahedral shapes, with (111) and (100) facets.<sup>58</sup> Another group of geometrically magic clusters cannot be represented as fragments of bulk structures, but still are characterized by high symmetry, for example:  $Ce_5O_6$ ,  $Ce_5O_7$ ,  $Ce_5O_8$ ,  $Ce_6O_9$ ,  $Ce_6O_{10}$ ,  $Ce_7O_{11}$ ,  $Ce_7O_{12}$ ,  $Ce_9O_{12}$  and  $Ce_9O_{16}$  (Fig. S8, ESI†). Stable at high oxygen fugacities, the  $Ce_3O_{12}$  cluster is also doubly magic; its structure contains three peroxides  $O_2^{2-}$  adsorbed on the three terminal Ce atoms, two superoxides  $O_2^-$  occupying bridging positions between Ce atoms, and two oxide  $O^{2-}$  ions adsorbed above a three-hollow site and a Ce–Ce bridge.

Generally, the addition of O atoms to  $(CeO_2)_n$  clusters leads to a decrease of the average oxygen binding energy (Fig. S9b and S11b, ESI†) and the formation of O–O bonds. Destabilization is

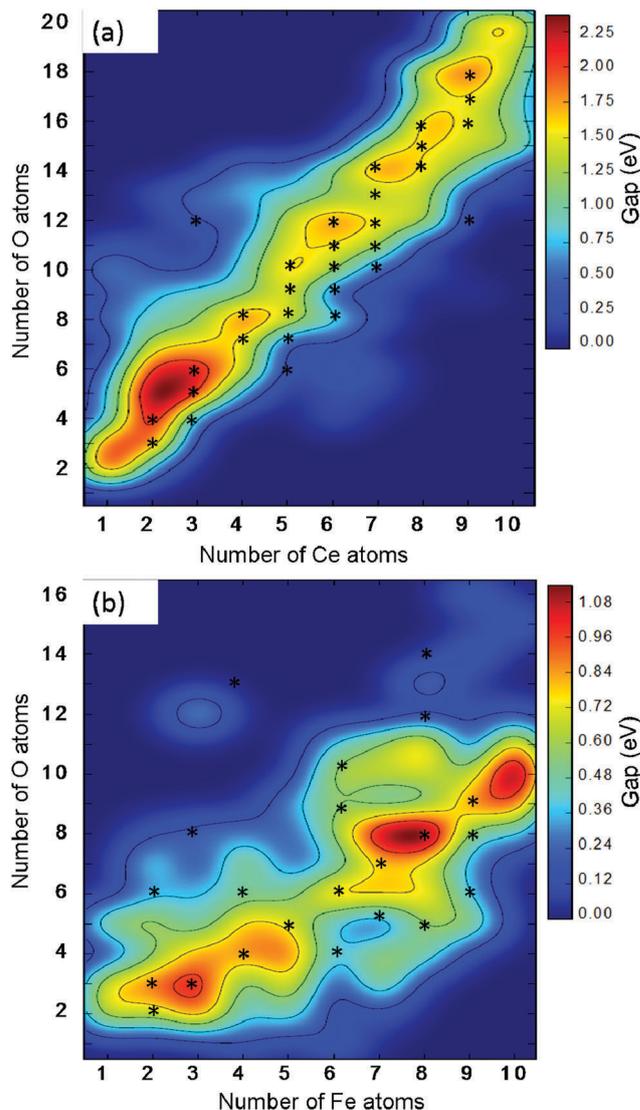


Fig. 5 HOMO–LUMO gap as a function of composition (a)  $Ce_mO_n$  (b)  $Fe_mO_n$ . The large ridge and small islands of stability can be easily seen.

signaled by positive values of the stepwise oxygen binding energy (Fig. S9a and S11a, dotted line, ESI†).

Fig. 4 shows the second energy difference of  $Fe_mO_n$  clusters. We can see that clusters in  $(FeO)_k$  and  $(Fe_2O_3)_k$  series are stable. Fe has two oxidation states,  $Fe^{2+}$  and  $Fe^{3+}$ ; and clusters based on these states are magic. For  $(FeO)_k$  clusters, we have computed spin gaps as  $\Delta_1 = -(\epsilon_{HOMO}^+ - \epsilon_{LUMO}^-)$  and  $\Delta_2 = -(\epsilon_{HOMO}^- - \epsilon_{LUMO}^+)$  (+, spin up; –, spin down), corresponding to the energy required to move an electron from the HOMO of one spin to the LUMO of the other. Large positive values for both spin gaps can guarantee that the system is a stable magnetic and electronic state.<sup>59,60</sup> The spin gaps of  $(FeO)_k$  ( $k = 3-10$ ) are all positive (0.2–0.95 eV).  $(Fe_2O_3)_k$  clusters also have a positive spin gap.  $(Fe_3O_4)_k$  clusters are particularly stable among the  $Fe_mO_n$  clusters, as can be seen from their large  $\Delta^2F$ , suggesting that  $(Fe_3O_4)_k$  nanoparticles may better survive in chemically complex environments, and this can be attributed to the wide abundance of magnetite  $Fe_3O_4$  in nature.

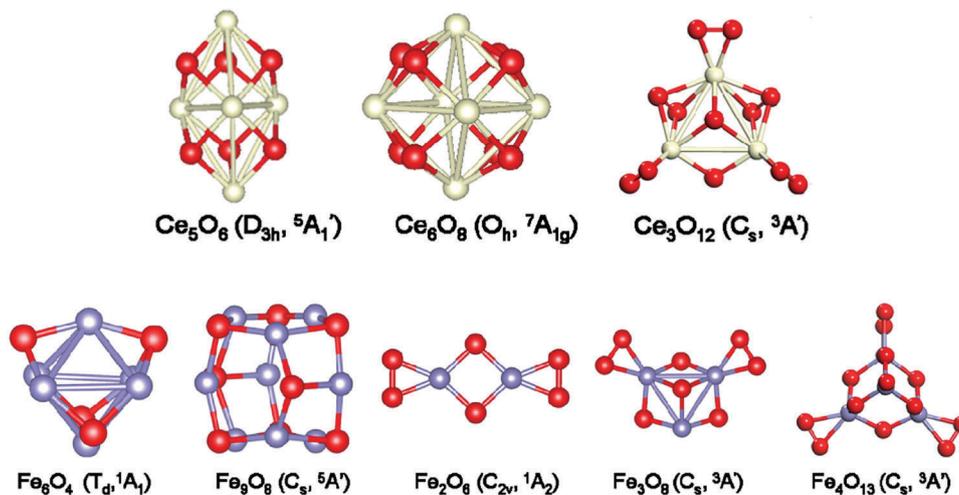


Fig. 6 Representative examples of doubly magic non-stoichiometric clusters: metal-rich  $\text{Ce}_5\text{O}_6$ ,  $\text{Ce}_6\text{O}_8$ ,  $\text{Fe}_6\text{O}_4$  and  $\text{Fe}_9\text{O}_8$ , and oxygen-rich  $\text{Ce}_3\text{O}_{12}$ ,  $\text{Fe}_3\text{O}_8$  and  $\text{Fe}_4\text{O}_{13}$ . Metal-rich clusters feature metal–metal bonds (which can be delocalized over the entire cluster, as in  $\text{Ce}_6\text{O}_8$ <sup>57</sup>), while oxygen-rich (superoxidized) clusters feature O–O bonds in peroxo- and superoxo-groups.

Similar to  $\text{Ce}_m\text{O}_n$  clusters, some magic clusters  $\text{Fe}_m\text{O}_n$  are determined by geometry, either deriving from the bulk structure or other high-symmetry atomic arrangements with filled geometric shells. For example, the following magic clusters have structures different from the bulk, but with high symmetry:  $\text{Fe}_6\text{O}_4$ ,  $\text{Fe}_7\text{O}_5$ ,  $\text{Fe}_8\text{O}_5$ ,  $\text{Fe}_8\text{O}_{14}$ ,  $\text{Fe}_9\text{O}_6$ , and  $\text{Fe}_9\text{O}_8$  (Fig. S7, ESI†). The doubly magic cluster  $\text{Fe}_9\text{O}_6$  is in agreement with previous theoretical and experimental reports.<sup>61</sup> Among the clusters with higher oxygen concentration,  $\text{Fe}_2\text{O}_6$ ,  $\text{Fe}_3\text{O}_8$  and  $\text{Fe}_4\text{O}_{13}$  clusters have both large  $\Delta^2F_m$  and  $\Delta^2F_n$ , and are therefore doubly magic clusters as well. Structures of some non-stoichiometric doubly magic  $\text{Fe}_m\text{O}_n$  and  $\text{Ce}_m\text{O}_n$  clusters are shown in Fig. 6. Suboxide clusters (with  $\text{Fe}/\text{O} > 1$  and  $\text{Ce}/\text{O} > 2/3$ ) typically feature (often multicenter) metal–metal bonds and can be expected to be excellent reducing agents. Superoxidized clusters ( $\text{Fe}/\text{O} < 2/3$  and  $\text{Ce}/\text{O} < 1/2$ ) feature O–O bonds and should be powerful oxidants; these are predicted to be abundant at room temperature and atmospheric oxygen pressure in  $\text{Ce}_m\text{O}_n$  and  $\text{Fe}_m\text{O}_n$  (Fig. 1), a situation recently discovered for  $\text{Si}_m\text{O}_n$  clusters,<sup>56</sup> and probably typical of most oxide clusters, and due to the presence of reactive oxygen species (peroxide, superoxide and ozonide groups) such clusters can display biotoxicity and carcinogenicity ( $\text{Si}_m\text{O}_n$ ,<sup>56</sup> and  $\text{Ti}_m\text{O}_n$ <sup>62</sup>).

To summarize, we employed the state-of-the-art evolutionary structure prediction algorithm USPEX, DFT+*U* calculations, and *ab initio* thermodynamics to study transition metal oxide clusters. For the first time, we systematically investigated magic clusters in such systems and singled out those compositions which become “magic” (*i.e.* conditionally thermodynamically stable) at a given temperature and oxygen pressure. We found that the most stable stoichiometric  $\text{Ce}_n\text{O}_{2n}$  clusters have a closed-shell electronic configuration (singlet), in contrast to nonstoichiometric  $\text{Ce}_n\text{O}_{2n-x}$  clusters: for example,  $\text{Ce}_n\text{O}_{2n-1}$  clusters have triplet state with the magnetic moment localized on the Ce atoms reduced from  $\text{Ce}^{4+}$  to  $\text{Ce}^{3+}$ . This suggests the possibility of tuning the

magnetic properties by changing partial oxygen pressure and temperature. For  $\text{Fe}_m\text{O}_n$  clusters, all  $\text{Fe}_m$  clusters are ferromagnetic, low-oxygen  $\text{Fe}_m\text{O}_n$  clusters are ferri- or antiferromagnetic, and high-oxygen  $\text{Fe}_m\text{O}_n$  clusters are all antiferromagnetic when  $m = \text{even}$  and ferrimagnetic when  $m = \text{odd}$ . This indicates that a competition between ferromagnetism (induced by direct exchange) and antiferromagnetism (induced by superexchange or double exchange) plays an important role in determining the stability of low-oxygen  $\text{Fe}_m\text{O}_n$  clusters. At high partial pressures of oxygen, dominant magic nanoparticles contain excess oxygen as peroxide and superoxide groups attached to their surface. We can also notice that magic clusters form ridges and islands of stability, surrounded by the “sea of instability” in the compositional space – a picture very similar to what is found for atomic nuclei (which can also be described as clusters made of two types of particles, protons and neutrons). We expect that the topology of the ridges and islands of stability should be an important factor determining the feasibility or infeasibility of a particular chemical transformation of a nanoparticle.

## Conflicts of interest

The authors declare no competing financial interest.

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