## Unexpected Reconstruction of the $\alpha$ -Boron (111) Surface

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We report a novel reconstruction of the  $\alpha$ -boron (111) surface, discovered using *ab initio* evolutionary structure prediction, and show that this unexpected neat structure has a much lower energy than the recently proposed (111)- $I_{R,(a)}$  surface. In this reconstruction, all single interstitial boron atoms bridge neighboring  $B_{12}$  icosahedra by polar covalent bonds, and this satisfies the electron counting rule, leading to the reconstruction-induced metal-semiconductor transition. The peculiar charge transfer between the interstitial atoms and the icosahedra plays an important role in stabilizing the surface.

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The element boron had long attracted enormous attention owing to its fascinating properties, such as exceptional structural complexity, superhardness, unusual partially ionic bonding, and superconductivity at high pressure [1-5]. As carbon's neighbor in the periodic table, boron is in many ways an analog of carbon and its nanostructures (clusters, nanotubes, nanowires, nanobelts, fullerenes, and so on) have aroused extensive interest, in the hopes of replicating or even surpassing the unique properties and diversity of the forms of carbon [6-12]. By analogy with graphene [13,14], two-dimensional boron sheets with triangular and hexagonal motifs are predicted to be the most stable phases and likely precursors for boron nanostructures [15-22]. However, buckled bilayer structures appeared to be massively more stable; some of them turned out to have novel electronic properties, such as a distorted Dirac cone [23]. Hayami and Otani systematically studied the energies of low-index bare surfaces in the  $\alpha$ -boron,  $\beta$ -boron, and two tetragonal phases (*t*-I and *t*-II), which suggested that *t*-I and *t*-II can be more stable than  $\alpha$ -boron and  $\beta$ -boron for sufficiently small nanoparticles [24,25]. Amsler et al. performed the first study of the reconstruction of the  $\alpha$ -boron (111) and predicted several low-energy surface reconstructions by using the minima hopping method. In particular, a metallic reconstructed phase of (111)- $I_{R,(a)}$  was predicted to be the most stable configuration, where a conducting boron sheet was adsorbed on a semiconducting substrate, leading to numerous possible applications in nanoelectronics [26]. However, this seems to be in conflict with the general principle that the reconstructions usually lower their energies by atomic rearrangement leading to a semiconducting (as opposed to metallic) surface state [27]. Such an unexpected metallic reconstruction encourages us to explore other likely reconstructions and the stabilization mechanisms by first-principles calculations.

 $\alpha$ -boron structure is composed of  $B_{12}$  icosahedra [28], has two inequivalent atomic sites, polar  $(B_p)$  and equatorial  $(B_e)$  sites, the  $B_p$  atoms form upper and lower triangles of an icosahedron, and the  $B_e$  atoms form a slightly nonplanar hexagon along the equator of an icosahedron [2]. The arrangement of icosahedra in  $\alpha$ -boron can be described as a cubic close packing with the layer sequence ABC [1]. Compared with the (111)- $I_{R,(a)}$  surface built along the [111] direction of the primitive rhombohedral cell with surface vectors  $\mathbf{U}(2\overline{1}\ \overline{1})$  and  $\mathbf{V}(11\overline{2})$ , we cleaved the surface along the [111] direction with surface vectors  $\mathbf{U}(\bar{1}\,\bar{1}\,2)$  and  $V(1\overline{1}0)$ , which allowed us to reduce the computational costs drastically. The calculations were based on four layers of  $B_{12}$  icosahedra in the (111)-I substrate, and then the (111)-II substrate was also tested [25,26]. In both cases we obtained exactly the same reconstruction, regardless of the type of substrate if enough atoms and sufficient thickness were used. Searches for the stable reconstruction were performed using the ab initio evolutionary algorithm USPEX [29–31], which has been successfully applied to various materials (e.g., [32–34]). The number of surface atoms was allowed to vary from 1 to 20, constrained within a surface layer of thickness 4 Å, and we used vacuum regions of 10 Å thickness. Given that the thickness of  $B_{12}$  icosahedron is  $\sim$ 3.7 Å, there is enough space to fully explore the chemical landscape (more than 2500 structures were sampled, to achieve full confidence in the final result) in our calculations. The all-electron projector-augmented wave method [35] was employed, as implemented in the Vienna *ab initio* simulation package (VASP) [36], in combination with the generalized gradient approximation (GGA) and functional of Perdew, Burke, and Ernzerhof (PBE) [37]. A plane-wave



FIG. 1 (color online). (a) Projection of the  $2 \times 2 \times 1$  supercell of the (111)- $I_{R,(z)}$  structure along the [111] direction. (b) Projection of the  $2 \times 2 \times 1$  supercell of the (111)- $I_{R,(z)}$  structure along the  $[\bar{1}\ \bar{1}\ 2]$  direction. The inequivalent surface atoms are shown by different colors.

cutoff energy of 500 eV and for sampling the Brillouin zone we used uniform gamma-centered *k*-points grids with resolution  $2\pi \times 0.04$  Å<sup>-1</sup>. In addition, the hybrid HSE06 functional with the screening parameter ( $\omega$ ) 0.2 Å<sup>-1</sup> was also employed to check the robustness of surface energies [38].

For the (111)- $I_{R,(a)}$  reconstruction, proposed by Amsler et al., there is buckling and coupling among three outer atomic layers above the icosahedral  $B_{12}$  units, which results in structural complexity [26]. In contrast to (111)- $I_{R,(a)}$ , our reconstruction [designated as (111)- $I_{R,(z)}$ ] has an unexpected neat structure, as shown in Fig. 1(a), where a single boron atom occupies the interstitial position (named as  $B_i$ , colored in red), connects the  $B_p$  atoms (colored in blue), and forms bridges with bond lengths of 1.793 Å and bond angles of 113.06°. The  $B_i$  atoms are slightly above the topmost icosahedral atoms ( $B_p$ ), see Fig. 1(b), and together

they form the modulated "3 + 9" membered rings on the topmost surface (the calculated lattice constants and atomic positions of the related surfaces are given in the Supplemental Material [39]). We have computed the surface energy as  $\sigma = (1/2A)$  ( $N\epsilon_{\text{bulk}} - E_{\text{slab}}$ ), where A denotes the surface area,  $\epsilon_{\text{bulk}}$  is the energy per atom in the bulk  $\alpha$ -boron, and  $E_{slab}$  is the energy of the substrate containing N atoms [26]. Surface energies of (111)-I and (111)- $I_{R,(a)}$  obtained using the GGA-PBE and HSE06 functionals (Table I), are in excellent agreement with previous results [26]. This establishes the reliability and accuracy of our calculations. Strikingly, the surface energy of (111)- $I_{R,(z)}$  is 128.23 meV/Å<sup>2</sup> for GGA-PBE and 136.79 meV/Å<sup>2</sup> for HSE06, which is considerably, by 42 and 60 meV/Å<sup>2</sup>, respectively, lower than the energy of the (111)- $I_{R,(a)}$  surface reconstruction. As an additional check, we performed a structure search with the same substrate proposed in Ref. [26], and obtained the same reconstruction and energy, and found that there is no dependence on the choice of the surface vectors or cleavage plane: it is the general rule that the ratio of  $B_i$  atoms to the exposed  $B_{12}$  icosahedra should be 1:1 on the  $\alpha$ -boron (111) surface.

Figure 2 shows band structures of the unreconstructed (111)-I and the reconstructed (111)- $I_{R,(z)}$  surfaces from the GGA-PBE calculations. Because of the unsaturated dangling bonds, the (111)-I surface is metallic, as shown in Fig. 2(a). In contrast, Figure 2(b) shows that the (111)- $I_{R,(z)}$ surface is semiconducting, with a direct density functional theory band gap of 1.13 eV compared with the band gap of 1.50 eV of bulk  $\alpha$ -boron. The reconstruction-induced metalsemiconductor transition is consistent with the general principle that reconstructions usually lower their energies by atomic rearrangements that lead to a semiconducting surface [27]. According to Wade's rule [42,43], a  $B_{12}$ icosahedron has 36 valence electrons, 26 of which may be used for intraicosahedral bonds and 10 for intericosahedral bonds. Each icosahedron forms six two-electron-twocenter (2e2c) bonds with the icosahedra of neighboring layers, which requires  $6 \times 2/2 = 6$  electrons, as well as six closed two-electron-three-center (2e3c) bonds with the neighboring icosahedra in its own layer, these multicenter bonds require  $6 \times 2/3 = 4$  electrons [1]. The (111)-I surface cuts three intericosahedral bonds (2e2c) per

TABLE I. Surface energies of the unreconstructed (111)-I, reconstructed (111)- $I_{R,(a)}$ , and (111)- $I_{R,(z)}$  structures obtained using different functionals, in units of meV/Å<sup>2</sup>.

Surface	(111)-I	$(111)-I_{R,(a)}$	(111)- $I_{R,(z)}$	Reference
PBE	219.29	170.64	128.23	This work
	218.80	170.61		26
HSE	248.43	197.39	136.79	This work
	247.50	196.31		26



FIG. 2 (color online). Band structures of (a) (111)-I and (b) (111)- $I_{R,(z)}$ , the special k points are labeled as  $\Gamma(000)$ , Y(00.50), S(-0.50.50), and X(-0.500), respectively.

icosahedron, and there are two  $B_{12}$  icosahedra per stacking layer of the (111)- $I_{R,(z)}$  surface. Therefore, the unsaturated dangling bonds of the (111)- $I_{R,(z)}$  surface need additional  $3 \times 2 \times 2/2 = 6$  valence electrons. Two  $B_i$  atoms per surface unit cell, connect six  $B_p$  atoms, as shown in Fig. 1(a), form six bridging  $B_i$ - $B_p$  bonds (these are 2e2cbonds), which perfectly satisfy the electron counting rule (ECR) [27,44], and are responsible for the reconstructioninduced metal-semiconductor transition.

Projected density of states (PDOS) of the topmost atoms is plotted in Fig. 3. The (111)-I surface exhibits metallic character, see Fig. 3(a), predominantly due to the out-ofplane states ( $p_z$  orbitals), arising from the unsaturated  $B_p$ atoms, and located at the bottom of the conduction band. In comparison, strongly hybridized bonding states present in the vicinity of the Fermi level in Fig. 3(b), mainly derive from the  $B_p$ :  $p_z$  and  $B_i$ :  $p_{xy}$  orbitals, and are completely filled. The (111)- $I_{R,(z)}$  surface is therefore a semiconductor. PDOS [Figs. 3(c) and 3(d)] clearly shows that the out-ofplane  $p_z$  states and in-plane  $p_{xy}$  states near the valence band edge are predominantly from the  $B_p$  and  $B_i$  atoms, respectively. Because the  $B_i$  atoms are located above empty space, there is no  $p_z$  state for the  $B_i$  below the Fermi level in Fig. 3(d). Moreover, the distance between the  $B_i$  and  $B_e$  atoms is clearly non-bonding, 2.926 Å. All of these facts indicate that there is no interaction or bonding between the  $B_i$  and  $B_e$  atoms, which further confirms the reliability of the surface bonding configuration and the ECR applied for the (111)- $I_{R,(z)}$  surface.

Symmetry breaking of  $B_{12}$  icosahedra results in charge asymmetry on some B-B bonds, and in fact a small degree of ionicity of B-B bonds in the  $B_{12}$  icosahedron was predicted in  $\alpha$ -boron [3], while a much greater degree of ionicity was found in the high-pressure partially ionic  $\gamma$ -boron [4]. It is interesting to study charge transfer or ionicity for the (111)- $I_{R,(z)}$  surface. Chemical effects of the  $B_i$  adsorption can be seen in Fig. 4, which shows charge density difference [45]. There is a notable charge transfer from  $B_i$  to the neighboring  $B_{12}$  icosahedra through  $B_p$ atoms. Bader charges for  $B_i$  are +0.17e, and +0.04e for  $B_p$ [46]. Strikingly, the charge of  $B_p$  is close to the value in the bulk ( $\sim +0.05e$ ) [4,23], and the significant charge transfer between the  $B_i$  atoms and  $B_{12}$  icosahedra indicates that the bridging  $B_i$ - $B_p$  bonds are unique polar covalent bonds, in constrast with the intericosahedral purely covalent  $B_p$ - $B_p$ 



FIG. 3 (color online). [(a) and (b)] PDOS of the (111)-I and (111)- $I_{R,(z)}$  structures. [(c) and (d)] PDOS of the  $B_p$  and  $B_i$  atoms in (111)- $I_{R,(z)}$  structure.



FIG. 4 (color online). Charge density difference isosurface [45] showing bonding between the adsorbed boron atoms and the substrate. (a) Top view (for clarity, only the adsorbed atoms are shown), (b) side view (showing the substrate).

bonds [3]. Consistent with significant charge transfer, the  $B_i$ - $B_p$  bonds (1.793 Å) are much weaker and longer than the  $B_p$ - $B_p$  bonds (1.673 Å). In Fig. 4(b), each  $B_{12}$ icosahedron comprises four atomic layers (labeled from 1 to 4); the charge transfer for these four layers should be in the "+ – –+" order with the values of +0.56, –0.46, –0.46, and +0.36*e*, compared with the corresponding values of +0.20, –0.22, –0.22, and +0.24*e* in  $\alpha$ -boron. It is clear from these numbers that the surface region as a whole is charge-neutral. The charge transfer of the (111)- $I_{R,(z)}$  surface is rebalanced within the top four atomic layers (including the  $B_i$  atoms and  $B_{12}$ icosahedra), which plays an important role in stabilizing the surface.

In conclusion, the most stable reconstruction of the  $\alpha$ -boron (111) surface has been predicted using an *ab initio* evolutionary structure search. Our results show the (111)- $I_{R,(z)}$  reconstruction is lower in energy than the earlier reported structures [26], and confirm that the classical ECR governs the reconstructions, and is

satisfied through adsorption of extra boron atoms that occupy non-icosahedral sites and form 2e2c bonds with substrate, leading to the metal-semiconductor transition. These newly formed bonds involve a significant charge transfer and can be characterized as polar covalent bonds. Charge redistribution between the  $B_i$  atoms and the  $B_{12}$  icosahedra contributes to the stabilization of this reconstruction.

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