

# Evolutionary Crystal Structure Prediction and USPEX

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# Structure is the basis for understanding materials and their properties



## The Nobel Prize in Physics 1914

"for his discovery of the diffraction of X-rays by crystals"



Max von Laue



## The Nobel Prize in Physics 1915

"for their services in the analysis of crystal structure by means of X-rays"



Sir William Henry Bragg



William Lawrence Bragg



## The Nobel Prize in Chemistry 1985

"for their outstanding achievements in the development of direct methods for the determination of crystal structures"

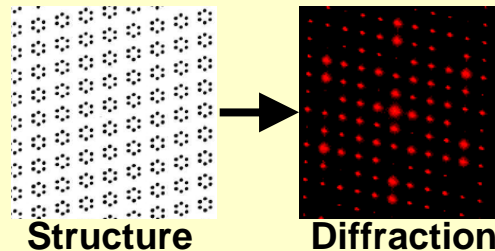


Herbert A. Hauptman

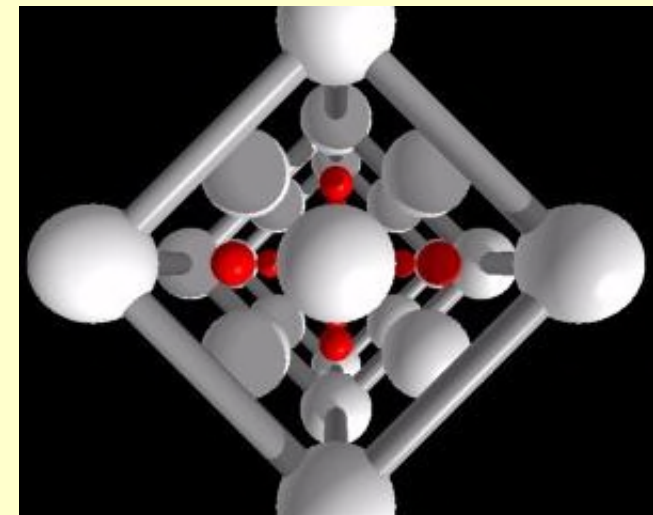


Jerome Karle

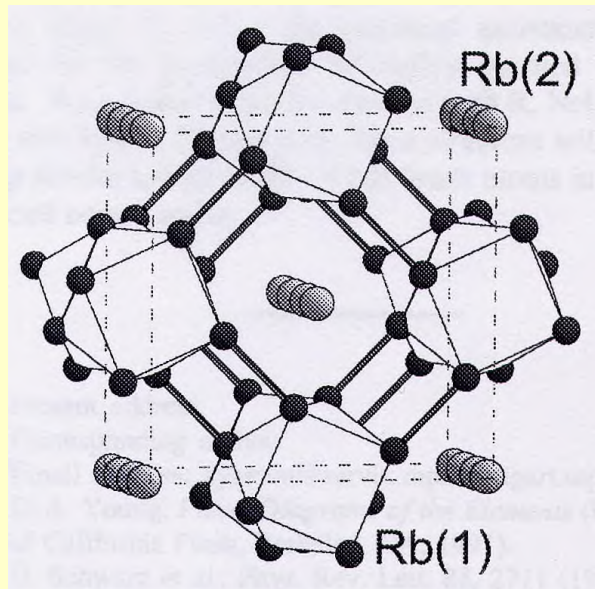
(from <http://nobelprize.org>)



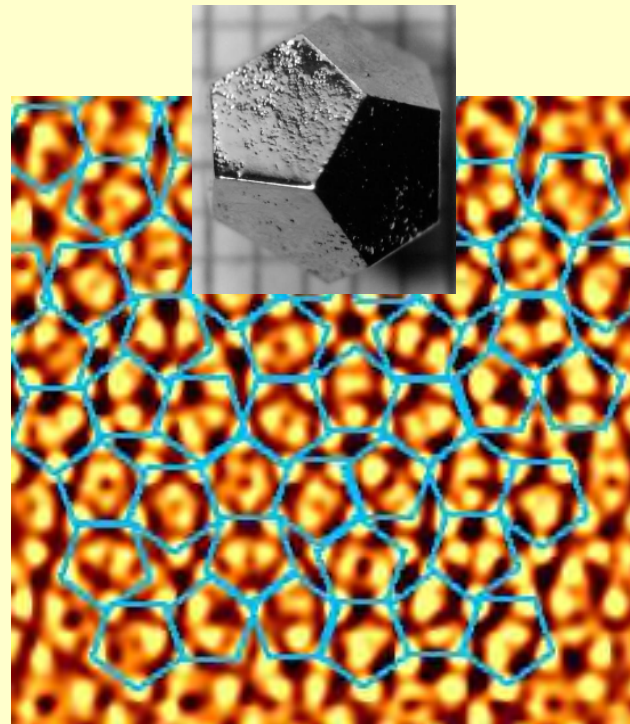
**Zincblende ZnS.**  
One of the first structures solved by Braggs in 1913.



# With time, incredibly complex structures were discovered



**Host-guest elements**  
(Rb-IV phase, U.Schwarz'99)



**Quasicrystals**

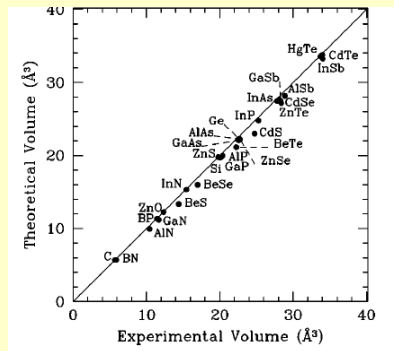
New state of matter discovered in lab  
in 1984. In nature found only in 2009!



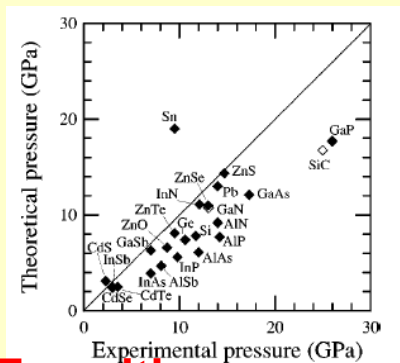
**Proteins**

# When the structure is known, many properties can be computed reliably

- State of art: DFT.
- Molar volumes: within 1-3% (LDA,GGA).
- Transition pressures: 10% underestimated (LDA),  $\pm 5$  GPa (GGA).
- Band gaps:  $\sim 30\%$  underestimated (LDA, GGA),  $\pm 10\%$  (GW).
- Unsatisfactory – for van der Waals crystals, systems with localised d- and f-electrons.



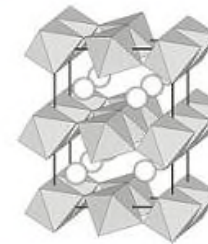
**Volumes**  
LDA: Mujica'03



**Transition pressures**  
LDA: Mujica'03

International journal for structural, physical,  
and chemical aspects of crystalline materials

ZEITSCHRIFT FÜR  
KRISTALLOGRAPHIE



Volume 220 5-6/2005

Computational Crystallography

Edited by A. R. Oganov

Oldenbourg



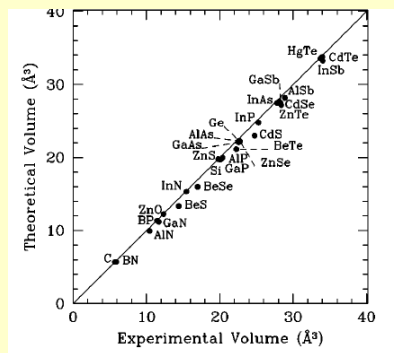


## Experimentalists also like DFT

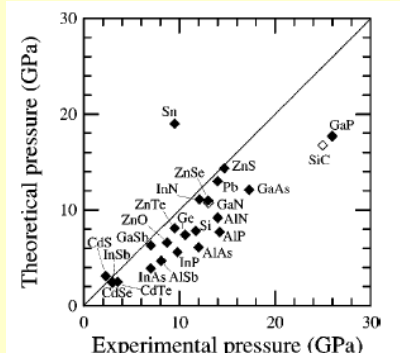


## When the structure is known, many properties can be computed reliably

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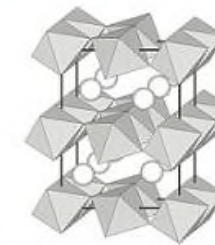
**Volumes**  
LDA: Mujica'03



**Transition pressures**  
LDA: Mujica'03

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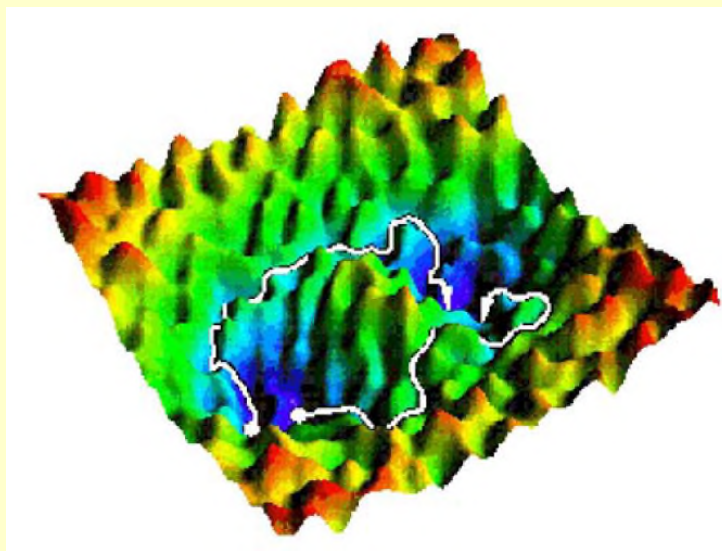
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# (Free) energy landscape: key to thermodynamics and kinetics



$$\begin{aligned} E \\ H=E+PV \\ F=E-TS \\ G=E+PV-TS \end{aligned}$$

**Transition paths, rate constants etc.**

**Global minimum & low-energy local minima**

J. Phys.: Condens. Matter 12 (2000) A147–A152. Printed in the UK

## Transition path sampling: throwing ropes over mountains in the dark

Peter G Bolhuis<sup>†</sup>, Christoph Dellago<sup>‡</sup>, Phillip L Geissler<sup>‡</sup> and David Chandler<sup>‡</sup>

<sup>†</sup> Department of Chemistry, Lensfield Rd, Cambridge CB2 1EW, UK

<sup>‡</sup> Department of Chemistry, University of California, Berkeley, CA 94720, USA

Received 5 October 1999

THE JOURNAL OF CHEMICAL PHYSICS 124, 244704 (2006)

## Crystal structure prediction using *ab initio* evolutionary techniques: Principles and applications

Artem R. Oganov<sup>a)</sup> and Colin W. Glass

Laboratory of Crystallography, Department of Materials, ETH Zurich, HCI G 515, Wolfgang-Pauli-Strasse 10, CH-8093 Zurich, Switzerland

(Received 20 February 2006; accepted 12 May 2006; published online 28 June 2006)



*Acc. Chem. Res.* **1994**, *27*, 309–314

## Are Crystal Structures Predictable?

ANGELO GAVEZZOTTI\*



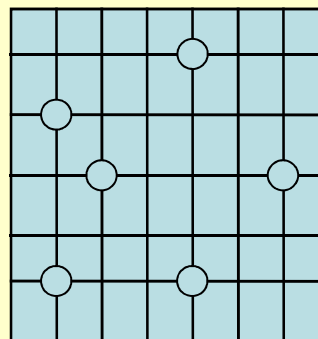
“No”: by just writing down this concise statement, in what would be the first one-word paper in the chemical literature, one could safely summarize the present state of affairs

J. Maddox  
(*Nature*, 1988)

Need to find GLOBAL energy minimum.

Trying all structures is impossible:

$$C = \frac{1}{(V/\delta^3)} \frac{(V/\delta^3)!}{[(V/\delta^3) - N]!N!}$$



N <sub>atoms</sub>	Variants	CPU time
1	1	1 sec.
10	10 <sup>11</sup>	10 <sup>3</sup> yrs.
20	10 <sup>25</sup>	10 <sup>17</sup> yrs.
30	10 <sup>39</sup>	10 <sup>31</sup> yrs.

### RESEARCH NEWS

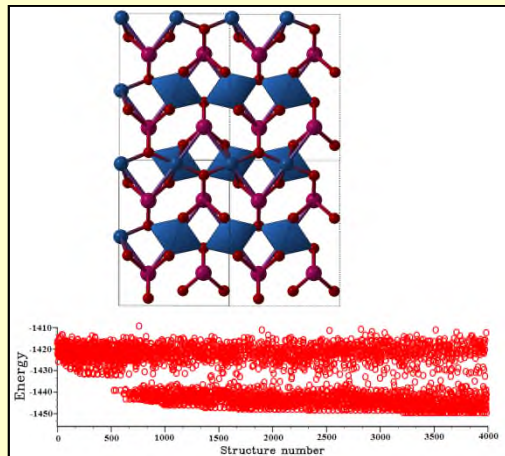
Crystal structure prediction – evolutionary or revolutionary crystallography?

S. L. Chaplot and K. R. Rao

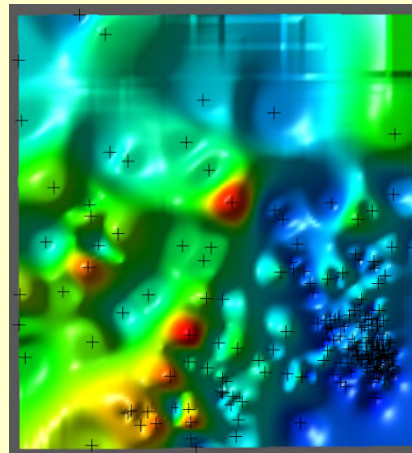
CURRENT SCIENCE, VOL. 91, NO. 11, 10 DECEMBER 2006

Overview of USPEX  
(Oganov & Glass,  
*J.Chem.Phys.* 2006)

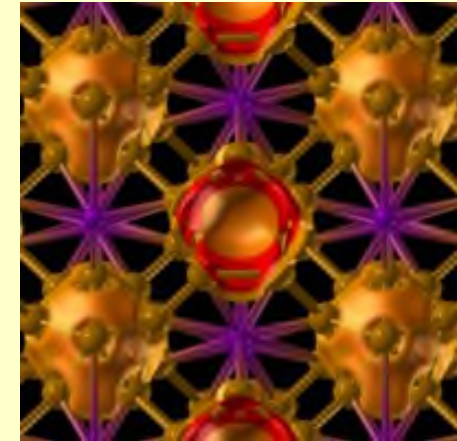
# New developments in crystal structure prediction extend the range of problems that can be solved



**1. Predicting crystal structures by evolution**

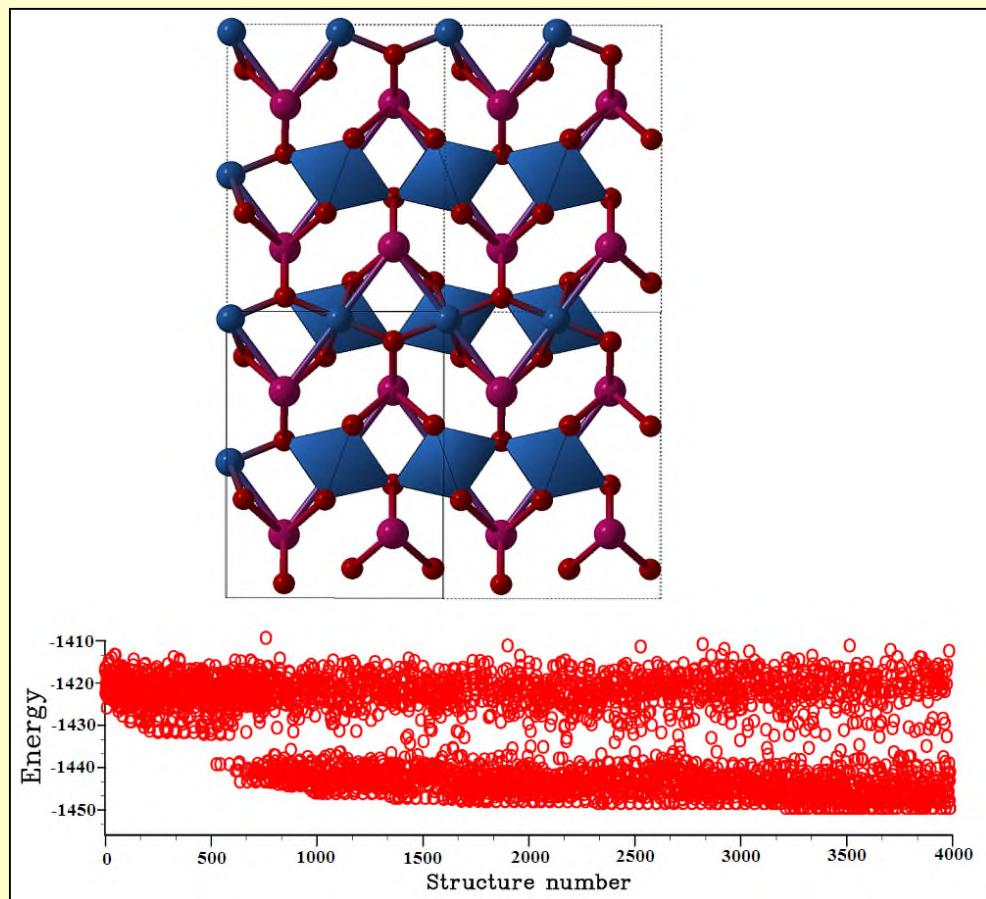


**2. Why does it work?**



**3. Predicting new materials**

# 1. Predicting structures by evolution



Oganov A.R., Lyakhov A.O., Valle M. (2011).  
How evolutionary crystal structure prediction works - and why.  
*Acc. Chem. Res.* 44, 227-237.

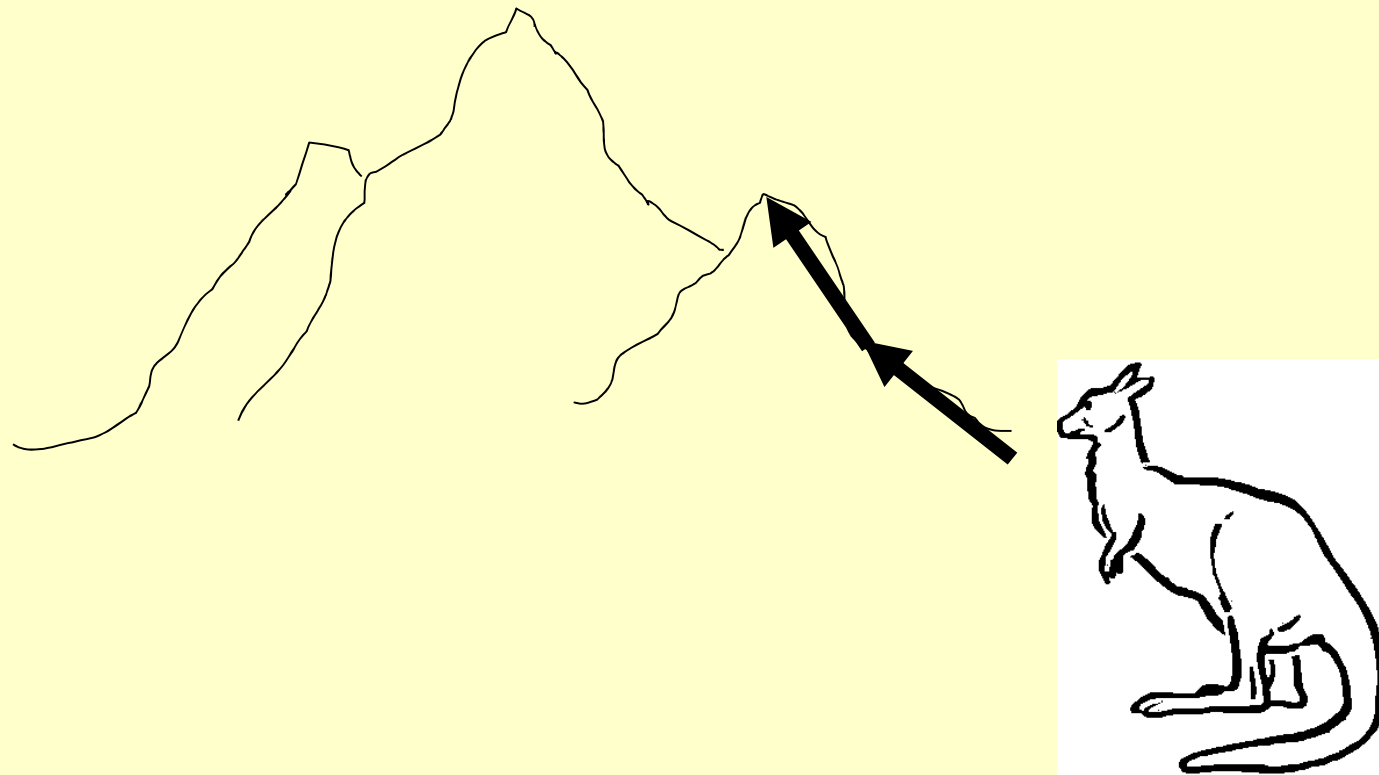


# Crystal structure prediction methods

1. Random sampling (Freeman & Catlow, 1992)
2. Simulated annealing (Pannetier, 1990)
3. Molecular dynamics and metadynamics (Martonak, 2003)
4. Data mining (Curtarolo, 2003)
5. Minima hopping (Goedecker, 2004)
6. Evolutionary algorithms / PSO

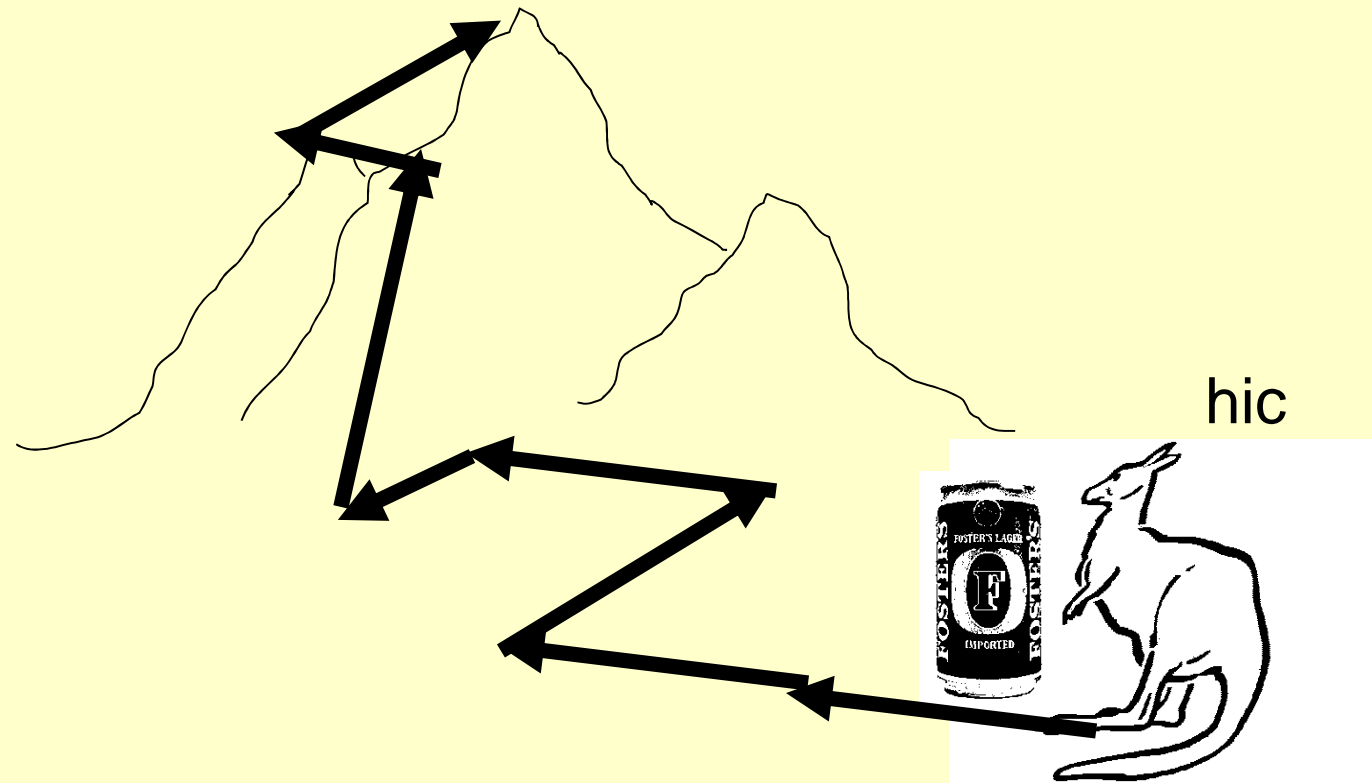
**All of them rely on local optimization methods!**

## Global optimisation methods: Kangaroo's climb to Mt. Everest



**Random sampling** is like dropping a kangaroo somewhere on the surface of the earth, telling it to only hop uphill and hoping it will get to the top of mount Everest.

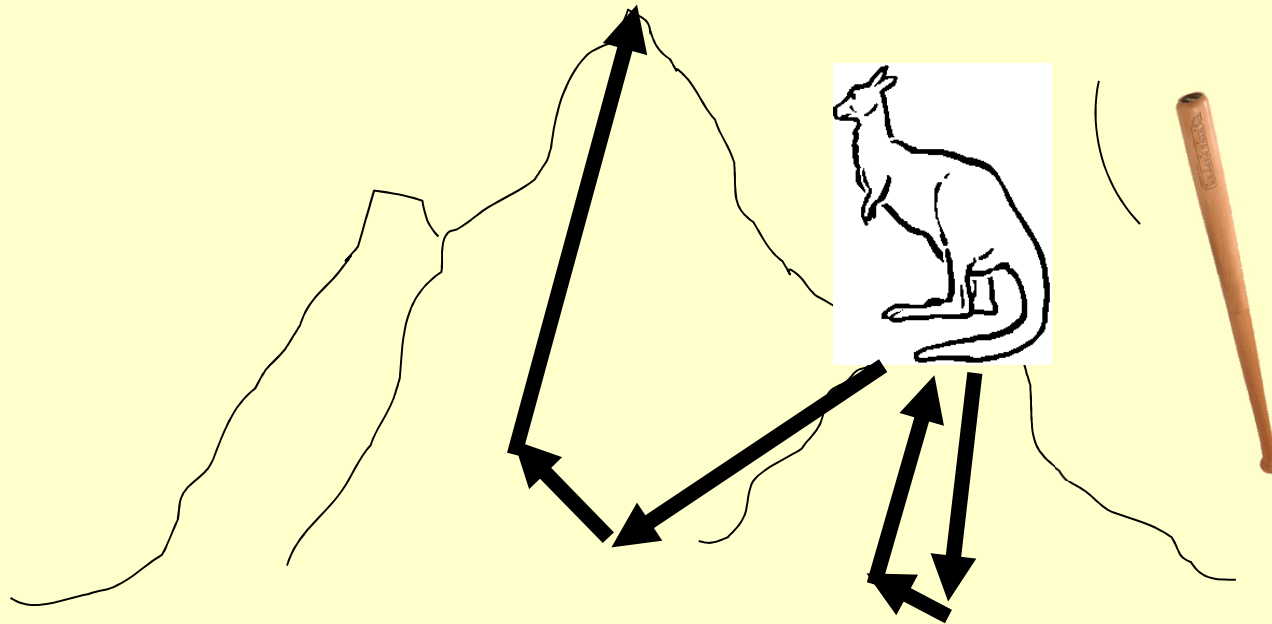
## Global optimisation methods: Kangaroo's climb to Mt. Everest



**Simulated Annealing** is like doing the same but getting the kangaroo very very drunk first.

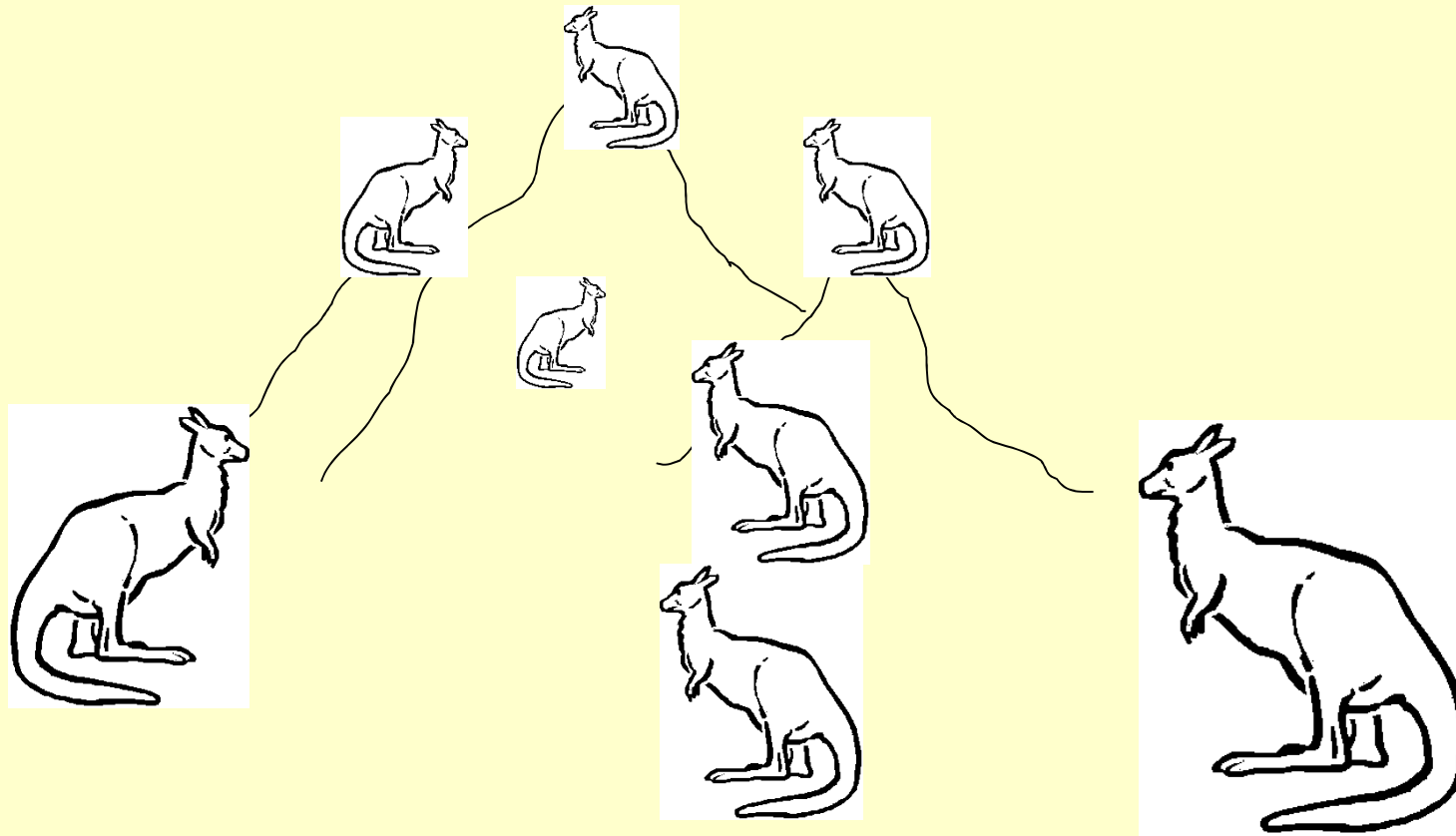


## Global optimisation methods: Kangaroo's climb to Mt. Everest



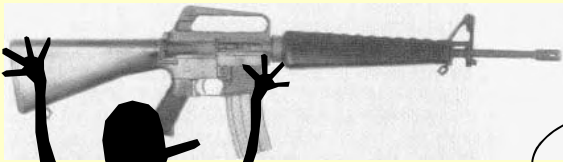
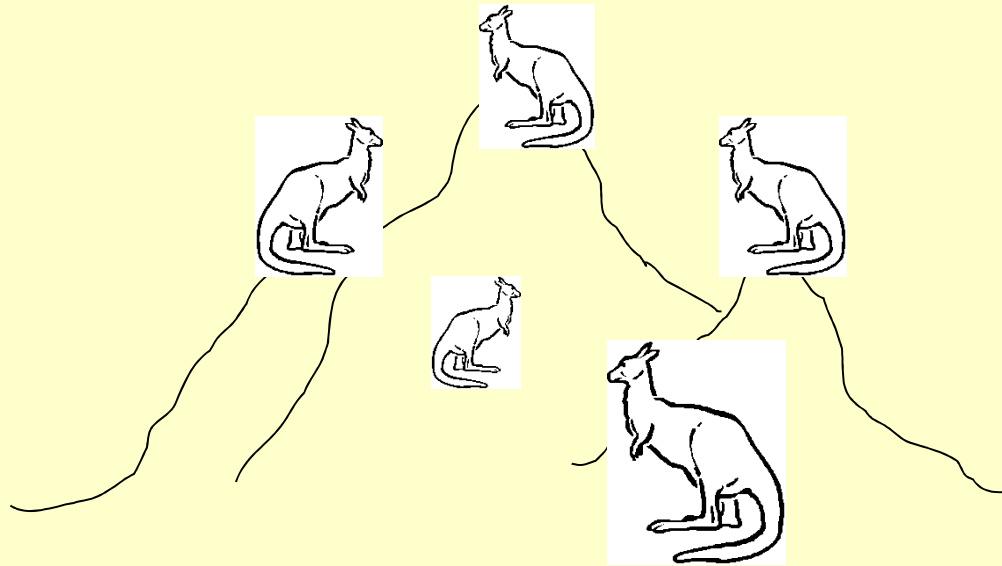
**Minima Hopping** is like knocking the kangaroo off the smaller hill with a bat harder and harder till it is close enough to another hill to climb it.

## Global optimisation methods: Kangaroo's climb to Mt. Everest



**Evolutionary Algorithms** are like taking a whole plane load of kangaroos and letting them reproduce freely (not pictured).....

# Global optimisation methods: Kangaroo's climb to Mt. Everest



Aaaargh  
!

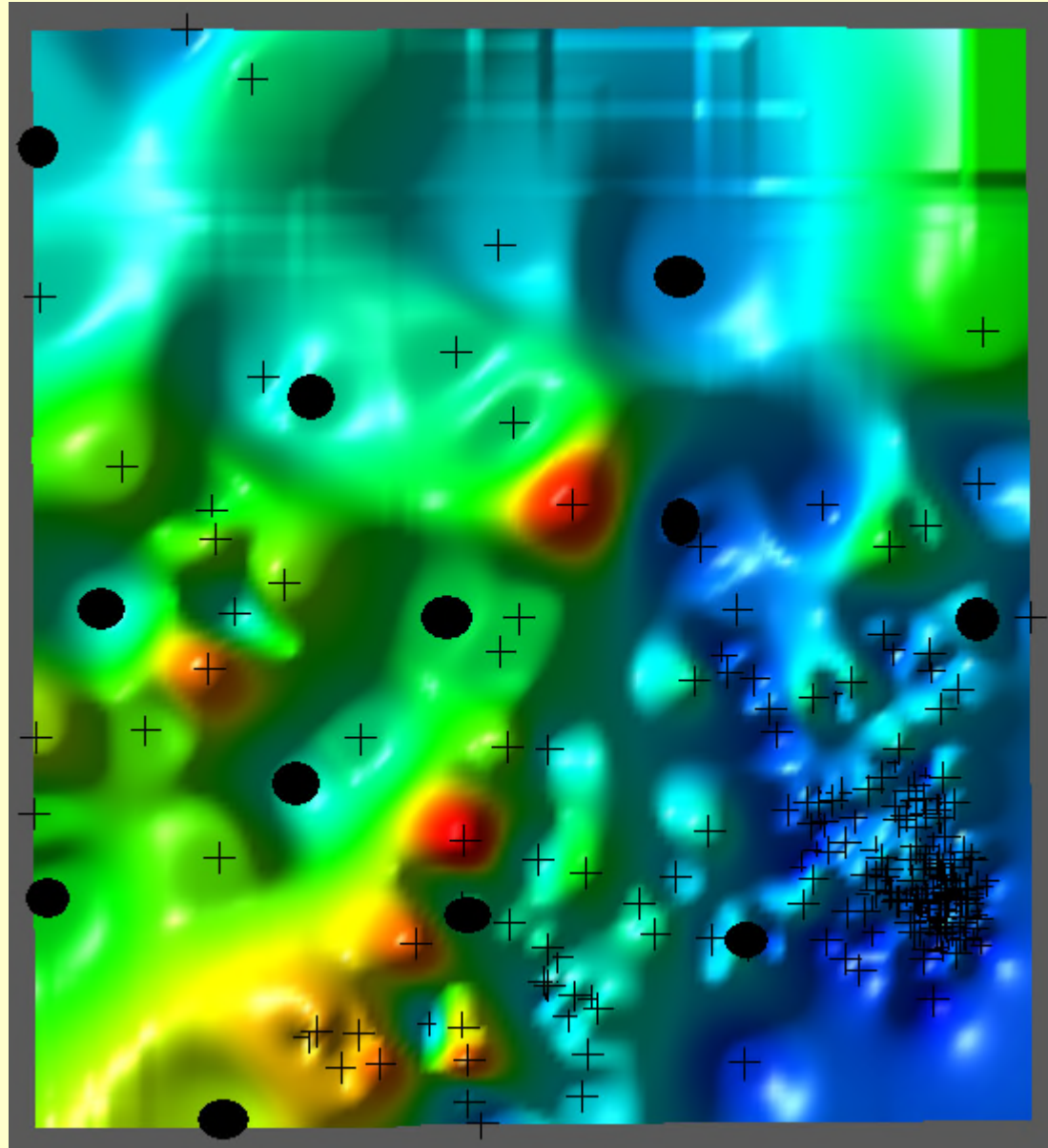


Ouch

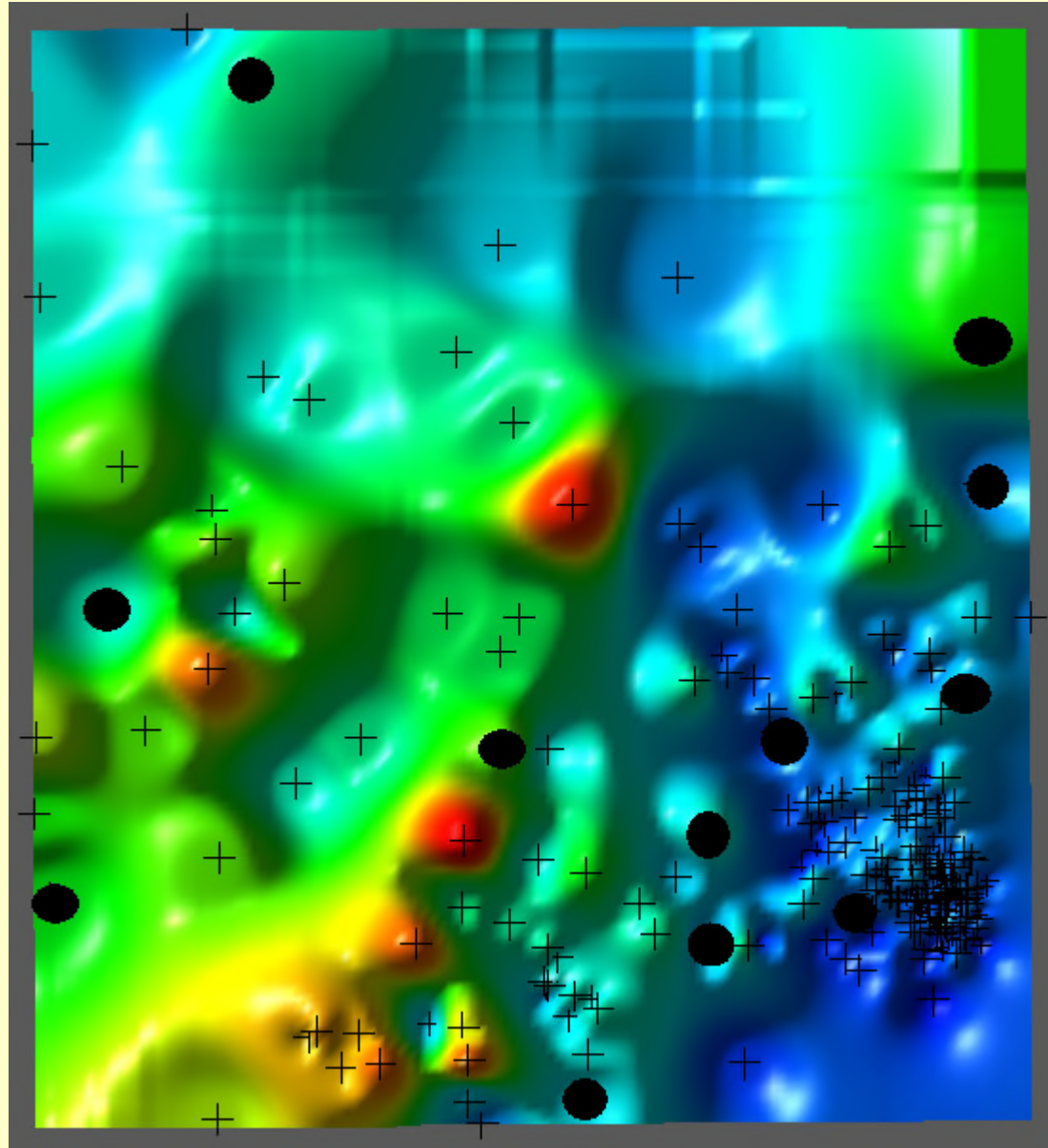
....and regularly shooting the ones at lower altitudes.



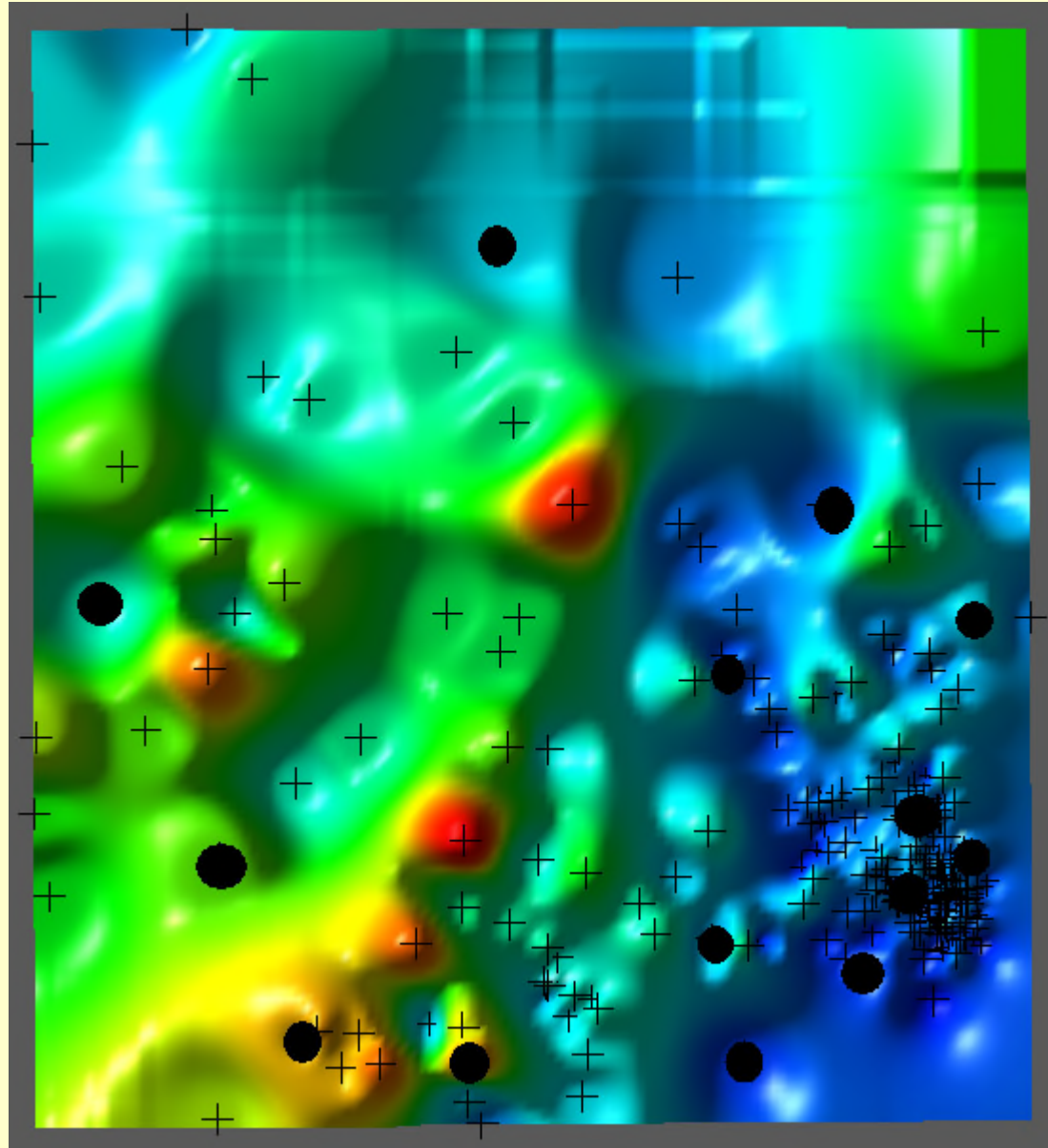
# Evolutionary simulations learn & explore the most promising regions of search space



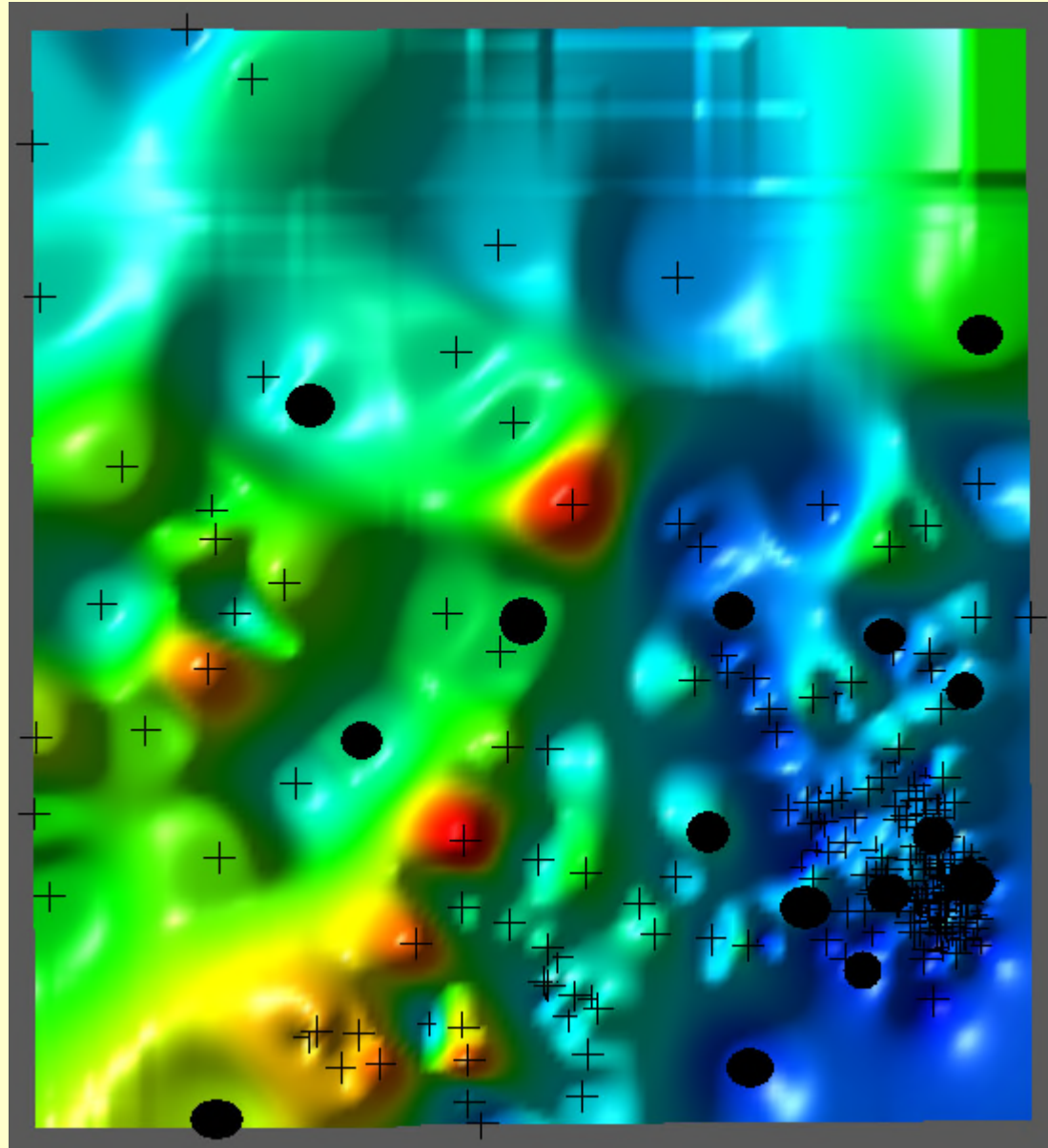
# Evolutionary simulations learn & explore the most promising regions of search space



# Evolutionary simulations learn & explore the most promising regions of search space

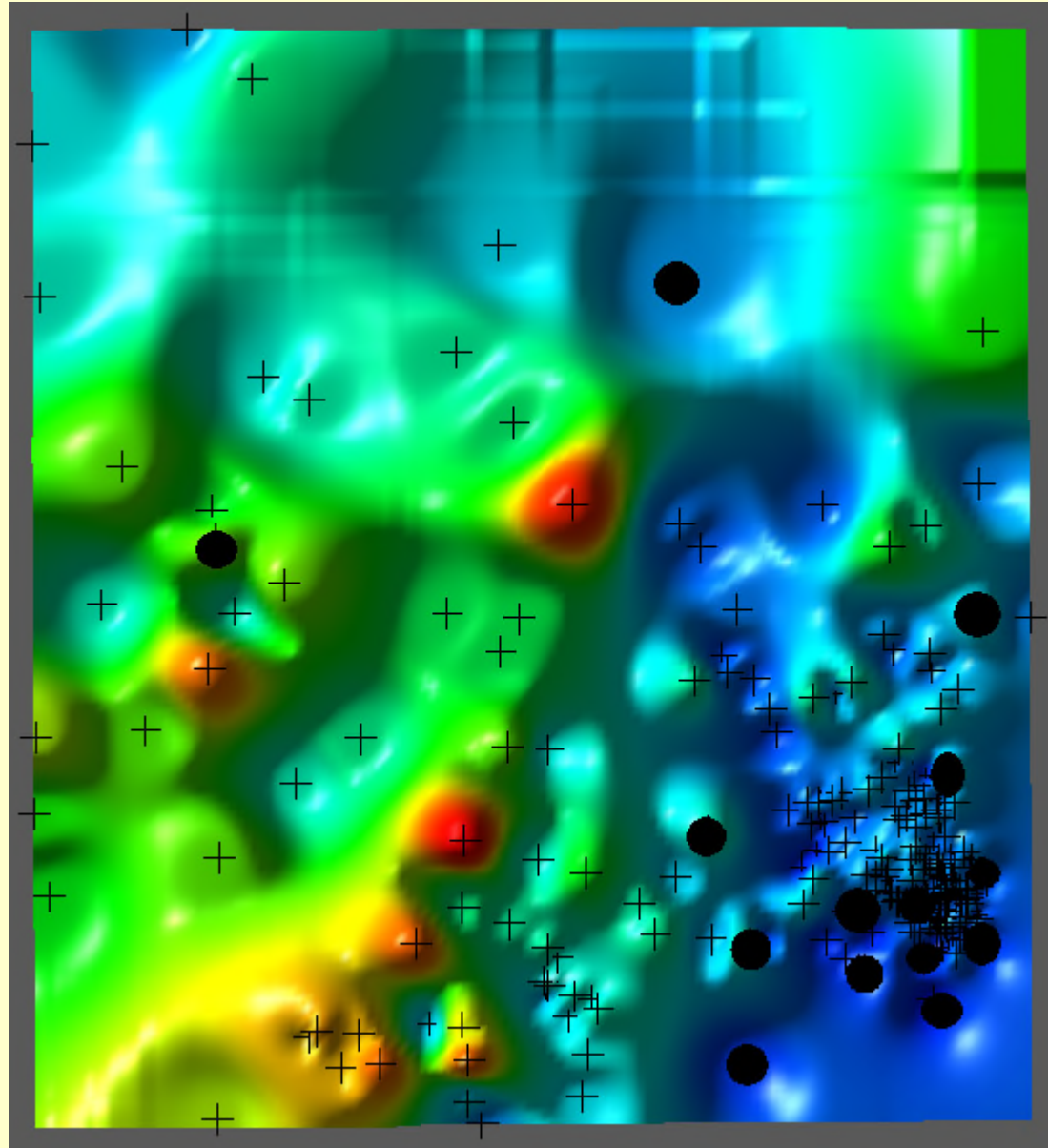


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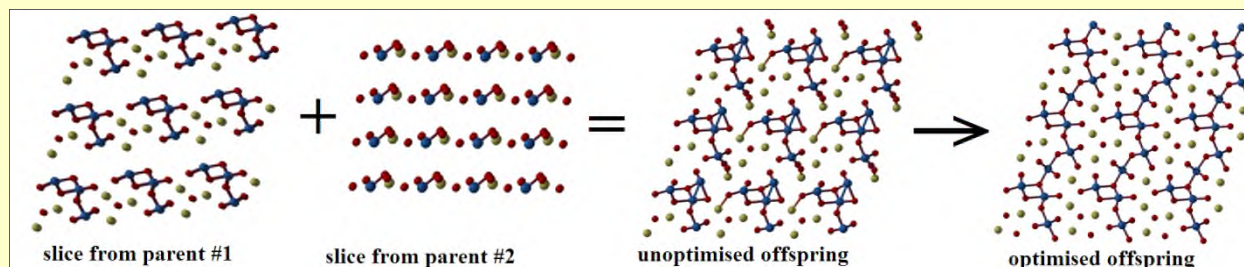
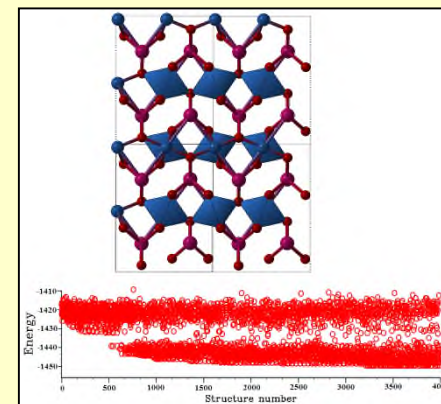




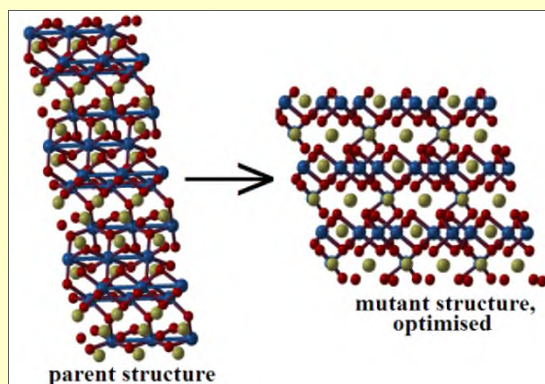
# USPEX

## (Universal Structure Predictor: Evolutionary Xtallography)

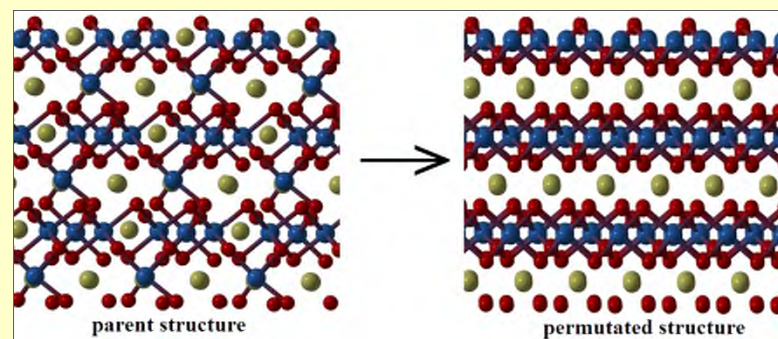
- (Random) initial population
- Evaluate structures by relaxed (free) energy
- Select lowest-energy structures as parents for new generation
- Standard variation operators:



(1) Heredity (crossover)

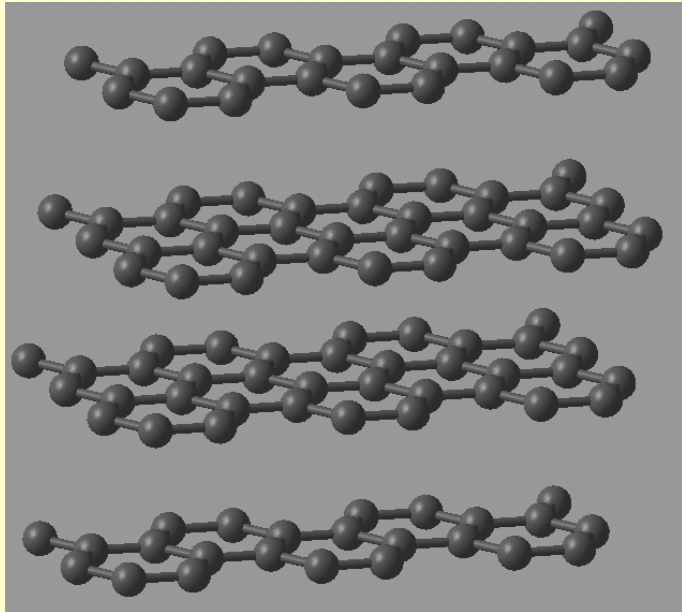


(2) Lattice mutation

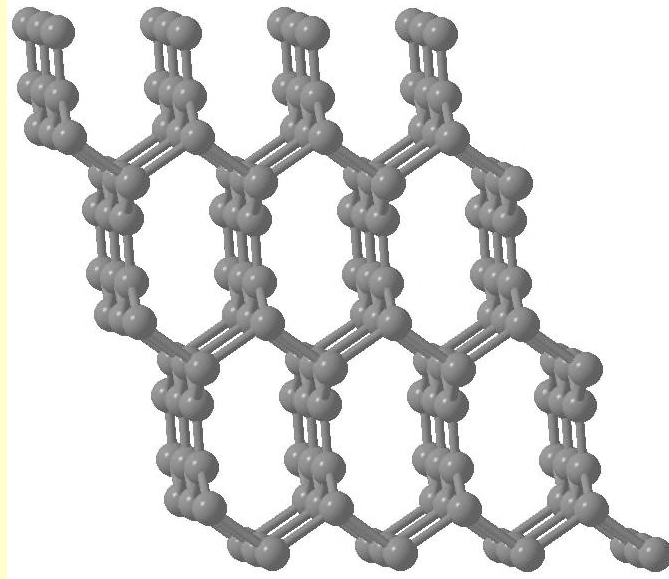


(3) Permutation

**Test:** „Who would guess that graphite is the stable allotrope of carbon at ordinary pressure?“ (Maddox, 1988)



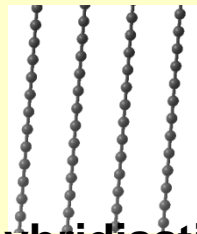
Graphite, correctly predicted to be the stable phase at 1 atm



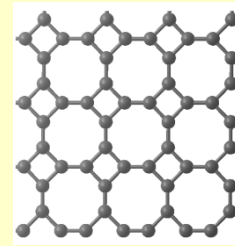
Metastable superhard  $sp^2$ -forms with 3D-topology. First proposed by R.Hoffmann (1983)



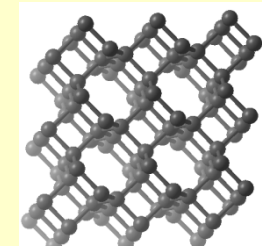
Low-energy structures reveal chemistry



$sp$ -hybridisation (carbyne)



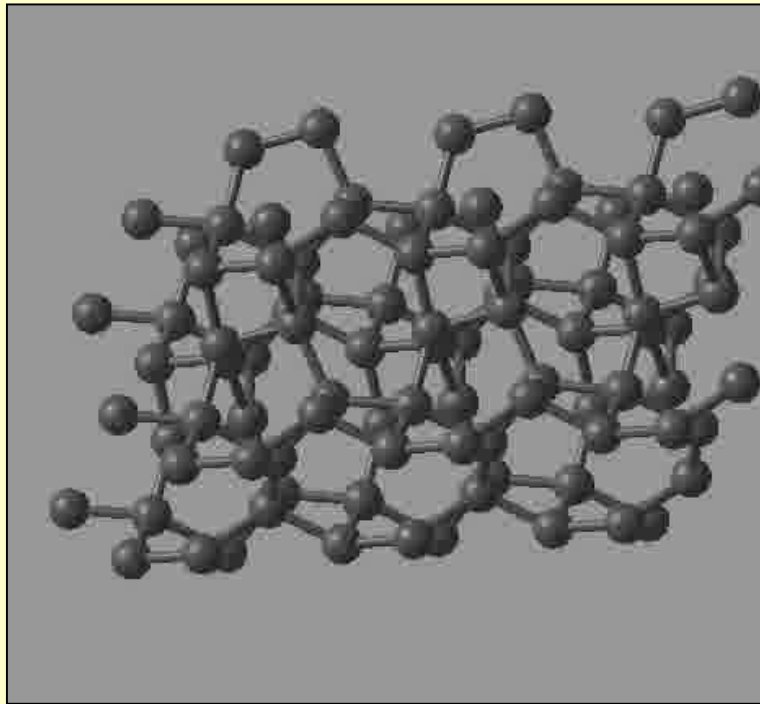
$sp^2$ -hybridisation



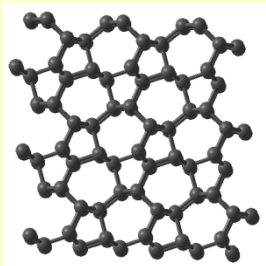
$sp^3$ -hybridisation

[ARO & Glass, J.Chem.Phys. (2006)]

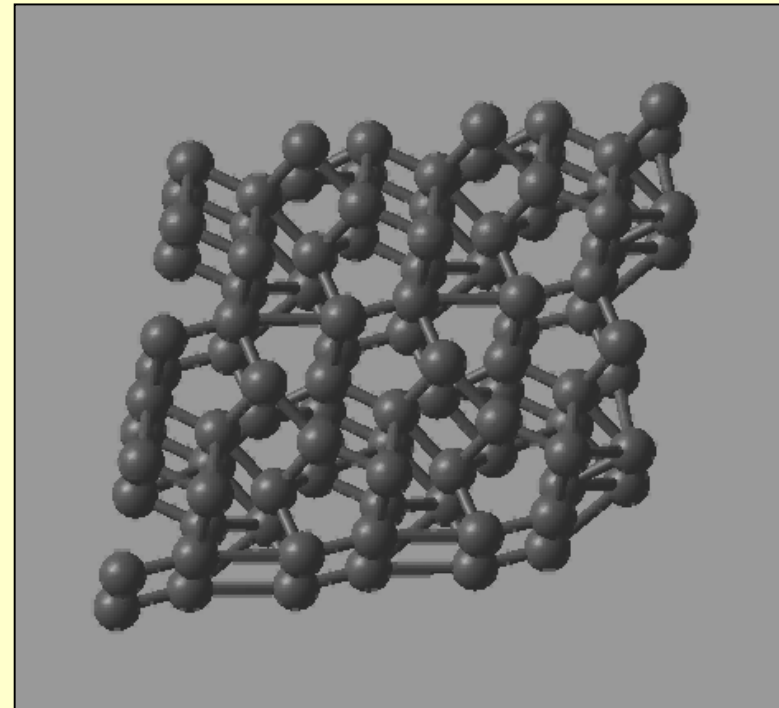
**Test:** High-pressure phases of carbon are also  
successfully reproduced



**100 GPa:** diamond is stable



+found metastable form  
that matches  
„superhard graphite“ of W.Mao  
(Li, ARO, Ma, et al., PRL 2009)



**2000 GPa:** bc8 phase, potentially  
important in astrophysics

Metastable bc8 form of Si  
is known (Kasper, 1964)

[ARO & Glass, J.Chem.Phys. (2006)]

## Alternative methods:

### Random sampling

(Freeman & Catlow, 1993; Schmidt et al., 1996; van Eijck & Kroon, 2000; Pickard & Needs, 2006)

- No „learning“. Works well only for small problems (<30 degrees of freedom – e.g. 10 atoms).

### Simulated annealing (Pannetier 1990; Schön & Jansen 1996)

- Random walk. Ever decreasing probability to accept step to worse solution
- Difficult to control parameters.
- No „learning“ - only current position as source of information!

### Metadynamics (Martonak, Laio, Parrinello 2003)

- Taboo search with reduced dimensionality.

$$G^t(\mathbf{h}) = G(\mathbf{h}) + \sum_{t' < t} W e^{-\frac{|\mathbf{h} - \mathbf{h}^{t'}|^2}{2\delta h^2}}$$

$$\mathbf{h}^{t+1} = \mathbf{h}^t + \delta h \frac{\phi^t}{|\phi^t|}$$

### Minima hopping (Gödecker 2004)

- Keep history of visited minima. Escape minima with MD, using feedback to control temperature

### Genetic and evolutionary algorithms

- Bush (1995), Woodley (1999) – works only for small systems, inefficient.
- Deaven & Ho (1995) – developed only for clusters. Efficient.
- ARO& Glass (2006), Abraham (2006), Fadda (2010), Wang (2010), Lunie (2011)

## Blind test (2010): USPEX is superior to random sampling and simulated annealing

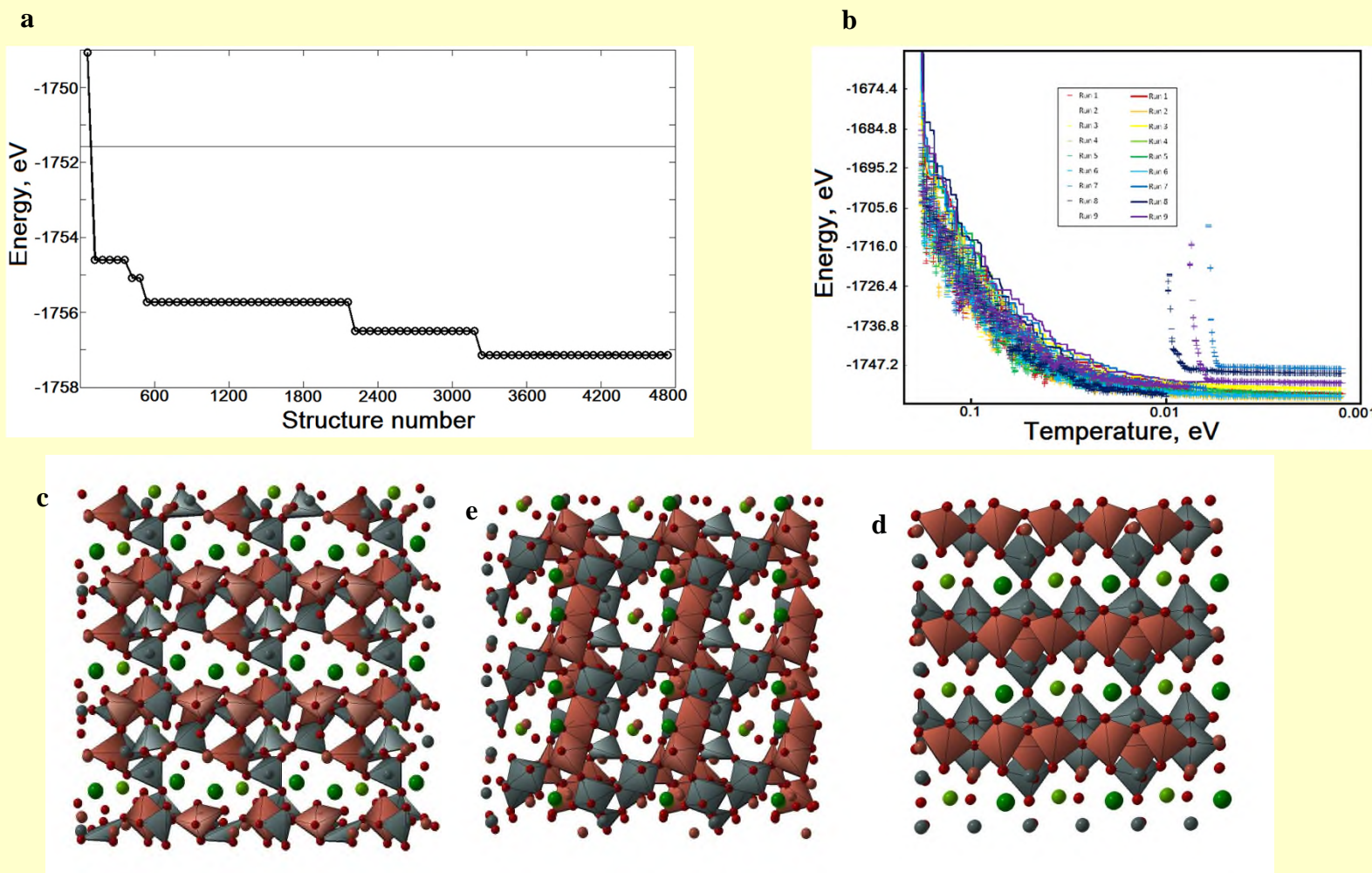
- Blind test (ARO, Schon, Hennig, 2010) – on extremely difficult cases:

	Random sampling	Sim. annealing	USPEX
Test #1, BaMgAl <sub>4</sub> Si <sub>4</sub> O <sub>16</sub> with fixed cubic cell (with forcefields)			
Number of runs (runs producing lowest E)	1 (1)	10 (1)	2 (2)
Minimum energy, eV	-876.94	<b>-877.99</b>	-877.71
# of structure relaxations before ground state	14794	7330	1465
Test #2, Ba <sub>2</sub> Mg <sub>2</sub> Al <sub>8</sub> Si <sub>8</sub> O <sub>32</sub> with fixed cubic cell (with forcefields)			
Number of runs (runs producing lowest E)	1 (1)	9 (1)	2 (1)
Minimum energy, eV	-1751.57	-1756.03	<b>-1757.14</b>
# of structure relaxations before ground state	14102	2435	3210
Test #3, Mg <sub>10</sub> Al <sub>4</sub> Ge <sub>2</sub> Si <sub>8</sub> O <sub>36</sub> with variable cell (with forcefields)			
Number of runs (runs producing lowest E)	1 (1)	9 (1)	1 (1)
Minimum energy, eV	-1943.46	-1949.10	<b>-1950.53</b>
# of structure relaxations before ground state	13029	685	4610
Test #4, Mg <sub>13</sub> Al <sub>8</sub> P <sub>3</sub> with variable cell ( <i>ab initio</i> )			
Number of runs (runs producing lowest E)	1 (1)	-	1 (1)
Minimum energy, eV	-68.82	-	<b>-70.37</b>
# of structure relaxations before ground state	978	-	4071

Random sampling failed to give lowest-enthalpy structures for 2 phases (out of 3 predicted) of SiH<sub>4</sub> (Pickard, PRL 2006), 1 for Nitrogen (Pickard PRL 2009), 1 for SnH<sub>4</sub> (Pickard, 2010)



# Benchmarking the power of the method



Test #2 ( $\text{Ba}_2\text{Mg}_2\text{Al}_8\text{Si}_8\text{O}_{32}$ , with fixed cell): (a) Variation of the lowest energy during the evolutionary USPEX run, (b) Summary of simulated annealing runs, (c-e) Lowest-energy structures obtained by random sampling, simulated annealing and USPEX, respectively. Thin horizontal line in (a) shows the lowest energy found in 14102 random sampling attempts.

# Crystal structure prediction methods

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4. Data mining (Curtarolo, 2003)
5. Minima hopping (Goedecker, 2004)
6. Evolutionary algorithms / PSO

**Test on TiO<sub>2</sub>: USPEX vs PSO**  
(data from Wang et al., 2012; Lyakhov et al., 2013)

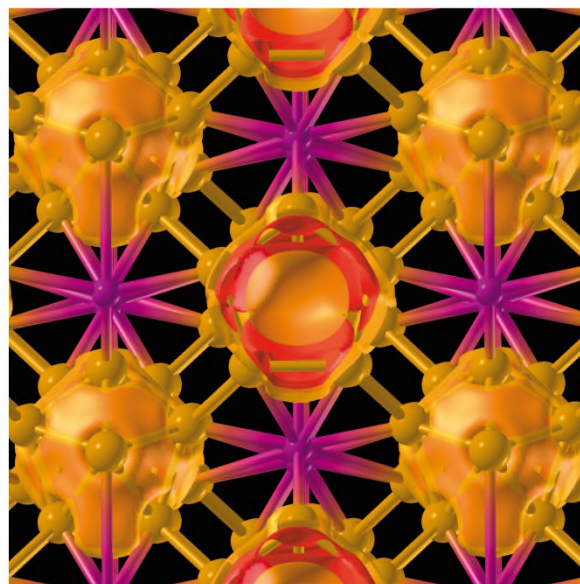
Method	Success rate	<#relaxations>
USPEX, cell splitting	100%	41
USPEX, no symmetry	100%	80
USPEX, with symmetry	100%	77
CALYPSO with symmetry	100%	168 - 400
CALYPSO no symmetry	90%	508

# For more information...

Edited by Artem R. Oganov

 WILEY-VCH

## Modern Methods of Crystal Structure Prediction

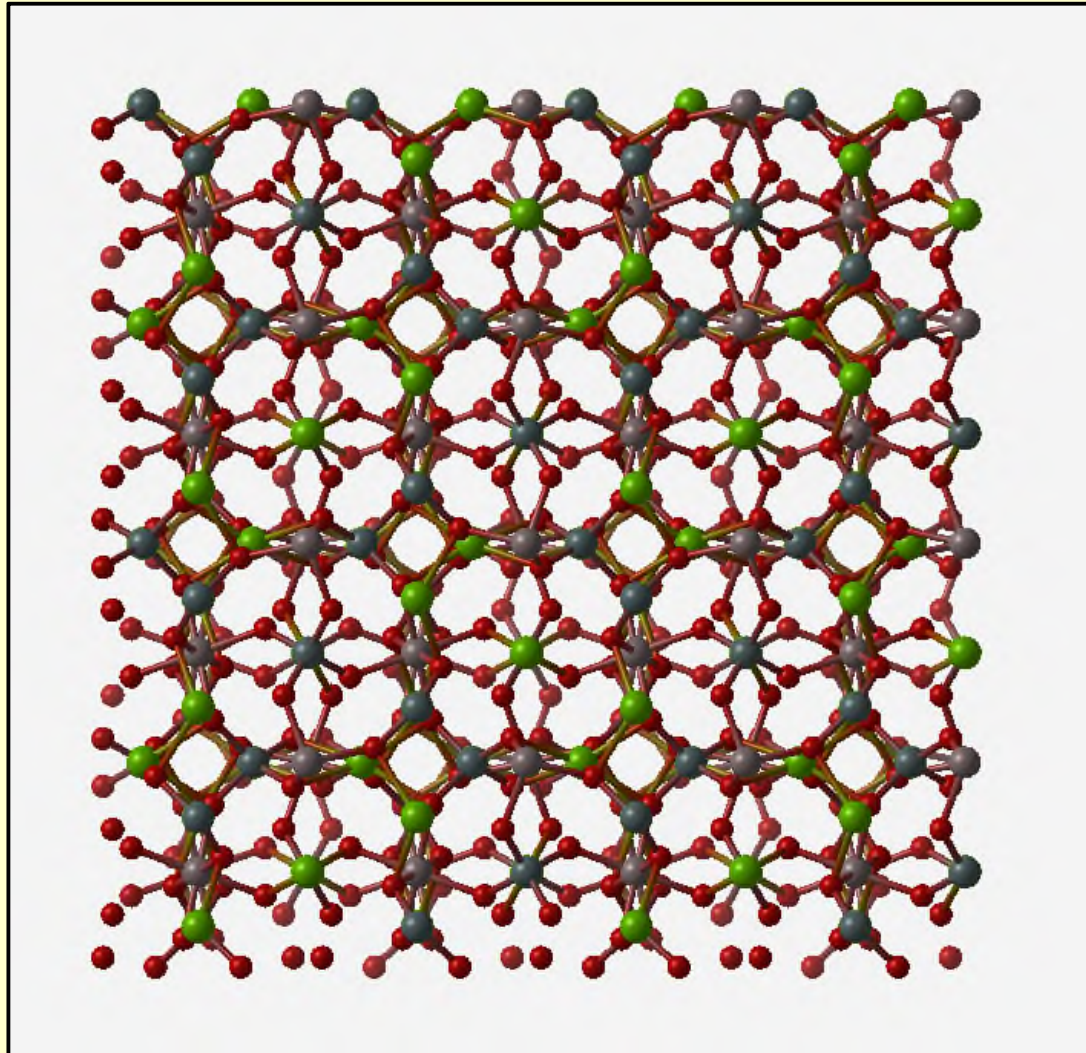


# USPEX

New developments:

- improved efficiency
- nanoparticles
- surfaces and interfaces
- properties optimization
- molecular crystals
- variable composition
- TPS and vcNEB modules
- evolutionary metadynamics

## Power of the new method



Garnet, 160 atoms/cell,  $\text{Mg}_{24}\text{Al}_{16}\text{Si}_{24}\text{O}_{96}$  :  
100% success rate;  $\langle N \rangle = 294$ ; 35 calculations so far

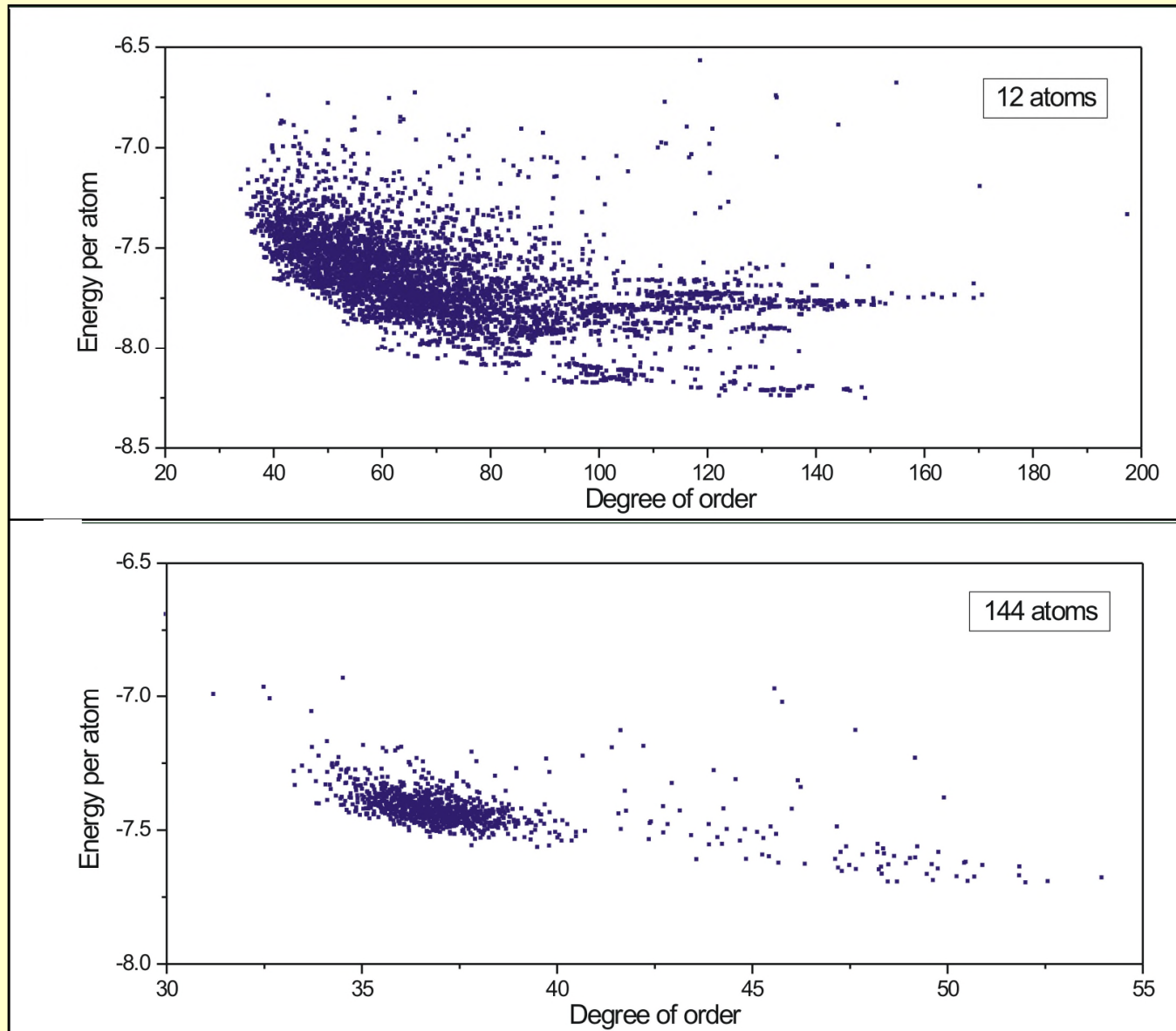


# Key improvements

- **Symmetrical initialization**
- **Aging technique**
- **Smart mutation**

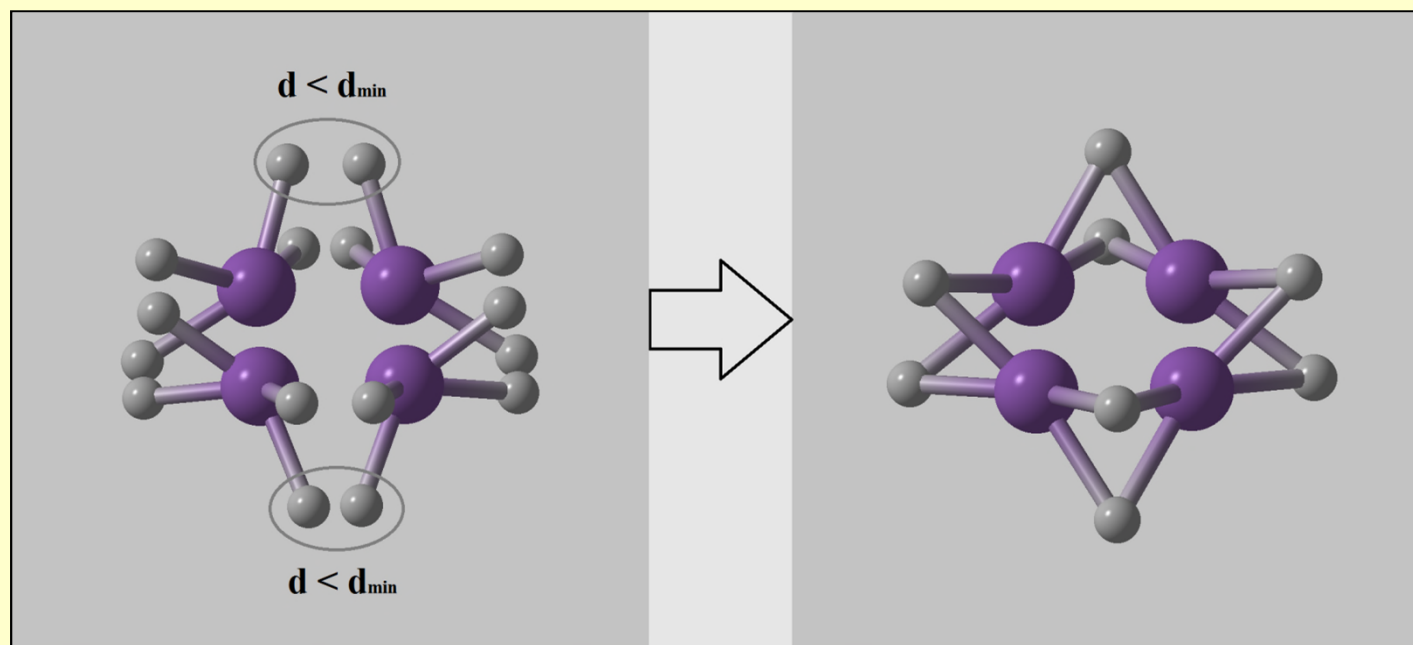
**(plus plethora of methodological developments published in 2010-2011)**

# Symmetrical initialization



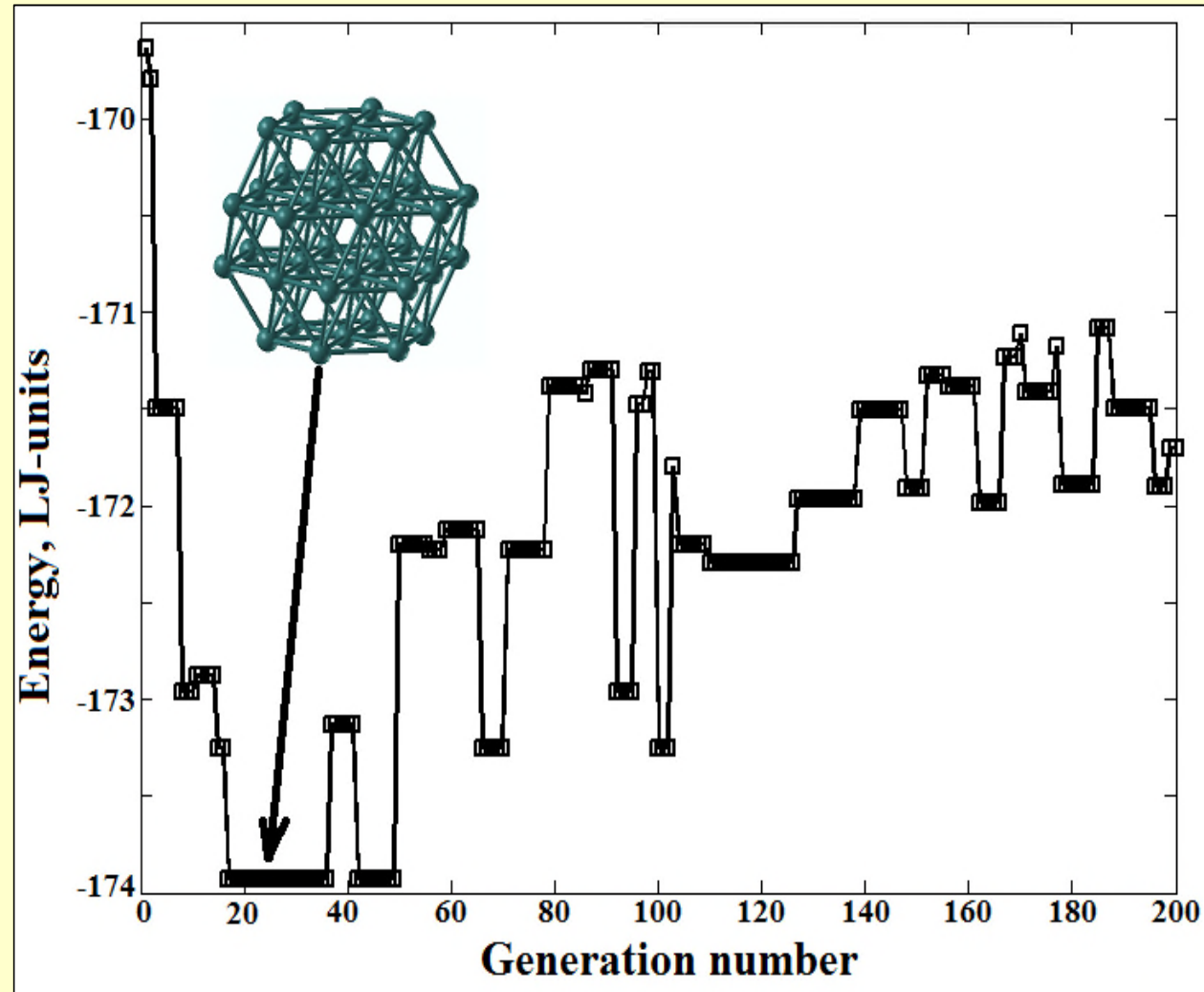
# Symmetrical initialization

**Crystals: 230 space groups**  
**Nanoparticles: point groups**



**Enables moderately efficient random sampling  
as one of possible USPEX regimes**

## 'Aging' technique (antiseeds)

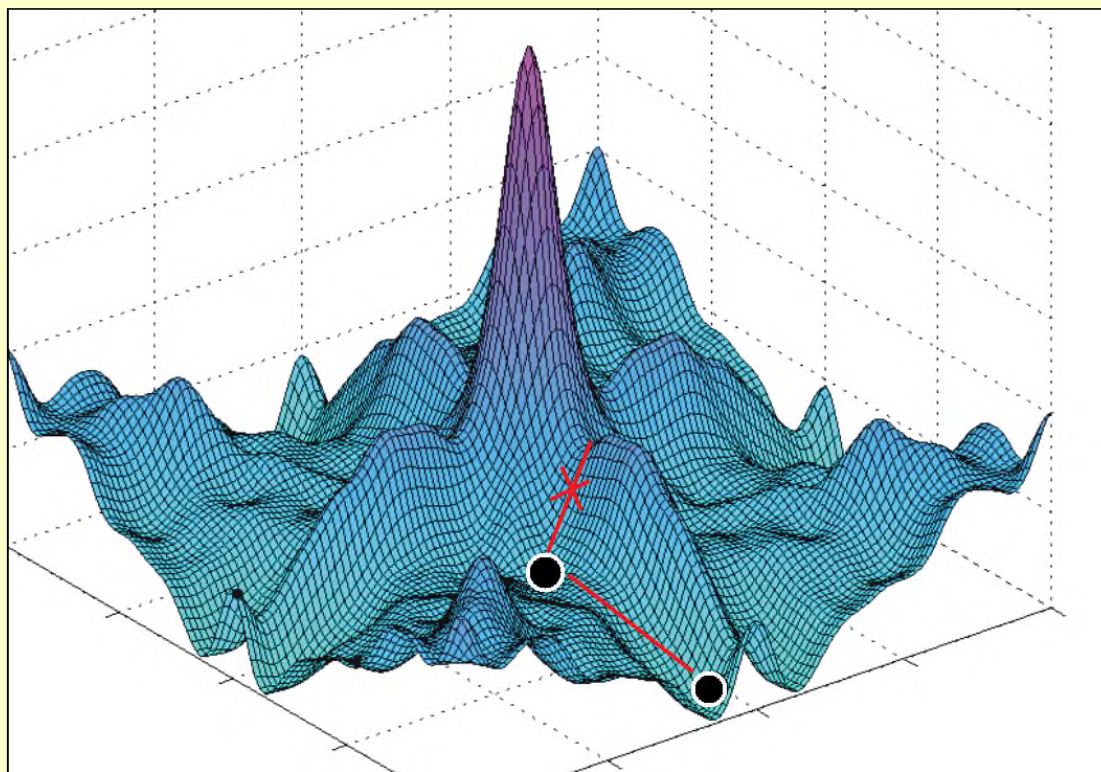


The metastable state is found first and the ground state is found shortly after

# Soft-mode mutation

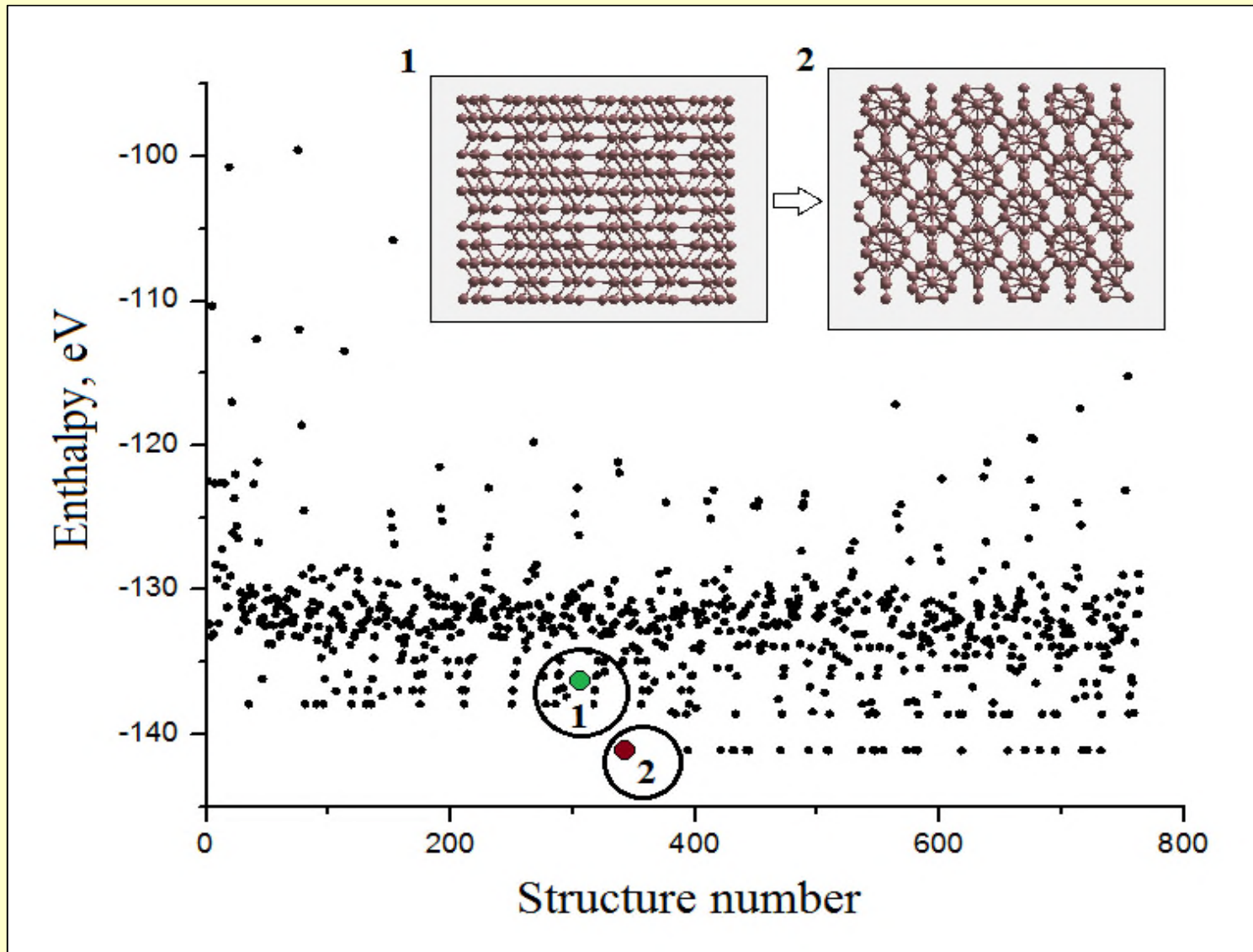
Atoms are moved along the eigenvector of the softest mode  
(both positive and negative directions need to be tried)

Requires the calculation of the dynamical matrix

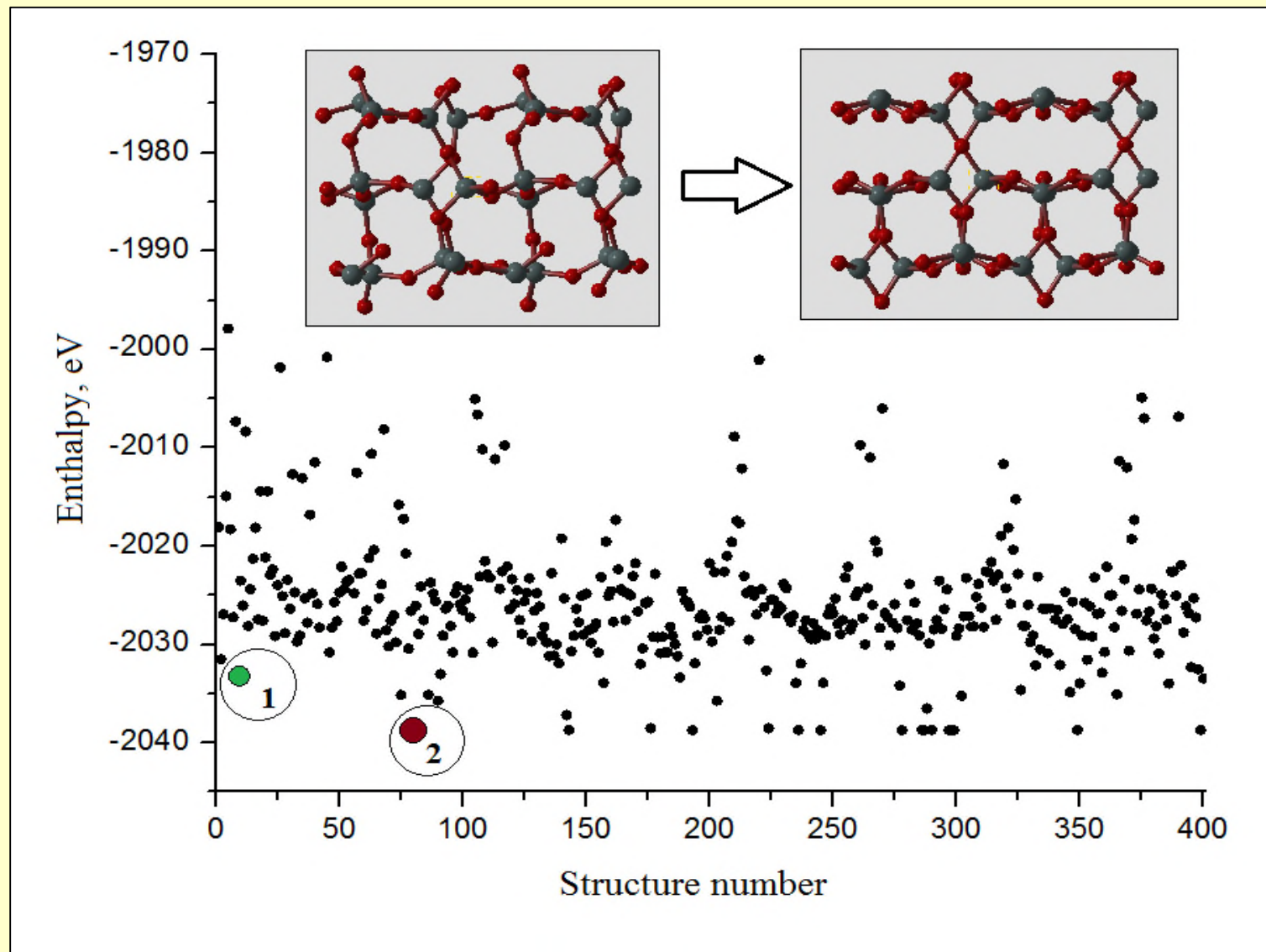




# Soft-mode mutation ( $\gamma$ -B<sub>28</sub>)



# Soft-mode mutation ( $\text{SiO}_2$ -coesite)

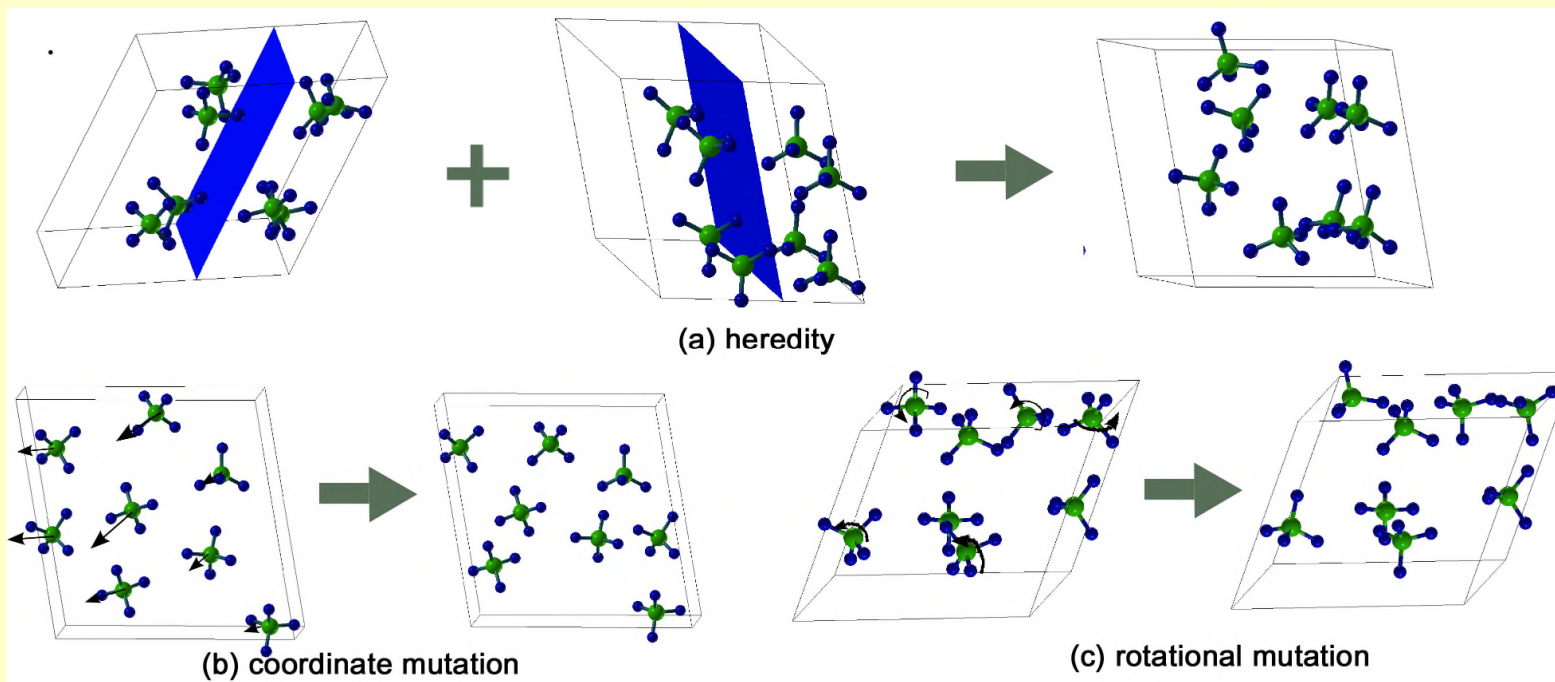


# New modules

- **Optimization of properties**
- **Molecular crystals, nanoparticles**
- **Variable composition**
- **TPS, vcNEB**

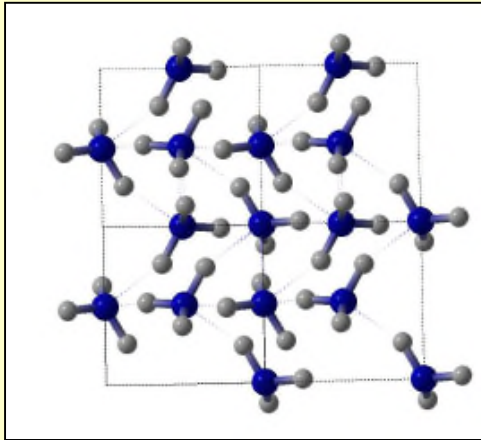
# USPEX for molecular crystals (Zhu & ARO, 2012)

- Apply operators to positions and orientations of the molecules
- Take into account the size and shape of molecules
- Introduce rotational mutation
- Introduce conformational mutation
- Project softmutation into rotational and translational components

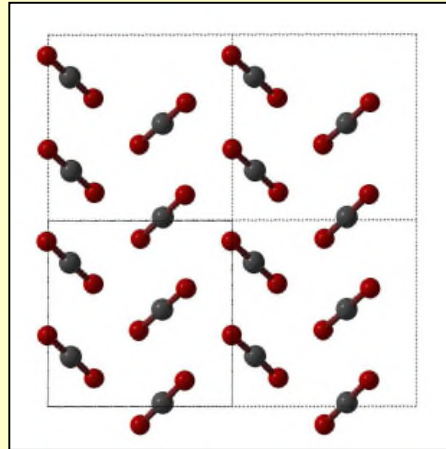


# Molecular crystals

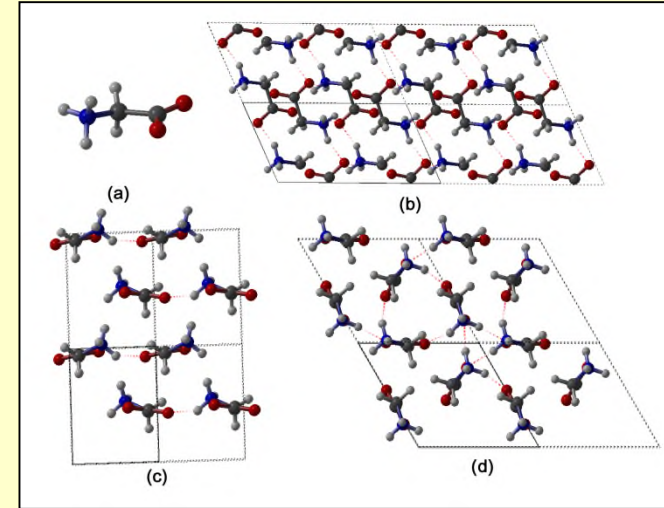
$\text{NH}_3$



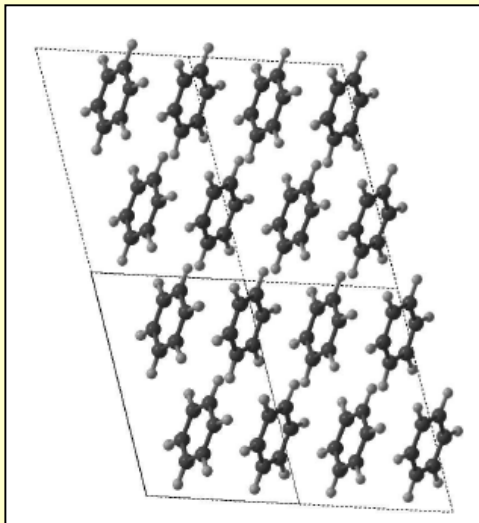
$\text{CO}_2$



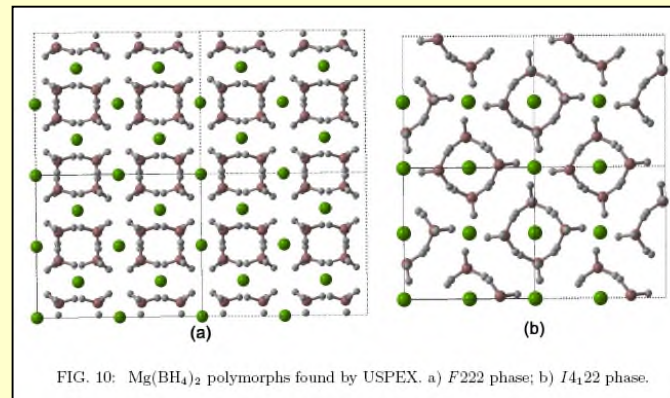
Pharmaceutical: Glycine



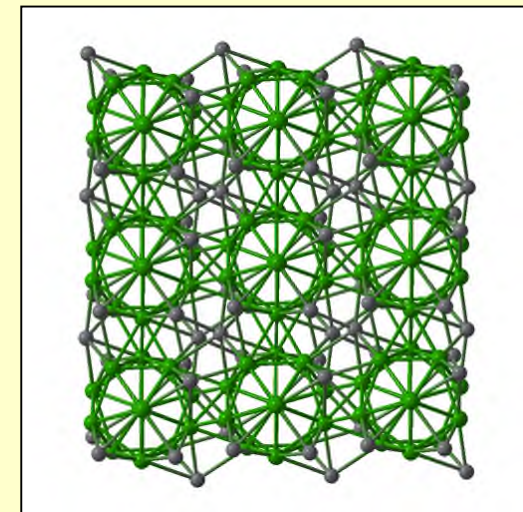
Benzene



$\text{Mg}(\text{BH}_4)_2$



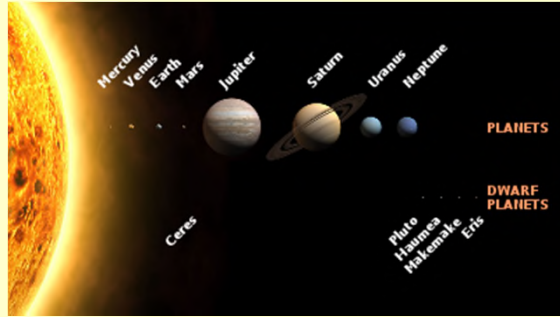
Methane



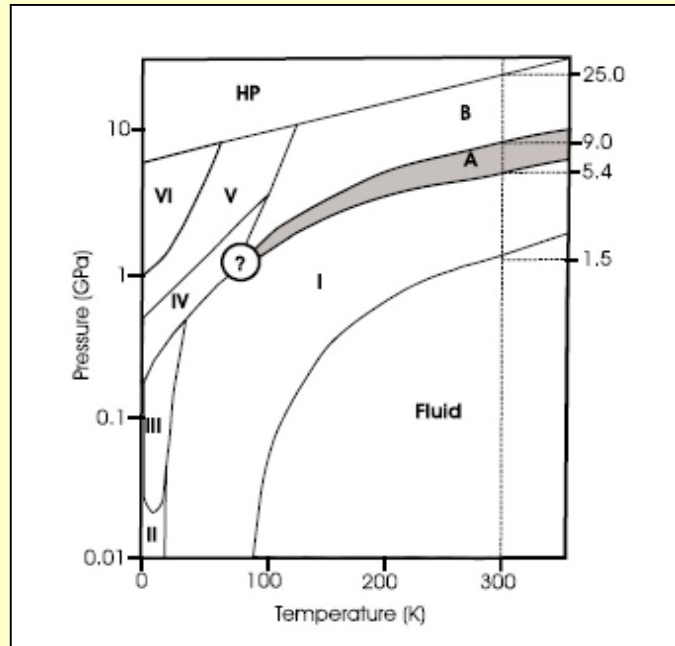
Q. Zhu et al, 2012



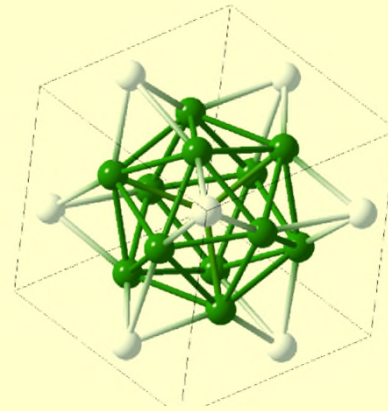
# Molecular crystals: Methane at 10 GPa



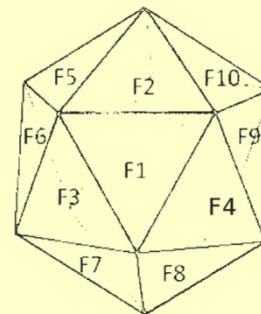
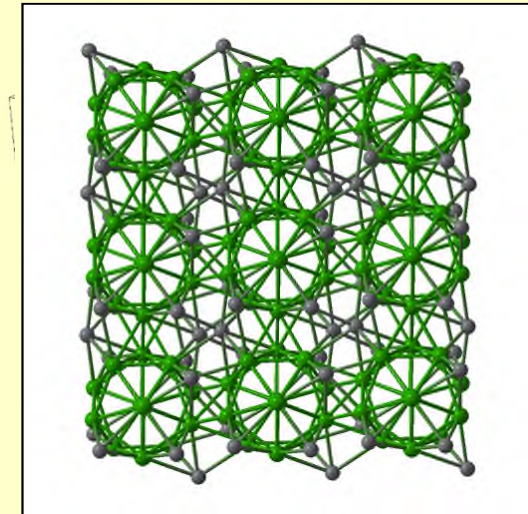
**21 molecules/cell**  
**Nakahata, CPL, 1999**  
**Refined by Sun et al., 2009**



1 day, 8 cores, 105 atoms, first principles



(111) view



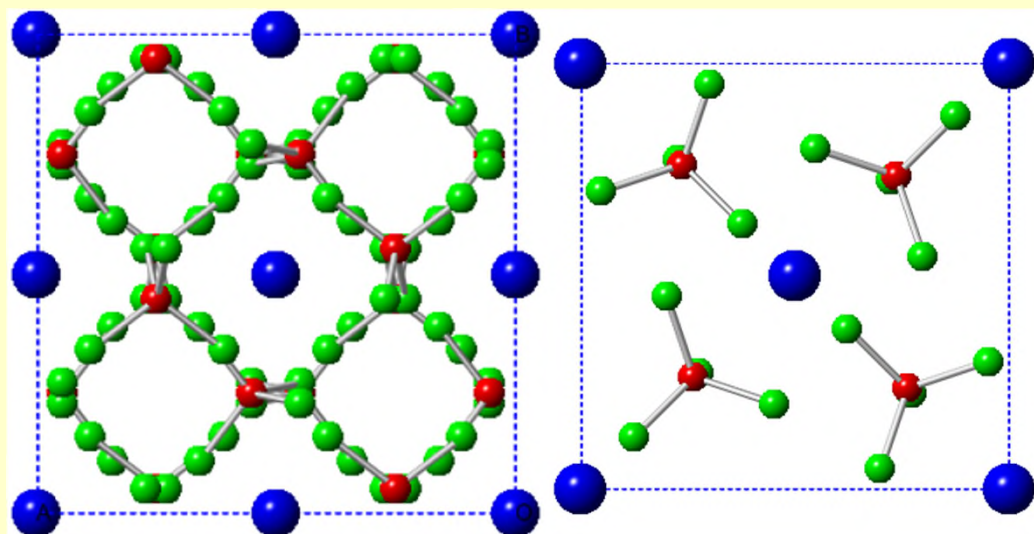
$$21 = 12 + 1 + 8$$

12: icosahedron site  
 1 : icosahedron center  
 8 : outside icosahedron

**Zhu & ARO, 2012**

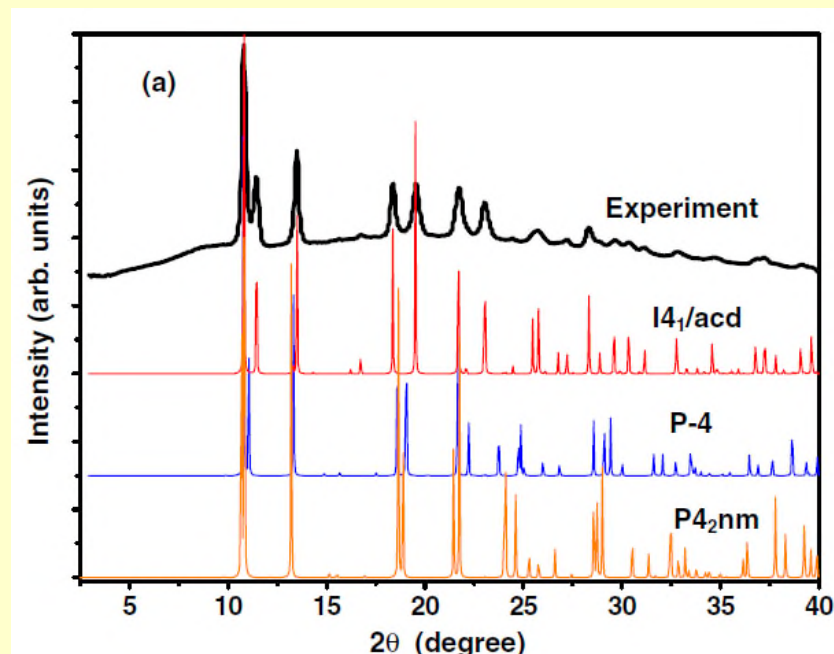
**Solved by: Maynard-Casely et al., 2010**

# $\delta$ -Mg(BH<sub>4</sub>)<sub>2</sub>: example of how theory corrects experimental structure determination



a

b



c

Theoretical  $I4_1/acd$  (a) and “experimental”  $P4_2nm$  (b) structures are very different, but have nearly identical powder XRD patterns (c).

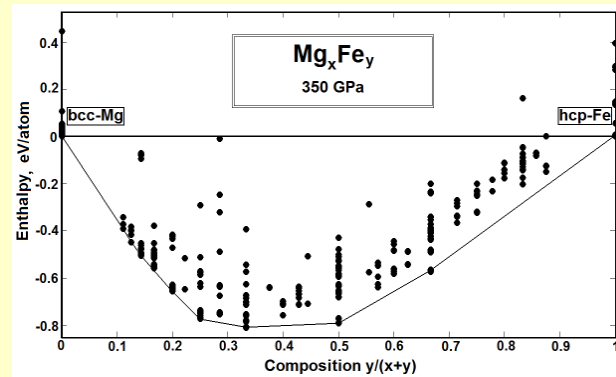
The “experimental” structure is massively energetically unstable and is ruled out!

“Experimental” structure – [Filinchuk et al., *Angew. Chem.* (2011)]

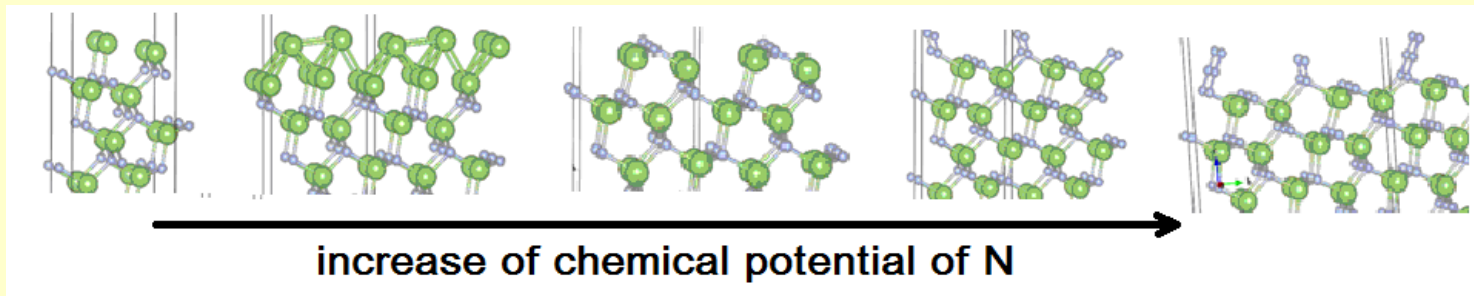
[Zhou, ARO, et al., *Phys. Rev. Lett.* (2012)]

# Variable-composition searches: crucial for discovering new compounds and for surface phenomena

In general, need to include chemical composition as a variable!



Non-trivial ground states of Fe-Mg system at 350 GPa (Lyakhov, ARO et al., 2010)



GaN(10-11) surface reconstructions as a function of chemical potential (Zhu, ARO, et al., PRB 2013)

# New developments of USPEX

A. Improved efficiency

B. Molecular crystals

C. Variable composition

**D. Low-dimensional systems**

-0D (nanoparticles)

-1D (polymers)

-2D (surfaces and 2D-crystals)

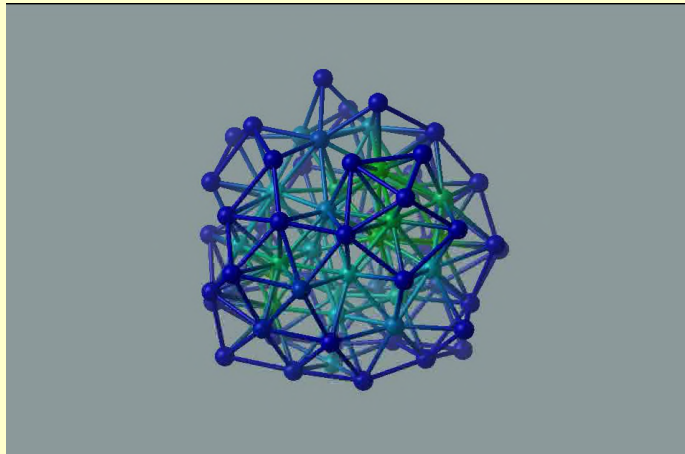
E. Evolutionary metadynamics

F. vc-NEB for transition pathways

G. Transition path sampling

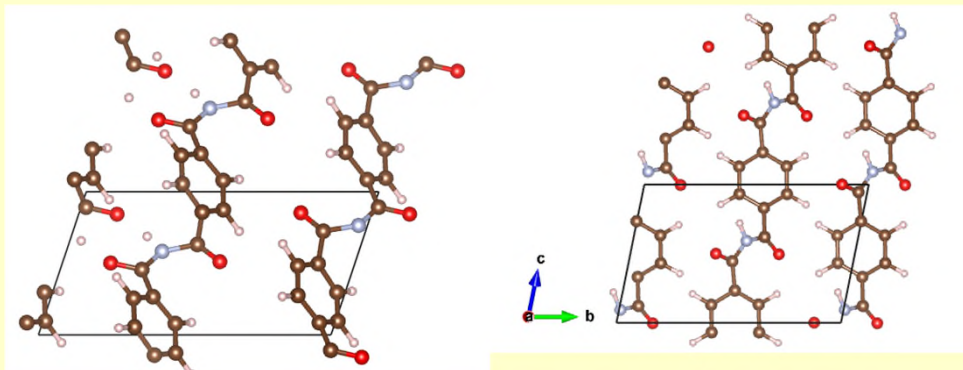
# 1D and 2D: Nanoparticles and polymers

**Famous tough case: Lennard-Jones  
nanoparticle with 75 atoms**



**0-D systems**

**USPEX predicts ground state 3-130 times  
faster than best available methods  
(Lyakhov, ARO et al., 2012)**

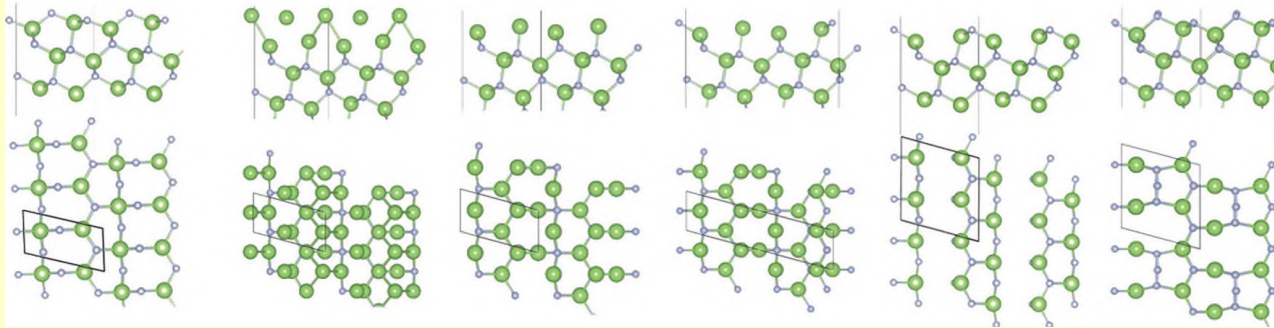


**1-D systems**

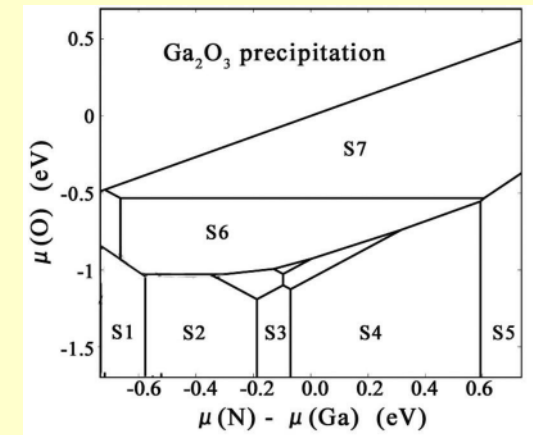
**Predicted high-k dielectric polymers -NH-CO-C<sub>6</sub>H<sub>4</sub>-CO-**



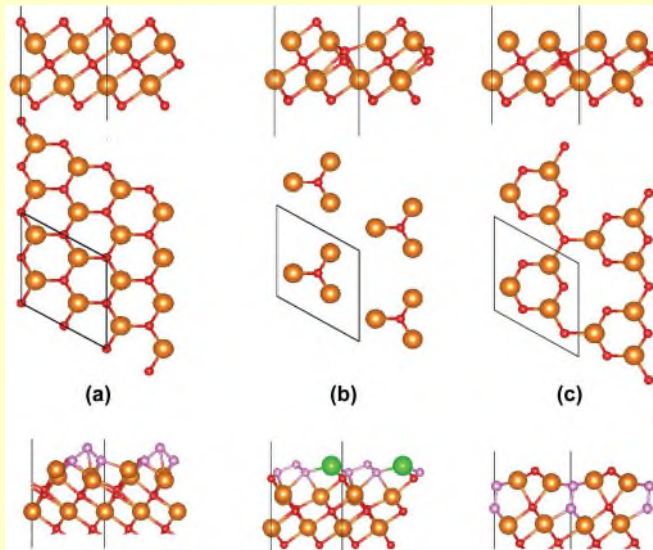
# 2D: Surfaces of crystals, new physics and chemistry... and medicine?



Structure and composition of surface phases of GaN (10-11)

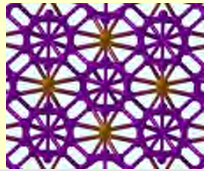


Phase diagram in presence of oxygen



Structure of (111) surface of MgO.

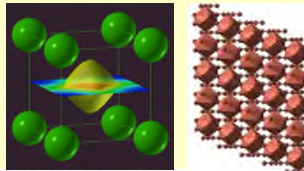
# Applications of this method proved its great utility:



Vol. 451 | 12 February 2009 | doi:10.1038/nature07726

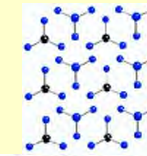
## Ionic high-pressure form of elemental boron

Artem R. Oganov<sup>1,2</sup>, Jihua Chen<sup>3,4</sup>, Carlo Gatti<sup>5</sup>, Yanzhang Ma<sup>6</sup>, Yanming Ma<sup>1,2</sup>, Colin W. Glass<sup>1</sup>, Zhenxian Liu<sup>7</sup>, Tony Yu<sup>1</sup>, Olexandr O. Kuraevych<sup>8</sup> & Vladimir L. Solozhenko<sup>9</sup>



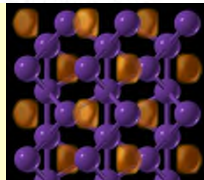
## Exotic behavior and crystal structures of calcium under pressure

Artem R. Oganov<sup>1,2</sup>, Yanming Ma<sup>1</sup>, Ying Xu<sup>1</sup>, Iuri Ersoz<sup>3,4</sup>, Aitor Bergara<sup>4,5</sup>, and Andriy O. Lyakhov<sup>1</sup>  
doi:10.1038/nature08107 | Published online 10 April 2010



## Dissociation of methane under high pressure

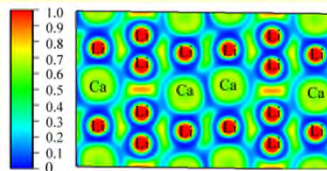
Guoying Gao<sup>1</sup>, Artem R. Oganov<sup>2,3,4</sup>, Yanming Ma<sup>1,5</sup>, Hui Wang<sup>1</sup>, Peifang Li<sup>1</sup>, Tian Cui<sup>1</sup>, Guangtian Zou<sup>1</sup>



nature | Vol. 458 | 12 March 2009 | doi:10.1038/nature07786

## Transparent dense sodium

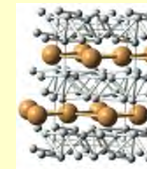
Yanming Ma<sup>1,2</sup>, Mikhail Eretnes<sup>3</sup>, Artem R. Oganov<sup>4,5</sup>, Yu Xie<sup>1</sup>, Ivan Trojan<sup>6</sup>, Sergey Medvedev<sup>7</sup>, Andriy O. Lyakhov<sup>1</sup>, Mario Valle<sup>8</sup> & Vitali Prakapenka<sup>9</sup>



PRL 104, 177005 (2010) | PHYSICAL REVIEW LETTERS | week ending 30 APRIL 2010

## Novel High Pressure Structures and Superconductivity of CaLi<sub>2</sub>

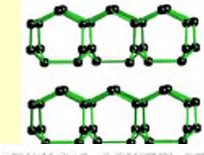
Yu Xie<sup>1,2</sup>, Artem R. Oganov<sup>2,3</sup> and Yanming Ma<sup>1,4</sup>\*



PRL 102, 087005 (2009) | PHYSICAL REVIEW LETTERS | week ending 27 FEBRUARY 2009

## Novel Structures and Superconductivity of Silane under Pressure

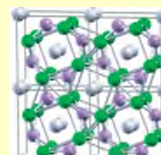
Miguel Martínez-Canales<sup>1,2</sup>, Artem R. Oganov<sup>3,4</sup>, Yanming Ma<sup>5</sup>, Yan Yan<sup>5</sup>, Andriy O. Lyakhov<sup>3</sup> and Aitor Bergara<sup>1,2,6</sup>



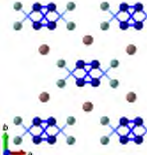
PRL 102, 063501 (2009) | PHYSICAL REVIEW LETTERS | week ending 13 FEBRUARY 2009

## Novel High Pressure Structures of Polymeric Nitrogen

Yanming Ma<sup>1,2,\*</sup>, Artem R. Oganov<sup>2,3</sup>, Zhenwei Li<sup>1</sup>, Yu Xie<sup>1</sup> and Jani Kotakoski<sup>4</sup>



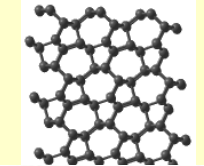
PNAS | A little bit of lithium does a lot for hydrogen  
Eva Zurek<sup>1</sup>, Rold Hoffmann<sup>2</sup>, N. W. Ashcroft<sup>3</sup>, Artem R. Oganov<sup>4,5</sup>, and Andriy O. Lyakhov<sup>1</sup>  
17640-17643 | PNAS | October 20, 2009 | vol. 106 | no. 42 | www.pnas.org/cgi/doi/10.1073/pnas.0905292106



PRL 101, 107002 (2008) | PHYSICAL REVIEW LETTERS | week ending 5 SEPTEMBER 2008

## Superconducting High Pressure Phase of Germane

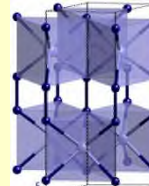
Guoying Gao<sup>1</sup>, Artem R. Oganov<sup>2,3</sup>, Aitor Bergara<sup>4,5</sup>, Miguel Martínez-Canales<sup>4,5</sup>, Tian Cui<sup>1</sup>, Toshiaki Inaka<sup>6</sup>, Yanming Ma<sup>1,2,4</sup> and Guangtian Zou<sup>1</sup>



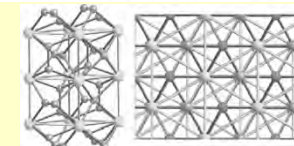
PRL 102, 175506 (2009) | PHYSICAL REVIEW LETTERS | week ending 1 MAY 2009

## Superhard Monoclinic Polymorph of Carbon

Quan Li<sup>1</sup>, Yanming Ma<sup>1,4</sup>, Artem R. Oganov<sup>2,3,5</sup>, Hongbo Wang<sup>1</sup>, Hui Wang<sup>1</sup>, Ying Xu<sup>1</sup>, Tian Cui<sup>1</sup>, Ho-Kwang Mao<sup>4,6</sup> and Guangtian Zou<sup>1</sup>

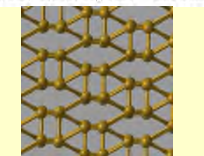


PHYSICAL REVIEW B 79, 132109 (2009)  
Ultra-incompressible phases of tungsten dinitride predicted from first principles  
Hui Wang<sup>1</sup>, Quan Li<sup>1</sup>, Yingwei Li<sup>1</sup>, Ying Xu<sup>1</sup>, Tian Cui<sup>1</sup>, Artem R. Oganov<sup>2,5</sup> and Yanming Ma<sup>1,4</sup>\*

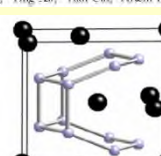


## High-pressure crystal structures and superconductivity of Stannane (SnH<sub>4</sub>)

Guoying Gao<sup>1</sup>, Artem R. Oganov<sup>2,3</sup>, Peifang Li<sup>1</sup>, Aitor Bergara<sup>4,5</sup>, Hui Wang<sup>1</sup>, Tian Cui<sup>1</sup>, Yanming Ma<sup>1,2,4</sup>, Andriy O. Lyakhov<sup>3</sup>, Toshiaki Inaka<sup>6</sup>, and Guangtian Zou<sup>1</sup>  
www.pnas.org/cgi/doi/10.1073/pnas.0905292106 | PNAS | January 28, 2009 | vol. 106 | no. 4 | 1072-1077



PHYSICAL REVIEW B 76, 064101 (2007)  
Structure of the metallic  $\zeta$ -phase of oxygen and isosymmetric nature of the  $\epsilon$ - $\zeta$  phase transition: *Ab initio* simulations  
Yanming Ma<sup>1,2</sup>, Artem R. Oganov<sup>1,3,\*</sup> and Colin W. Glass<sup>1</sup>

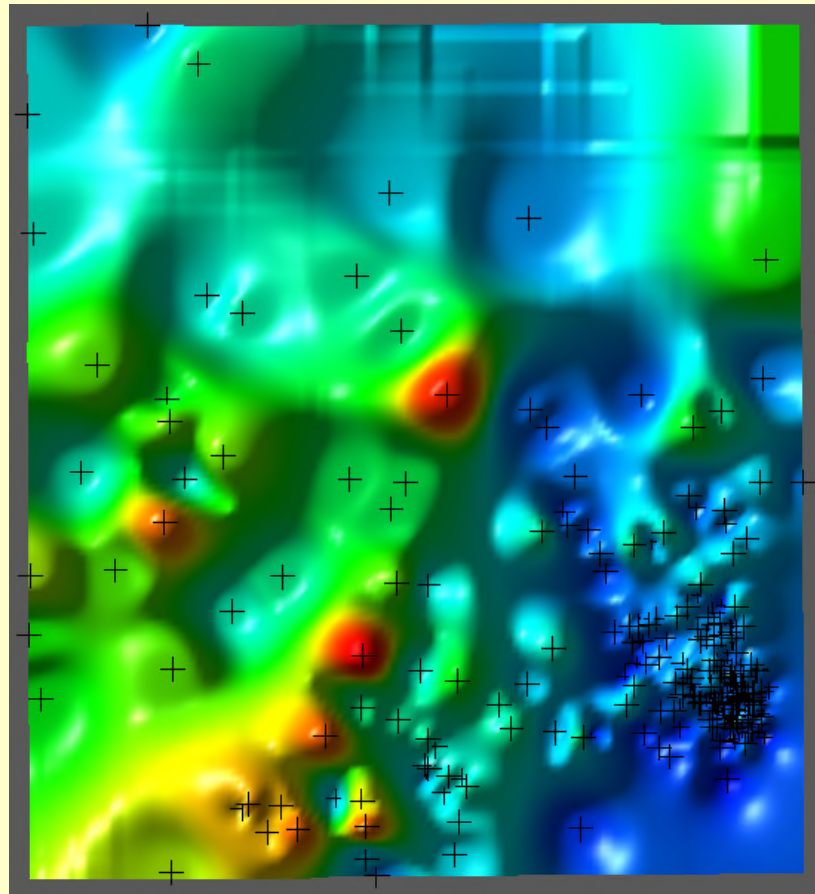


PHYSICAL REVIEW B 79, 054101 (2009)  
Absence of superconductivity in the high-pressure polymorph of MgB<sub>2</sub>  
Yanming Ma<sup>1,3,\*</sup>, Yanchao Wang<sup>1</sup> and Artem R. Oganov<sup>2,3</sup>



Earth and Planetary Science Letters 261 (2008) 31-33  
High-pressure phases of CaCO<sub>3</sub>: Crystal structure prediction and experiment  
Artem R. Oganov<sup>1,4</sup>, Colin W. Glass<sup>1</sup>, Shigeaki Ono<sup>5</sup>

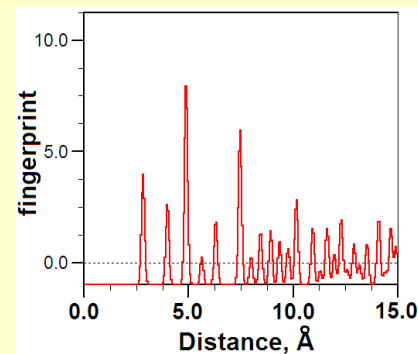
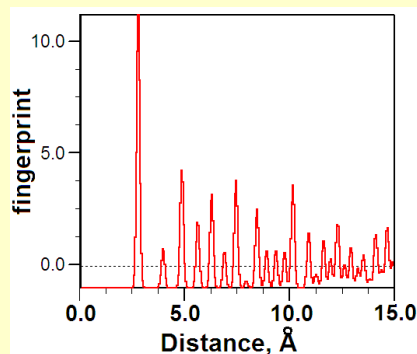
## 2. Why does it work?



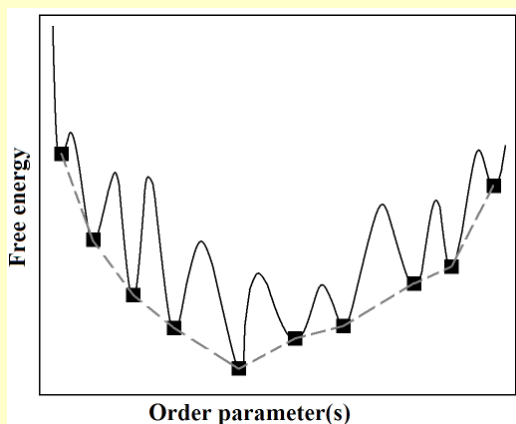
**ARO & Valle, J.Chem.Phys. 130, 104504 (2009)**

# Fingerprint theory is the basis of our analysis

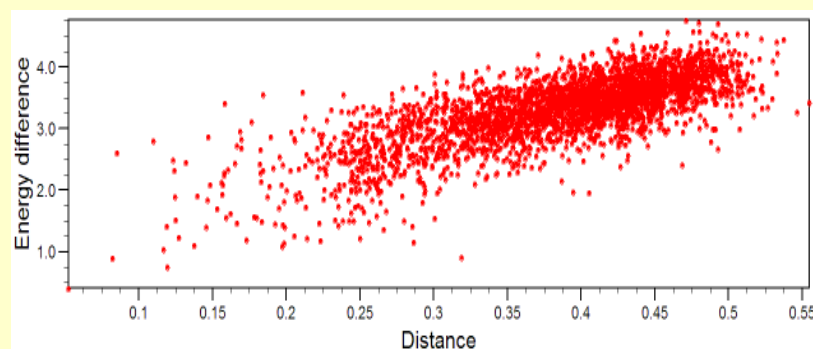
Fingerprint function is a 1D-descriptor of the structure:  
diffraction spectrum, PCF, ...



Difference between 2 structures is given by „distance“, e.g.:  $dist(i, j) = \left( \sum_k |fp_{i_k} - fp_{j_k}|^p \right)^{\frac{1}{p}}$



Pedagogical cartoon



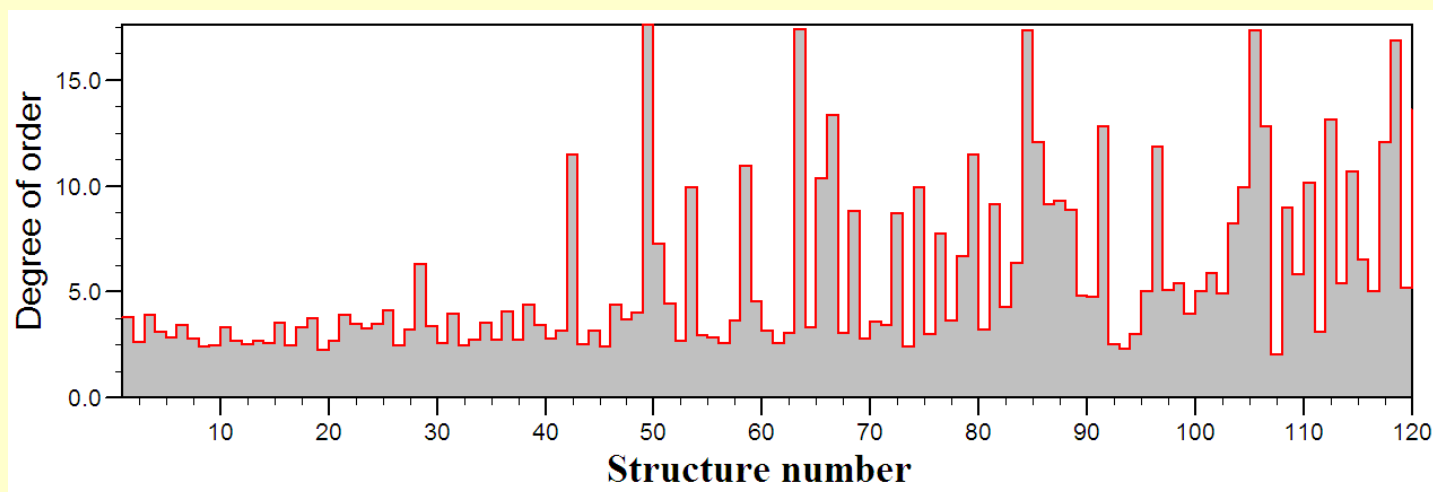
Real system (GaAs): correlation of energy and the distance from the ground-state structure

[ARO & Valle, J. Chem. Phys. 130, 104504 (2009)]



## Feature of evolution: emergence of order from chaos

$$\Pi = \frac{1}{V^{1/3}} \int_0^{\infty} f^2 dR$$

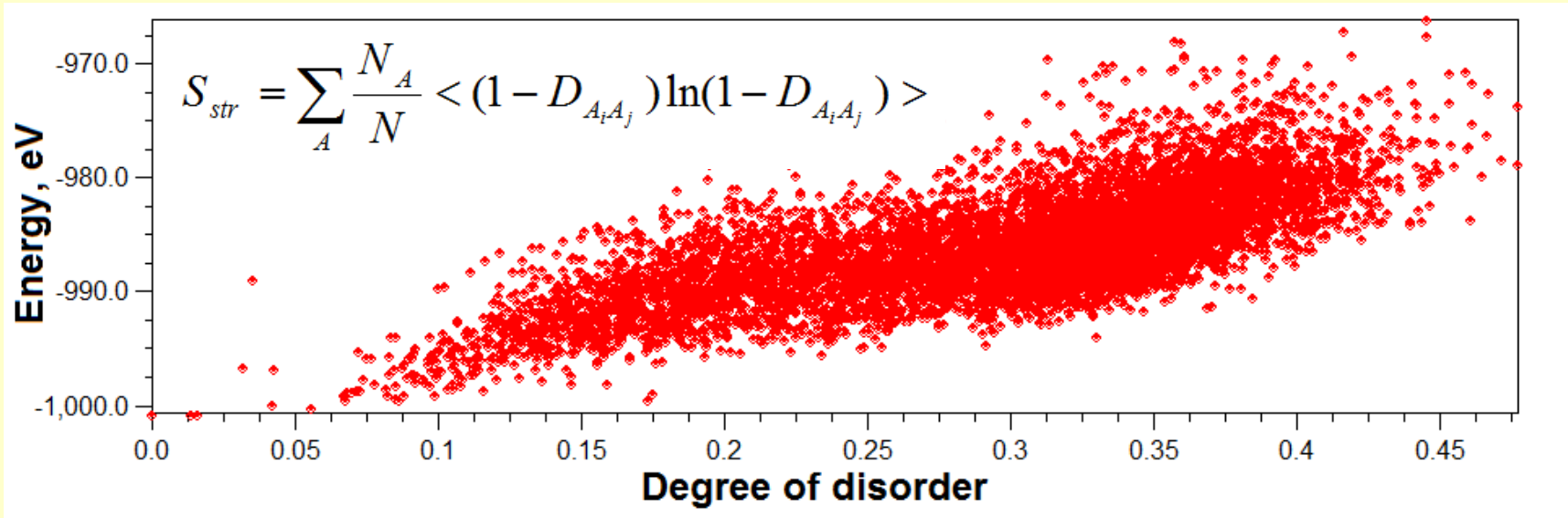


Increase of order during evolutionary simulation of GaAs



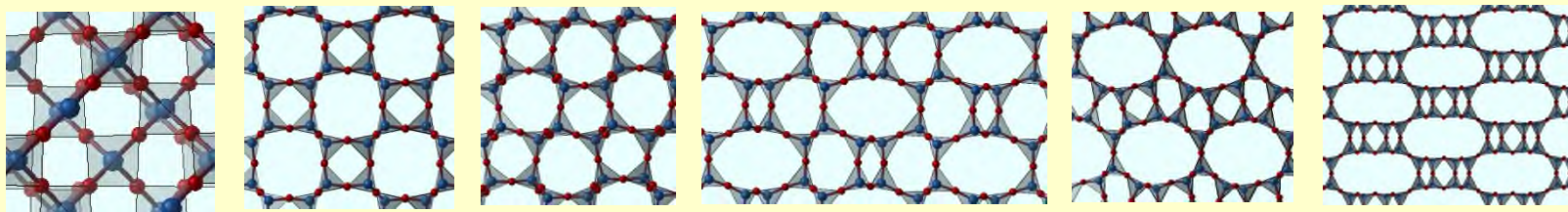
## Statistical confirmation of Pauling's 5th rule:

„The number of essential structural elements of stable structures tends to be small“



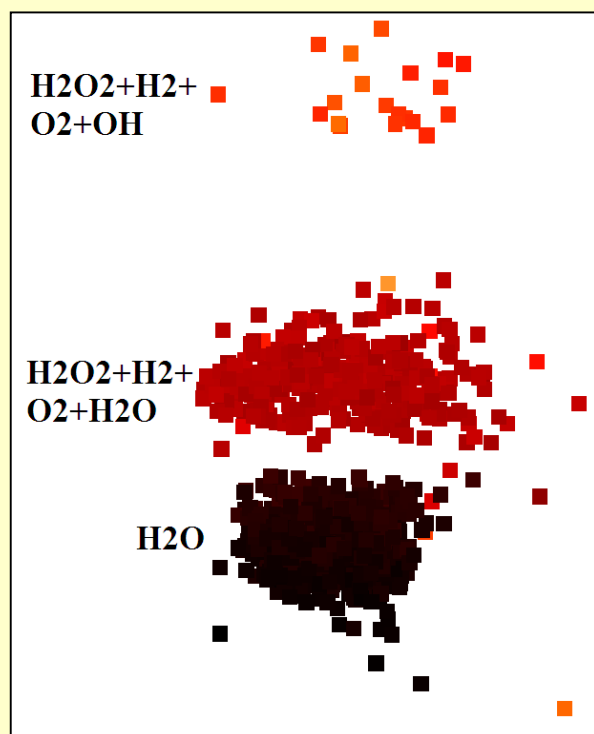
Correlation plot for 6900 structures of SiO<sub>2</sub> with 24 atoms/cell

Some of the (many) remarkable silicate frameworks:

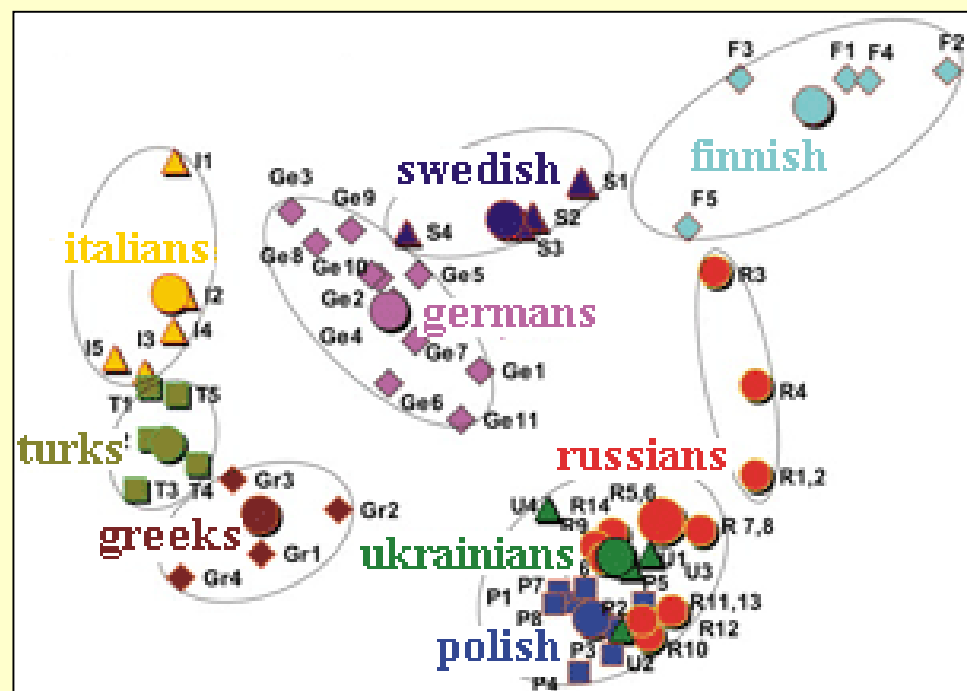


Increasing energy

## Grouping structures into similarity classes: quest for more insight in complex systems



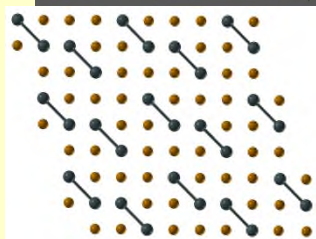
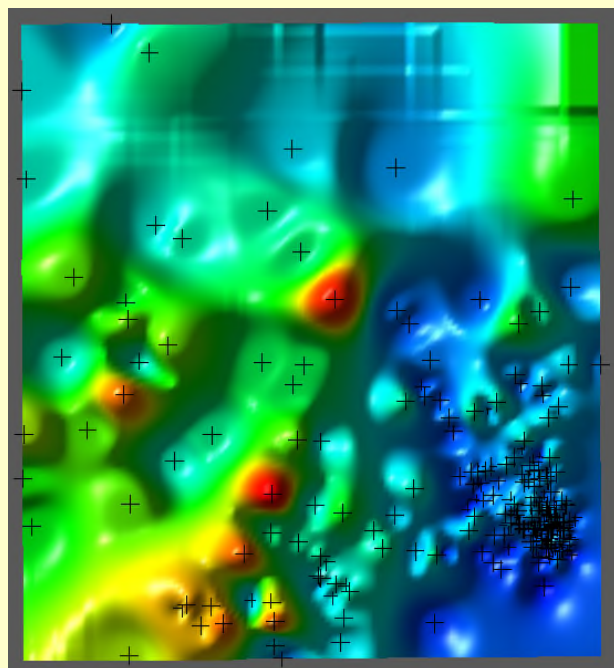
Distance-preserving mapping  
of crystal structures of H<sub>2</sub>O  
(*darker* – lowest E, *lighter* – higher E).



DNA grouping in Europe

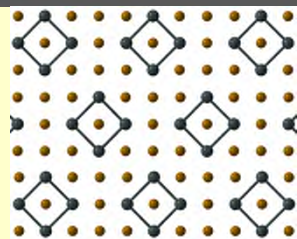
# Visualizing energy landscapes

## Au<sub>8</sub>Pd<sub>4</sub> - simple



-61.960 eV

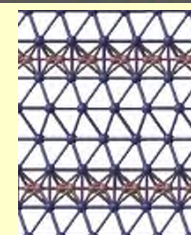
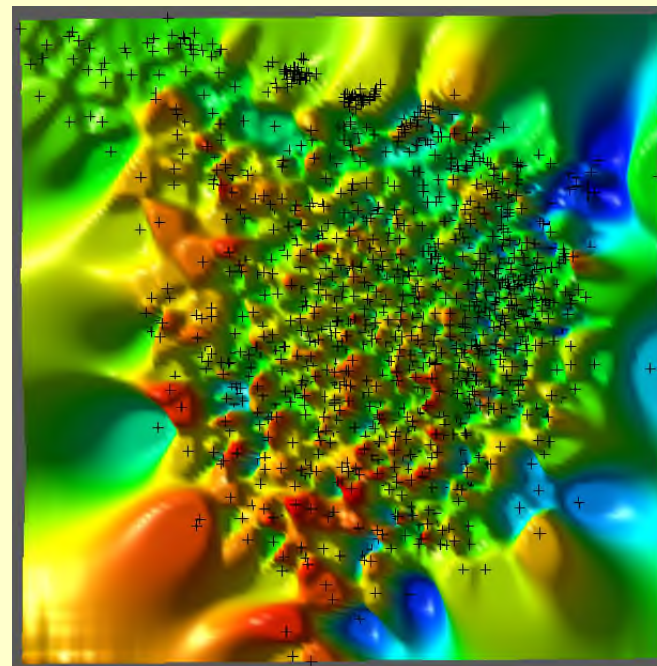
From USPEX



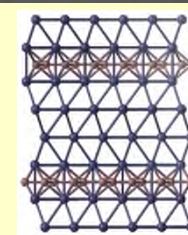
-61.957 eV

Cluster expansion

## L<sub>4</sub>J<sub>8</sub> - complex



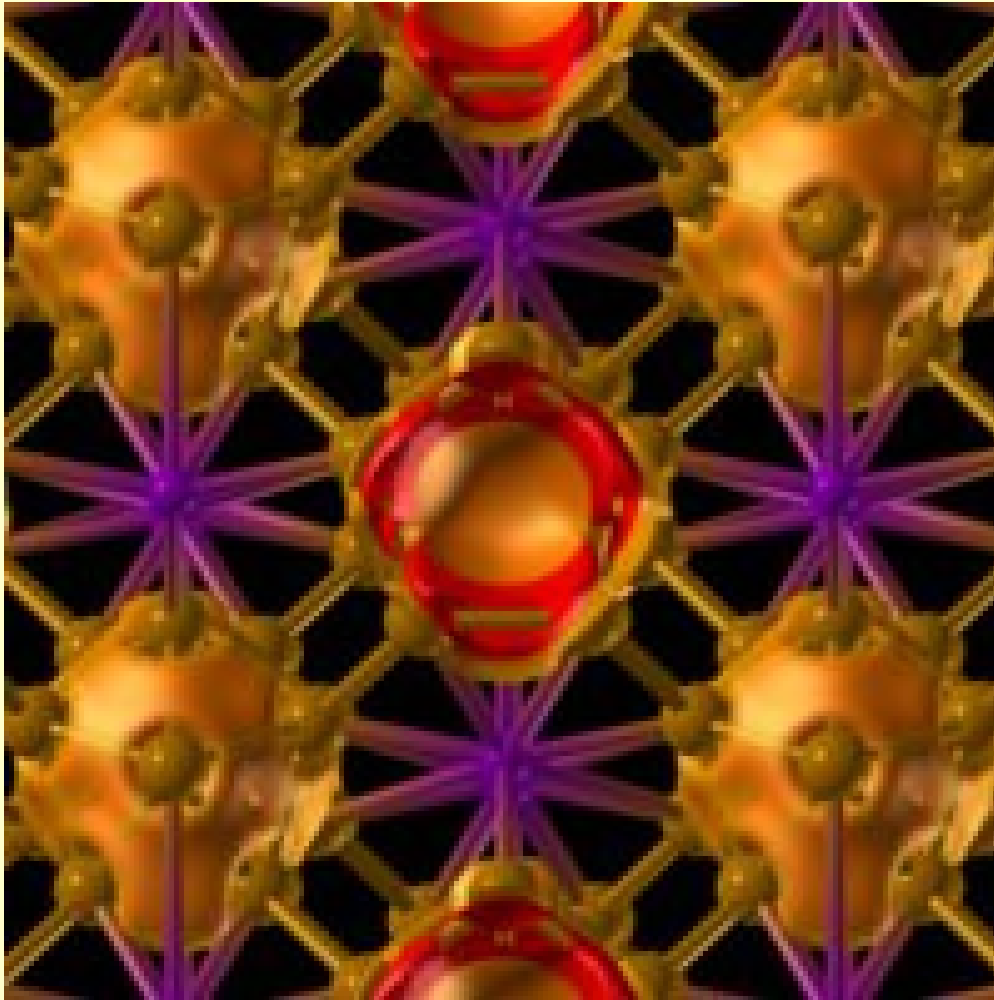
-99.12ε



-99.05ε

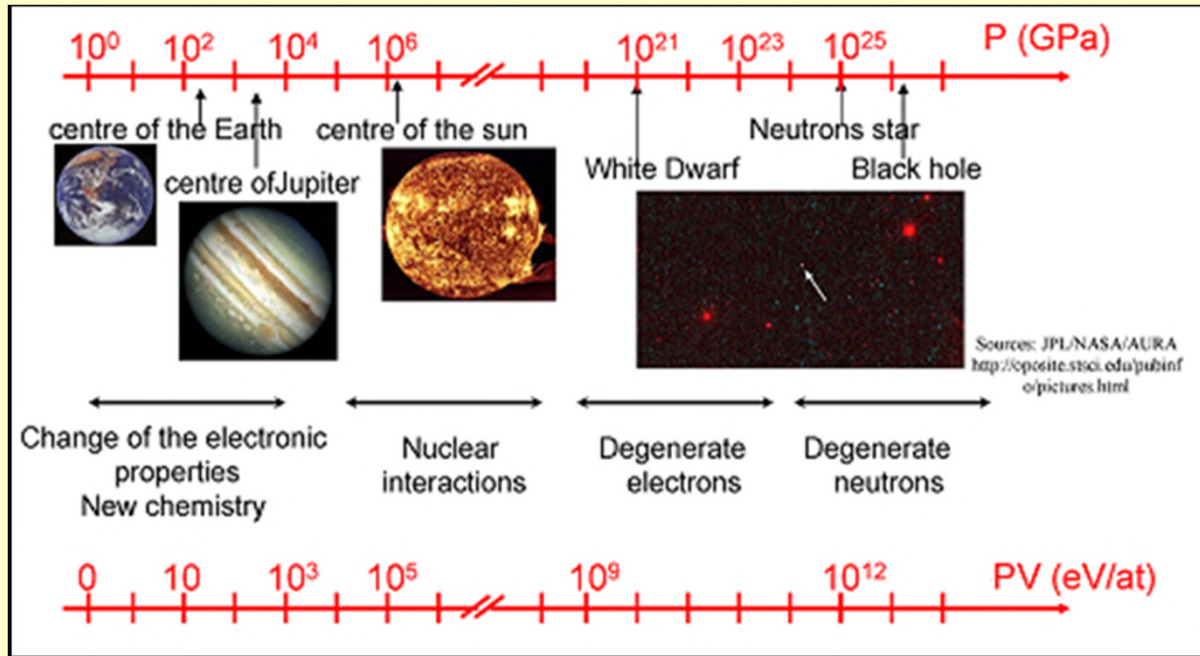
Binary Lennard-Jones crystal (R<sub>L</sub>:R<sub>J</sub>=1:2)

### 3. Predicting new materials



- Matter under pressure
- Materials with target properties

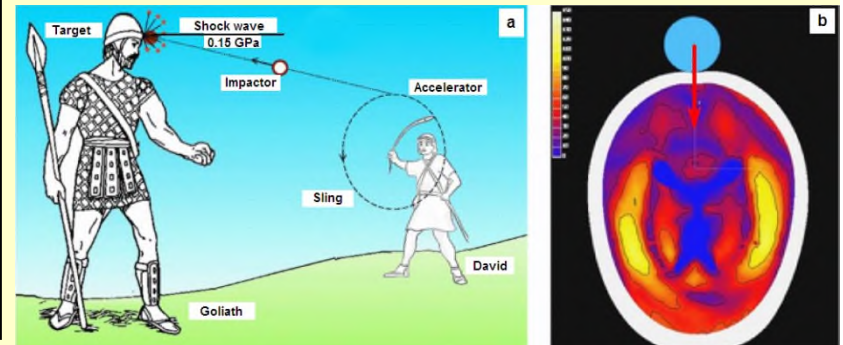
# Matter under pressure: new phenomena and ubiquity in nature



P.W. Bridgman  
1946 Nobel Prize for Physics

**Units:** 100 GPa = 1 Mbar =  $\frac{200x}{\text{---}}$

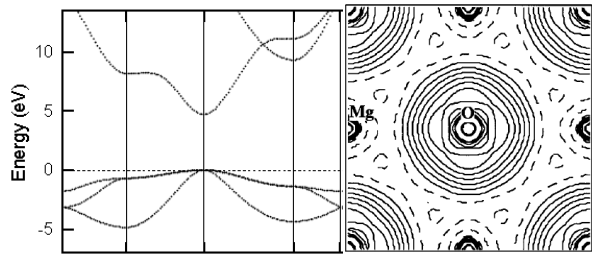
The diagram shows a large elephant and a small red high-heeled shoe. The elephant is positioned above the horizontal line of the fraction, and the shoe is below it. The text '200x' is placed to the left of the fraction line.





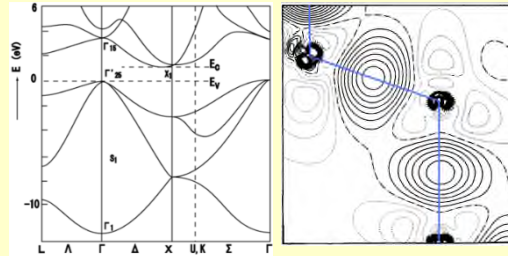
# Types of chemical bonding

## IONIC



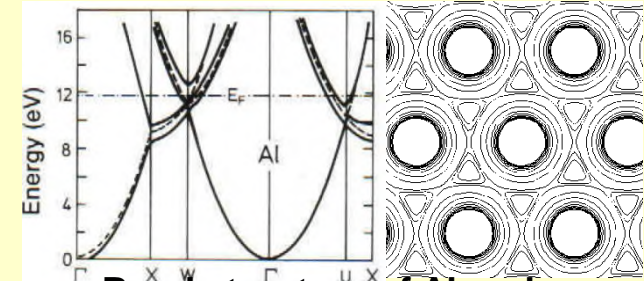
Band structure and deformation electron density of MgO

## COVALENT



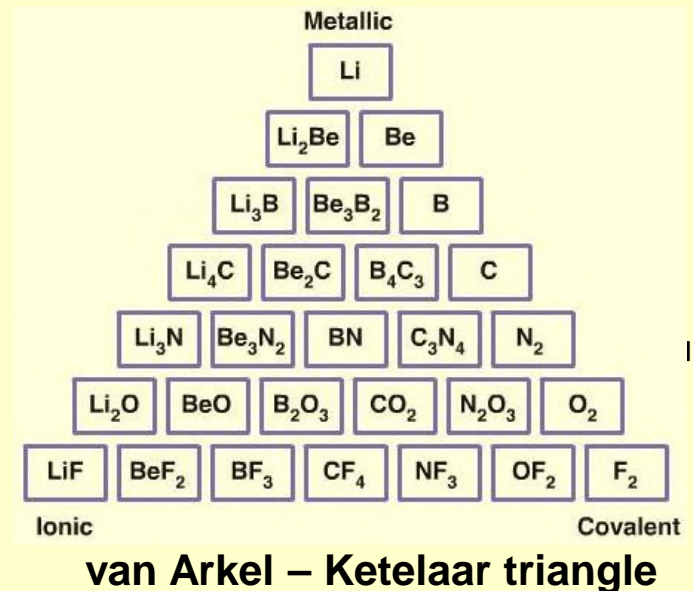
Band structure and deformation electron density of Si

## METALLIC



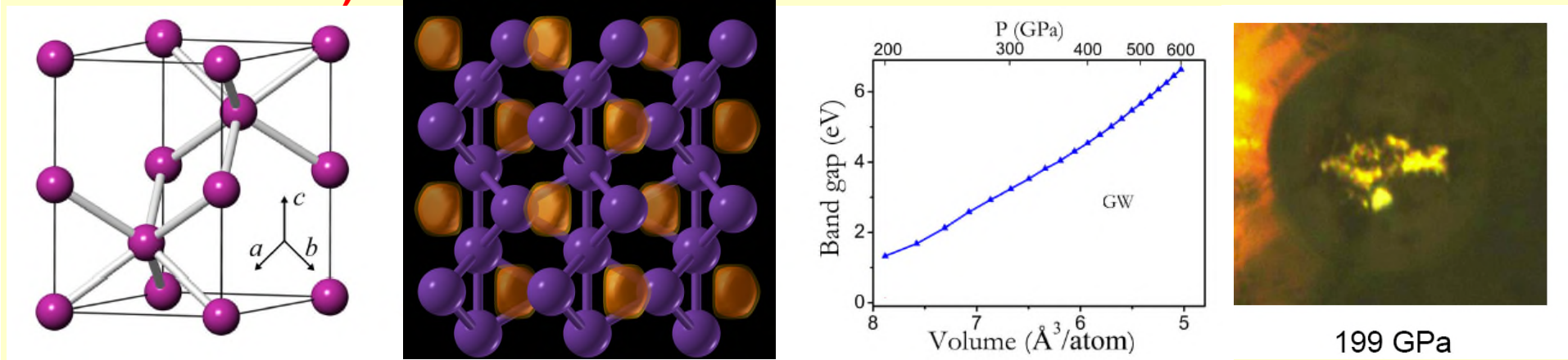
Band structure of Al and valence electron density of Cu

- **Ionic:** atoms have charges. Electrons localize on anions. Purely ionic bonding cannot exist.
- **Covalent:** shared electron pairs between the atoms. Directional.
- **Metallic:** delocalized, due to electron gas. Under sufficient pressure all matter should metallize.
- **van der Waals:** universal (present in all materials).



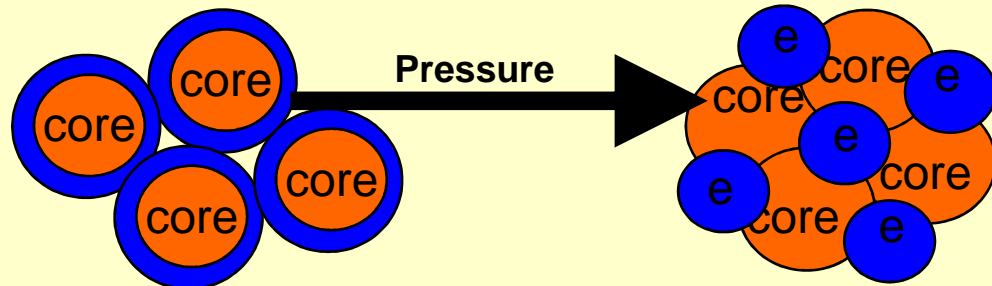
# Metals under pressure

- Studied Fe, Na, Li, K, Rb, Cs, Ca, CaLi<sub>2</sub>.
- Competition between steric (close packing), valence-electronic (Hume-Rothery), and core-electronic effects.
- Often open and low-symmetry structures emerge.
- **Surprise: core electrons become essential and cause demetallization of Na and Li. Na becomes transparent at 200 GPa (Ma, Eremets, Oganov, Nature 2009).**



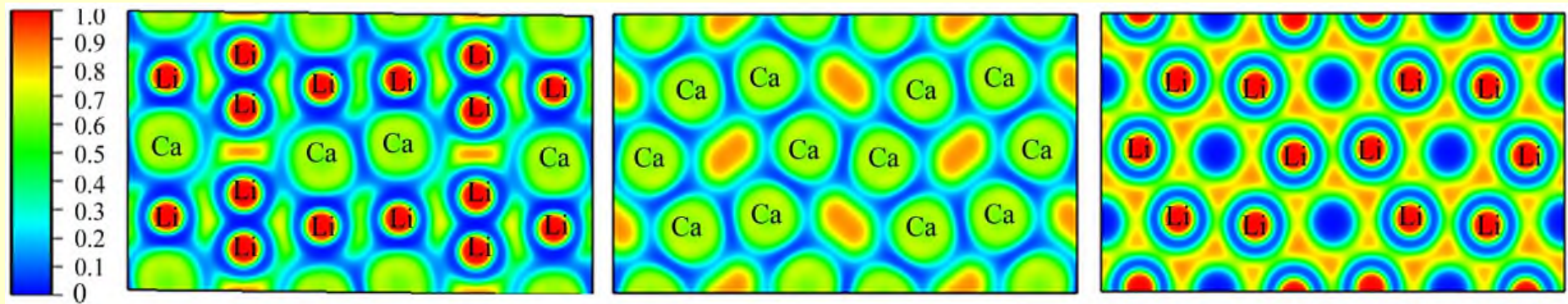
Localized interstitial electron pairs make Na insulating.

Explanation: Models of Ashcroft (1999, 2008)



# CaLi<sub>2</sub>: another illustration of the importance of core electrons

- Feng (2007), Debessai (2008), Tse (2009) gave mutually inconsistent results.
- Our study (Xie et al., 2010) reconciled theory and experiment and found unique new structures with Li-Li pairing.



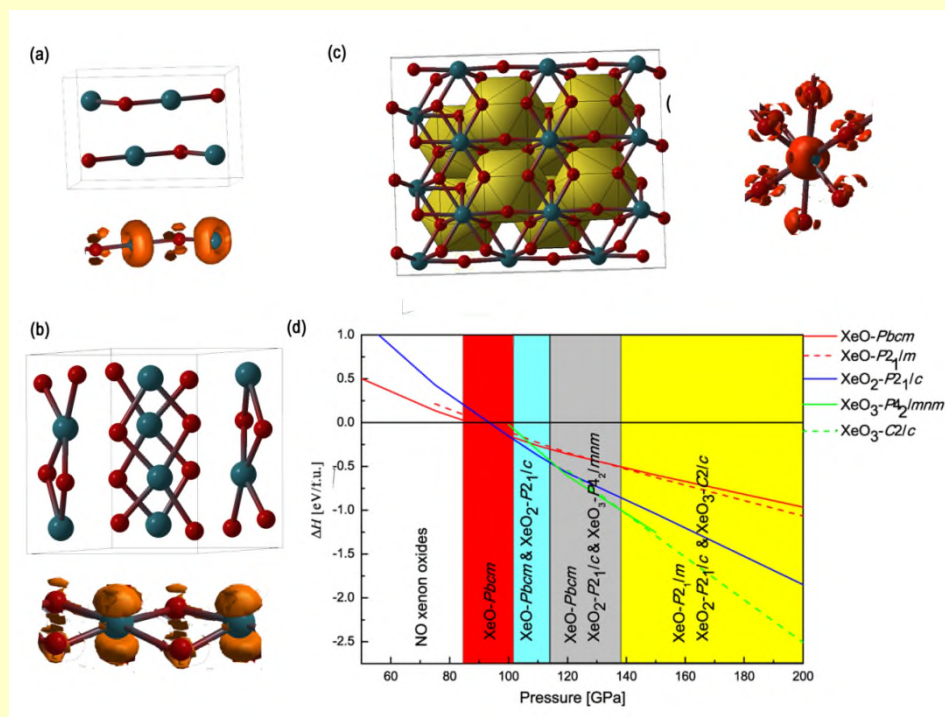
PRL 104, 177005 (2010) PHYSICAL REVIEW LETTERS 30 APRIL 2010

Novel High Pressure Structures and Superconductivity of CaLi<sub>2</sub>

Yu Xie,<sup>1,2</sup> Artem R. Oganov,<sup>2,3</sup> and Yanming Ma<sup>1,\*</sup>

# Ionic solids under pressure

- Studied  $\text{CaCO}_3$ ,  $\text{MgCO}_3$ ,  $\text{TiO}_2$ ,  $\text{MgSiO}_3$ ,  $\text{Al}_2\text{O}_3$ ,...
- Tendency to close packing, increase of coordination – as expected.
- Metallization under pressure (e.g.,  $\text{MgO}$  at 21 TPa) – as expected.
- **Xe oxides become stable at >80 GPa. Unexpectedly strong (~50%) Xe→O charge transfer.**
- **Oxidation state increases with pressure:  $\text{Xe}^{2+} \rightarrow \text{Xe}^{4+} \rightarrow \text{Xe}^{6+}$**

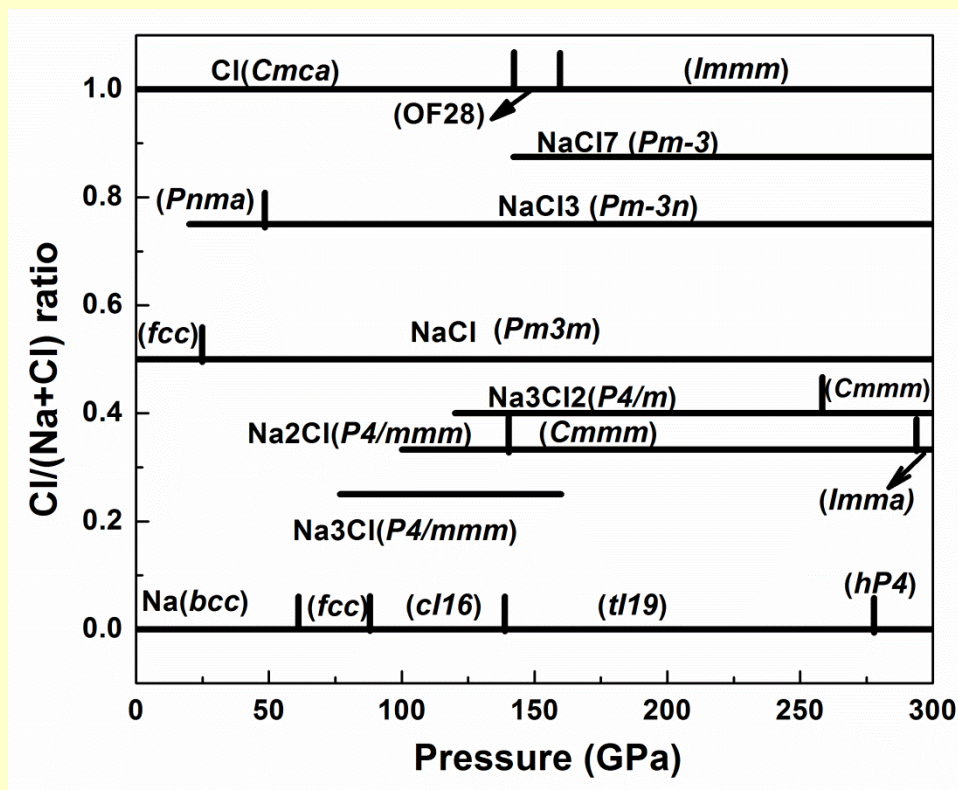


**Surprise: significantly ionic stable  $\text{XeO}_n$  compounds (Zhu, Jung, & ARO, *Nature Chemistry*, 2013)**

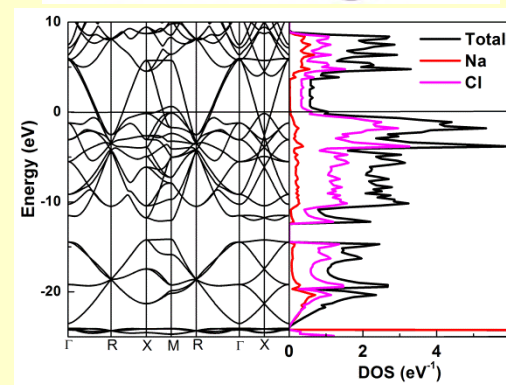
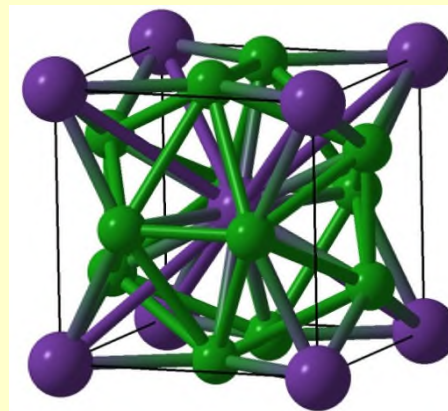


# Ionic solids under pressure

- Na-Cl system: compounds  $\text{Na}_3\text{Cl}$ ,  $\text{Na}_2\text{Cl}$ ,  $\text{Na}_3\text{Cl}_2$ ,  $\text{NaCl}$ ,  $\text{NaCl}_3$ ,  $\text{NaCl}_7$  are all stable under pressure (Zhang, ARO, et al. <http://arxiv.org/abs/1211.3644>). Confirmed by experiment!



Phase stability in the Na-Cl system

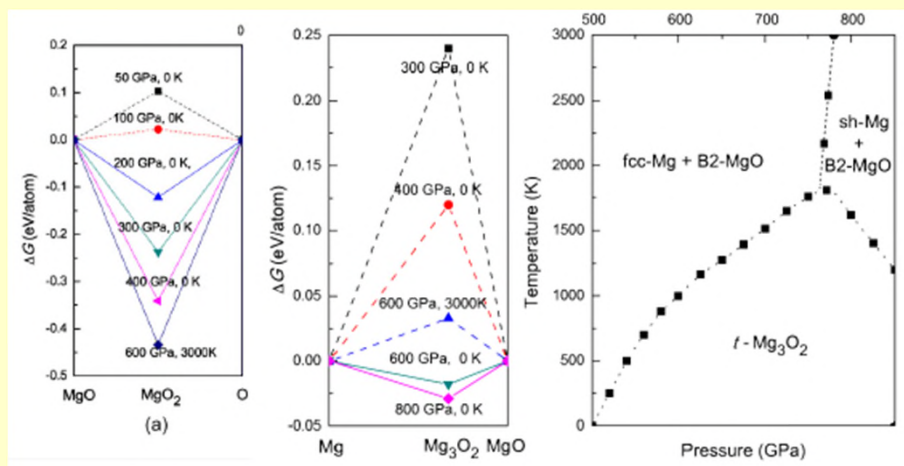


Atomic and electronic structure of cubic  $\text{NaCl}_3$

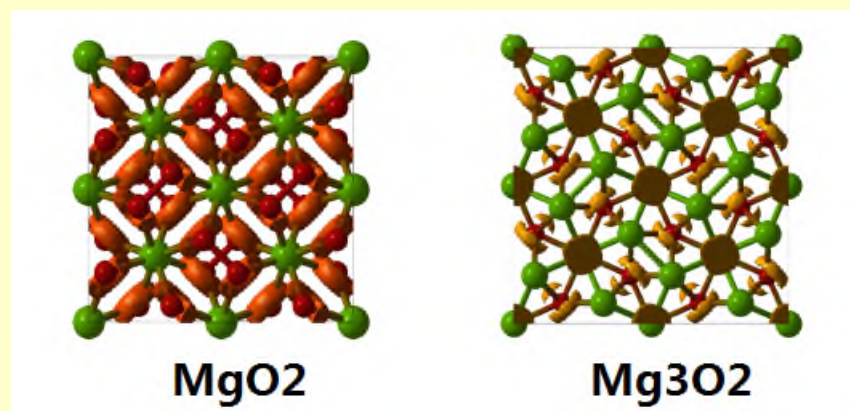


# Ionic solids under pressure

**Mg-O system: compounds  $\text{Mg}_3\text{O}_2$ ,  $\text{MgO}$ ,  $\text{MgO}_2$  are all stable under pressure (Zhu, ARO, 2013).**



**Phase stability in the Mg-O system**



**Crystal structure and electron localization function of  $\text{MgO}_2$  and  $\text{Mg}_3\text{O}_2$**

# van der Waals solids under pressure

- Studied Xe, CO<sub>2</sub>, CH<sub>4</sub>, SiH<sub>4</sub>, GeH<sub>4</sub>, SnH<sub>4</sub>, glycine, graphane (CH).
- Polymerization of molecules (CO<sub>2</sub>, CH<sub>4</sub>, SiH<sub>4</sub>, GeH<sub>4</sub>, SnH<sub>4</sub>) – as expected.
- Metallization (Xe, CO<sub>2</sub>, CH<sub>4</sub>, SiH<sub>4</sub>, GeH<sub>4</sub>, SnH<sub>4</sub>) – as expected.

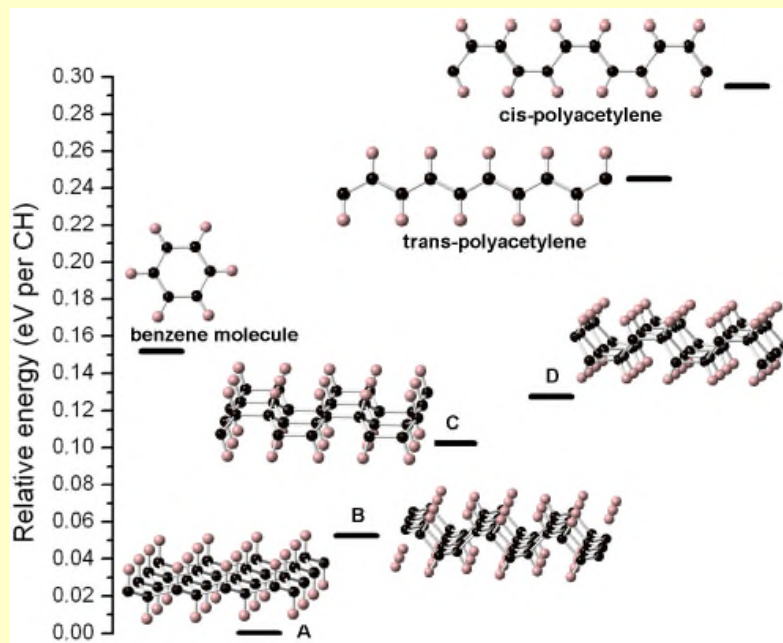


Fig. 3. The relative energy (in eV per CH; relative to single-sheet graphane)

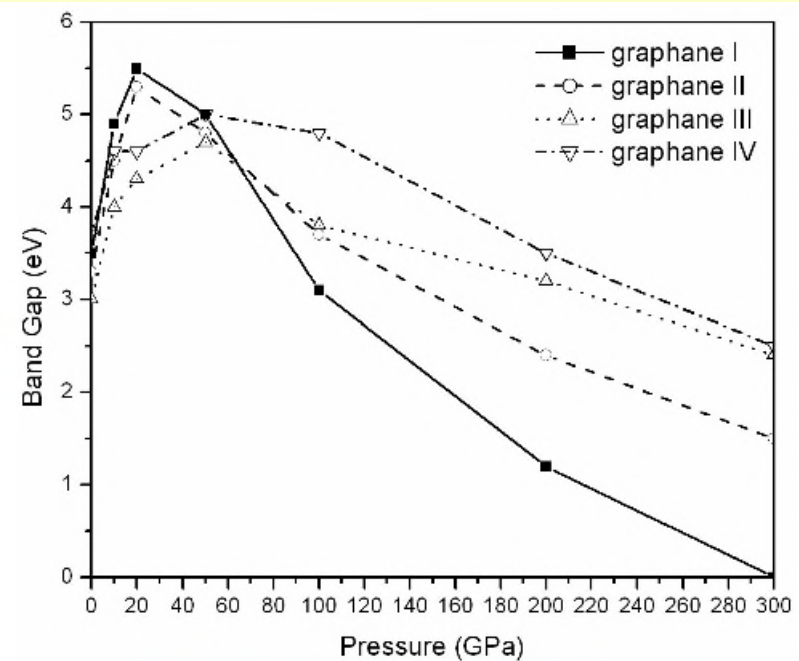


Fig. 6. Band gap of four graphanes as a function of pressure.

**Surprise: Many isomers exist for graphane (CH). Graphane is more stable than benzene (C<sub>6</sub>H<sub>6</sub>). (Wen, Hoffmann, ARO, 2011)**

# van der Waals solids under pressure

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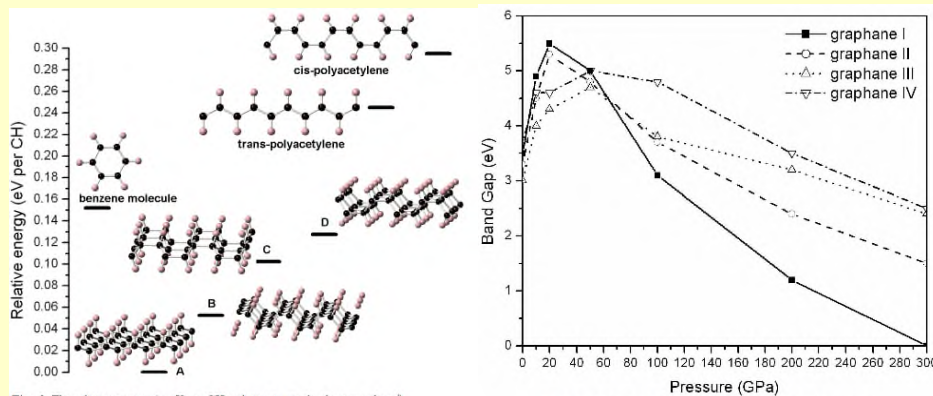
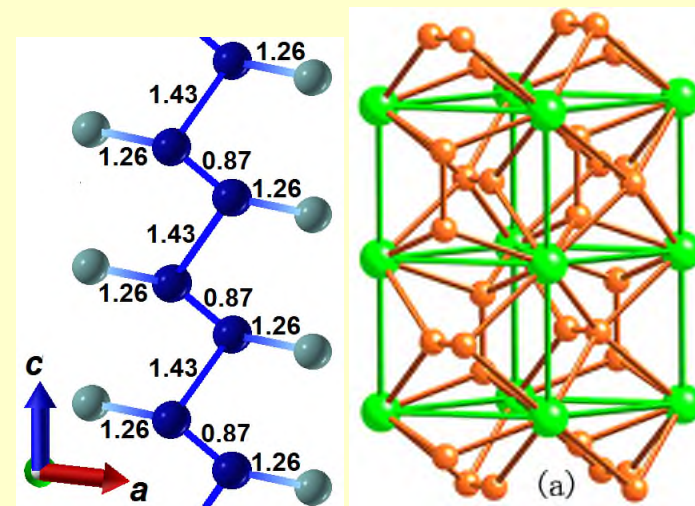


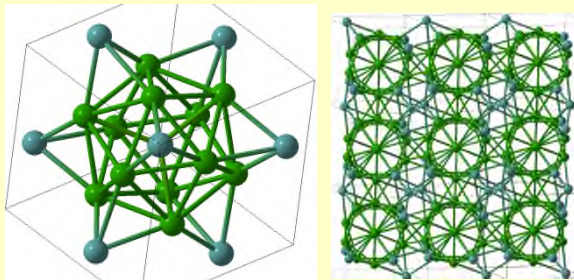
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Fig. 6. Band gap of four graphanes as a function of pressure.

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**Surprise: ubiquitous formation of perhydrides under pressure (GeH<sub>4</sub> – left, SnH<sub>4</sub> – right, LiHn).  
H-H distances of ~0.80-0.90 Å.**

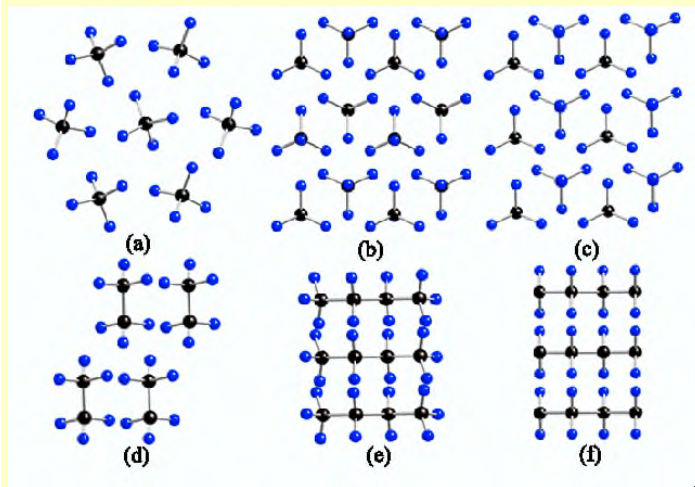


**Surprise: Icosahedral structure for plastic phase of methane at ~10 GPa (Zhu, ARO, 2012).  
Consistent with (Maynard-Caseley, 2010).**

# Planet Neptune has an internal source of heat. What is it?

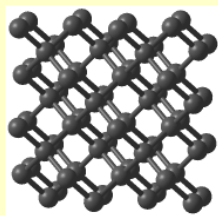
CH<sub>4</sub>

- Uranus and Neptune: H<sub>2</sub>O:CH<sub>4</sub>:NH<sub>3</sub> = 59:33:8.
- Neptune has internal heat source (Hubbard'99).
- Ross'81 (and Benedetti'99):  
 $\text{CH}_4 = \text{C}(\text{diamond}) + 2\text{H}_2$ . Sinking of diamond – main source of heat in Neptune?
- Theory (Ancilotto'97; Gao'2010) confirms this.

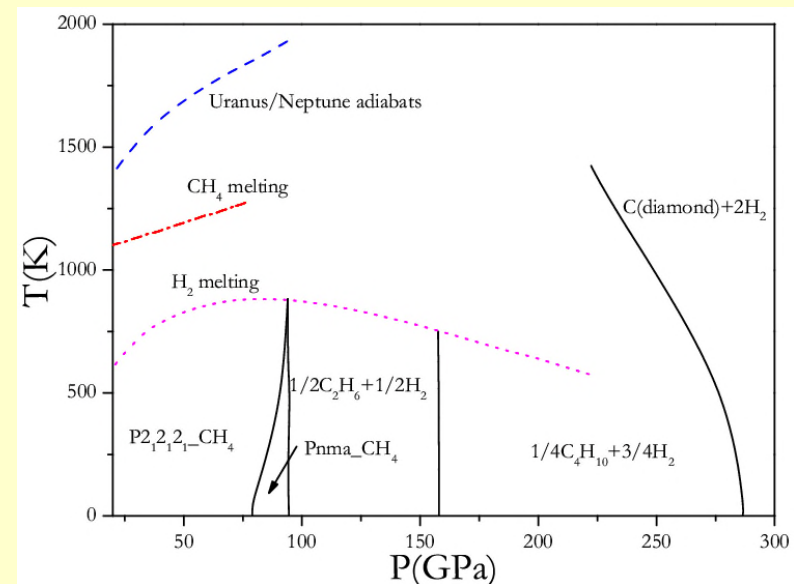


methane

hydrocarbons



diamond

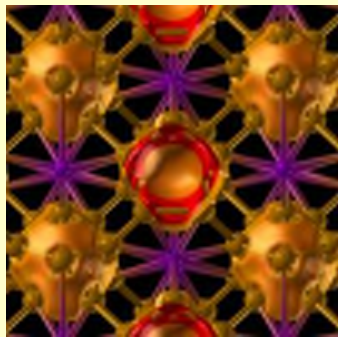


[Gao, ARO et al., *J. Chem. Phys.* 133, 144508 (2010) ]

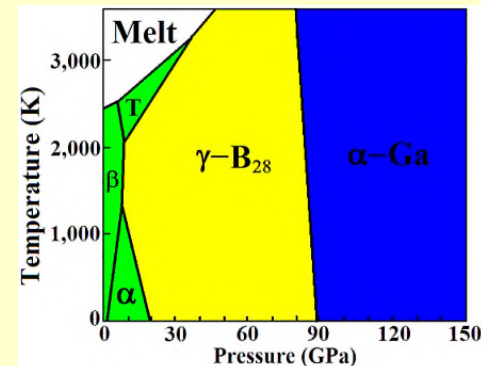


# Covalent solids under pressure

- Studied B, C, N, O, Cl, CO<sub>2</sub>, WN<sub>2</sub>, MgB<sub>2</sub>, B-C, B-P.
- Multiple bonds become less favorable, coordination increases under pressure (N, Cl, CO<sub>2</sub>).
- Metallization (B, C, N, O, Cl).
- **Surprise: before metallization – formation of unusual partially ionic states (B, H).**



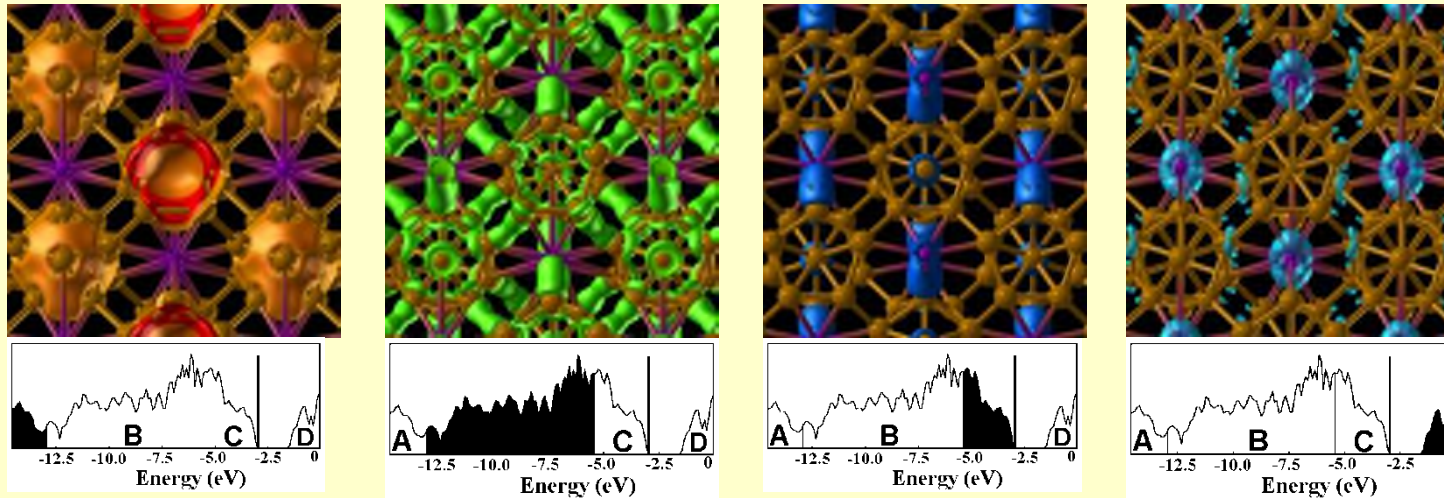
NaCl-like structure of  $\gamma\text{-B}_{28}$   
(Oganov et al., Nature 2009)



First phase diagram of boron  
(Oganov et al., Nature 2009)



# Non-trivial chemistry of boron – field for new discoveries



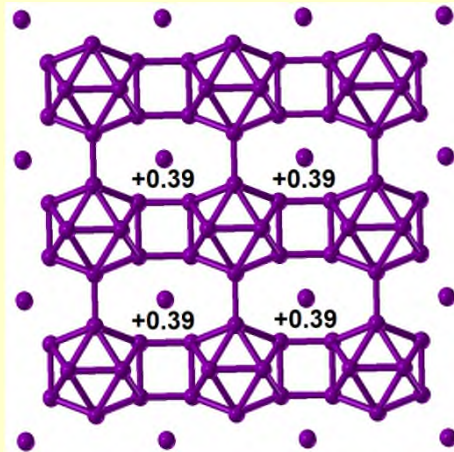
Charge separation between  $B_{12}$ -icosahedra and  $B_2$ -pairs is clear from DOS

Theoretical (Oganov, 2009 & 2011) and experimental (Mondal, 2011) atomic charges in  $\gamma$ - $B_{28}$

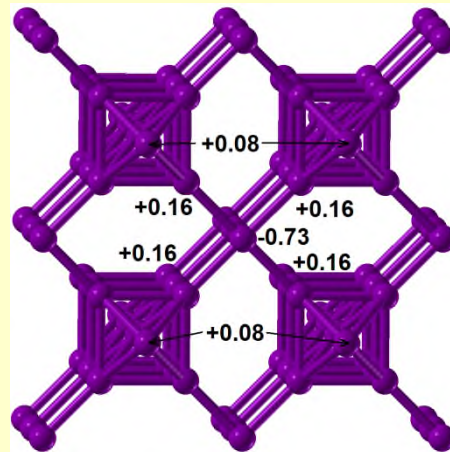
Site	GGA	EXX	HSE06	Experiments (Mondal, 2011)	
B1	+0.26	+0.34	+0.31	+0.41	+0.81
B2	-0.18	-0.31	-0.21	-0.19	-0.19
B3	+0.00	-0.04	+0.00	+0.06	-0.03
B4	+0.07	+0.25	+0.07	-0.14	-0.44
B5	+0.04	+0.11	+0.04	+0.00	+0.05

# Charge transfer (i.e. partial ionicity) – ubiquitous in metastable structures of boron

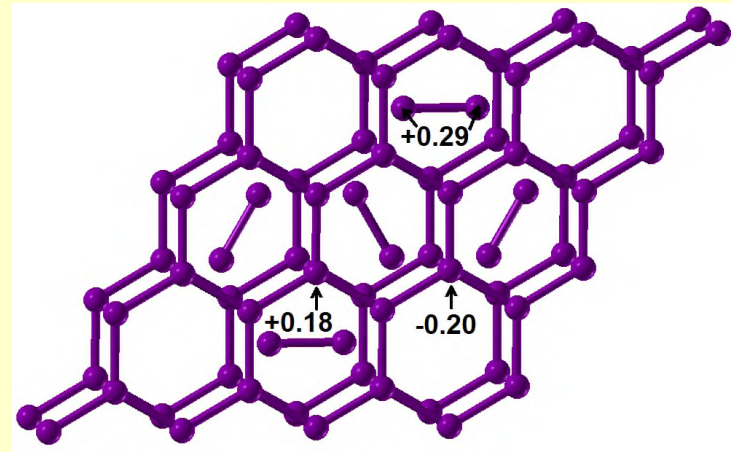
B



Icosahedra are negatively, single B-atoms positively charged



One B atom has 4 single bonds, achieved due to -1 charge on that atom

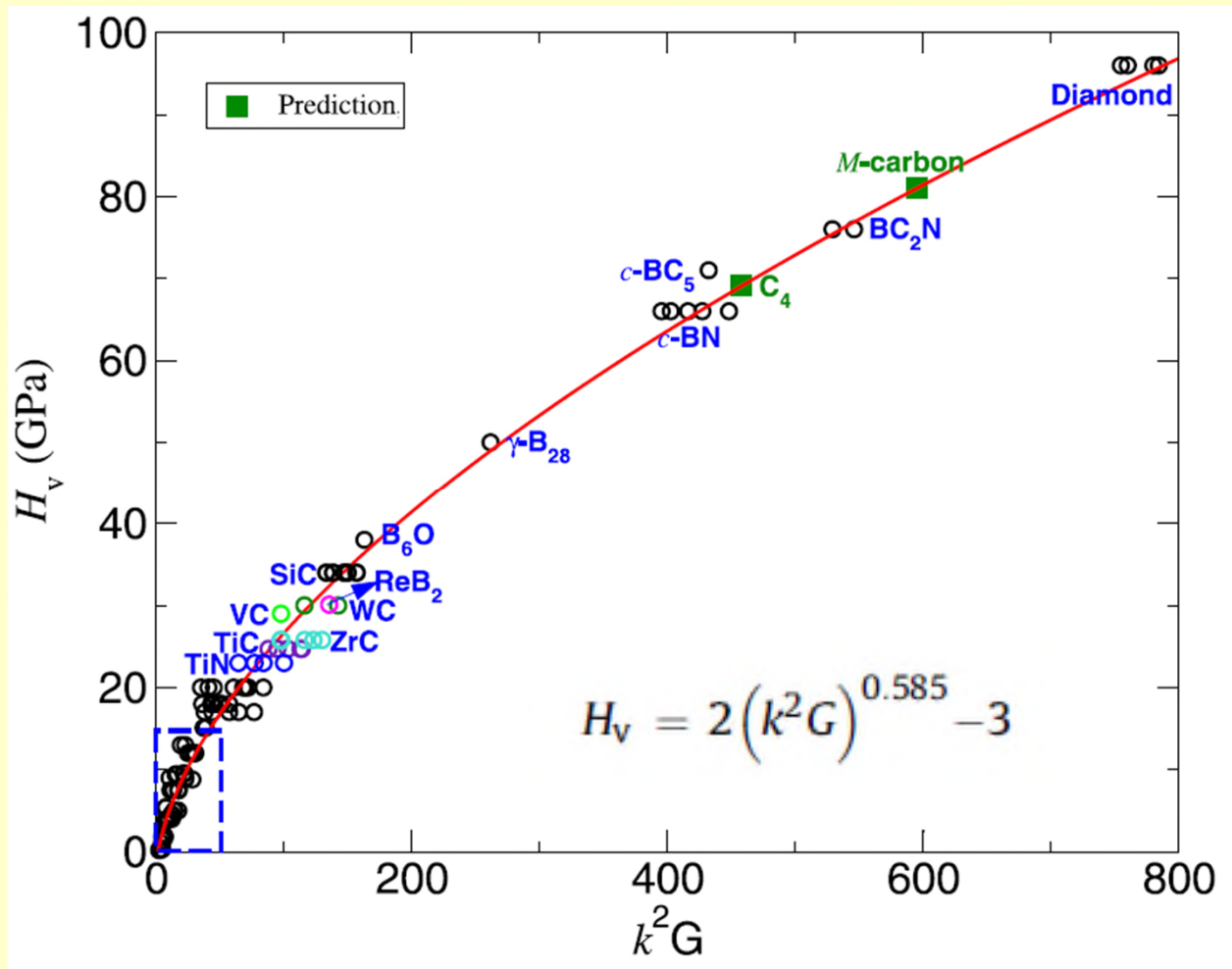


Graphene sheets require 4 electrons/atom. This is achieved by charge transfer

# Boron is the basis for novel superhard materials.

## $\gamma\text{-B}_{28}$ is one of the hardest known materials

B



Overview of hard materials (Chen, 2011)

$\gamma\text{-B}_{28}$  has hardness of 50 GPa (Solozhenko, ARO, 2008)

## Optimization of physical properties: Replacing Edisonian trial-and-error way of discovering new materials

*“I have not failed (ten thousand times). I've just found 10000 ways that won't work”*  
(T.A. Edison)



## Relevant USPEX input

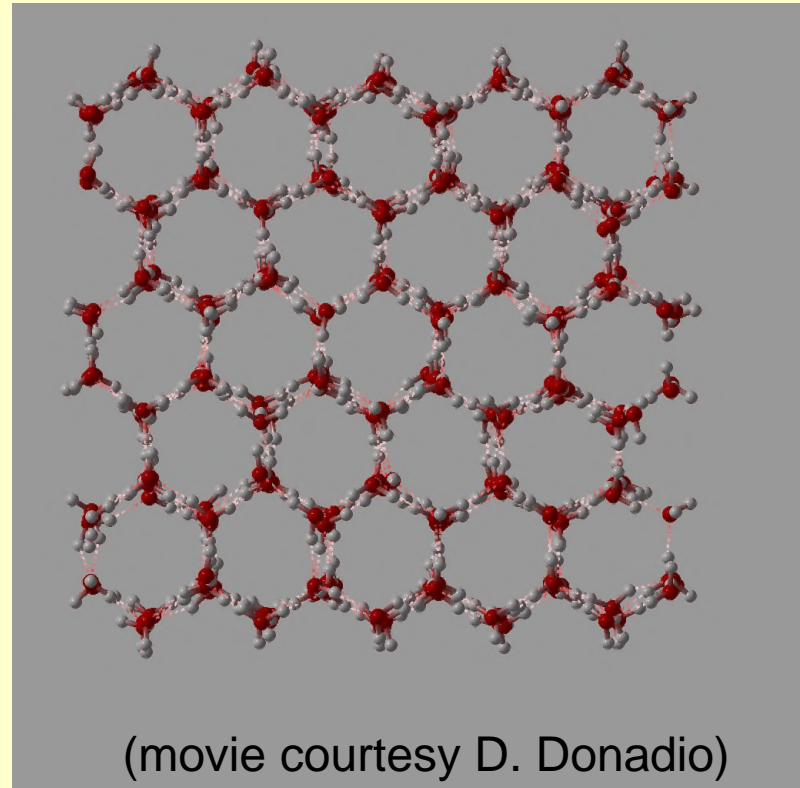
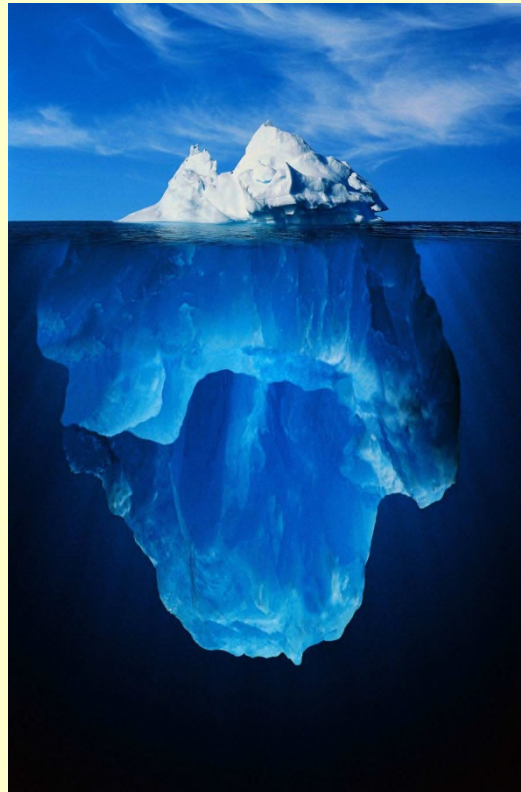
**enthalpy : optType**

**Possible values:**

- **enthalpy (default)**
- **volume**
- **hardness**
- **struc\_order**
- **aver\_dist**
- **diel\_sus**
- **gap**
- **diel\_gap**
- **mag\_moment**
- **struc\_entropy**

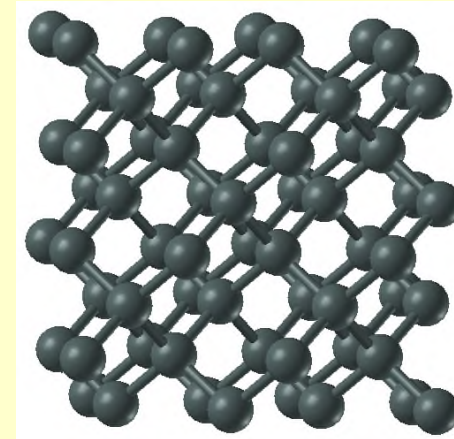
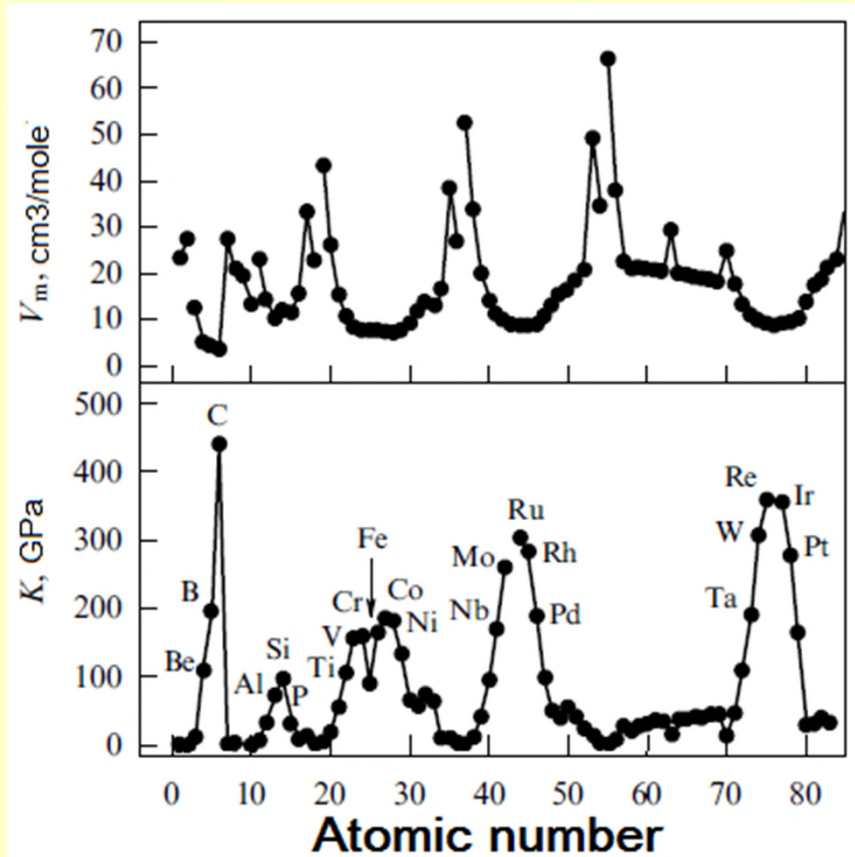


# Examples of interesting properties: Why is ice lighter than water?

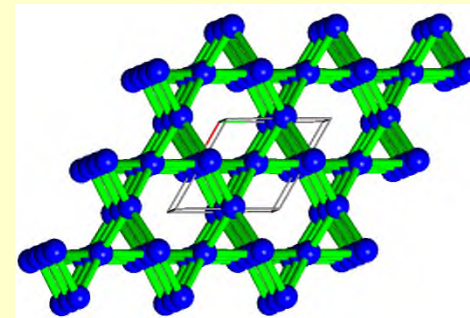


**Structure of ice contains large empty channels, which explain its low density.**

# Looking for the densest possible material: carbon allotrope(s) denser than diamond



**diamond structure**

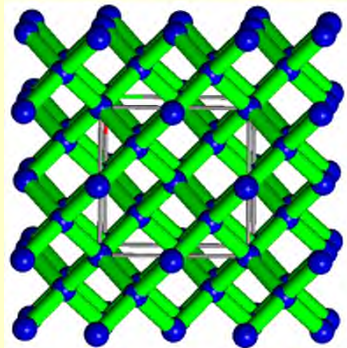


**new structure, 3.2% denser than diamond!**  
(Zhu, ARO, et al., PRB 2011)

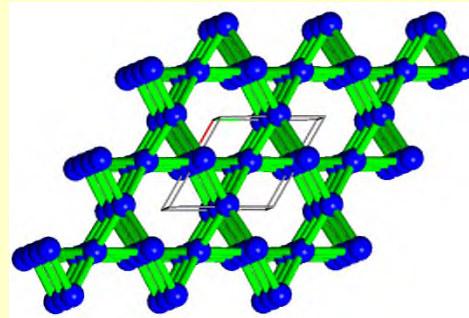
**Diamond has the highest bulk modulus and lowest atomic volume among all elements (and compounds) From Brazhkin (2009).**

# Finding materials with target properties: optimizing the density (rather than energy)

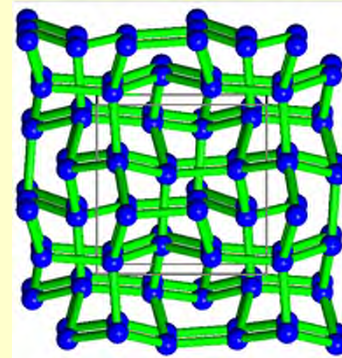
- Diamond has the highest hardness, highest bulk modulus and lowest atomic volume among all materials [Brazhkin (2009)]. Can this be improved?
- We found 3 interesting denser-than-diamond structures (by 1.1-3.2%). Structural analogy between C and SiO<sub>2</sub>.



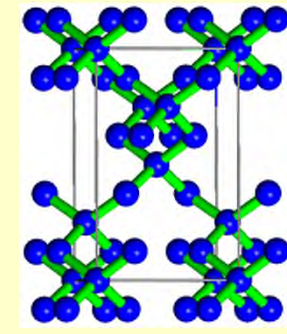
**diamond structure**



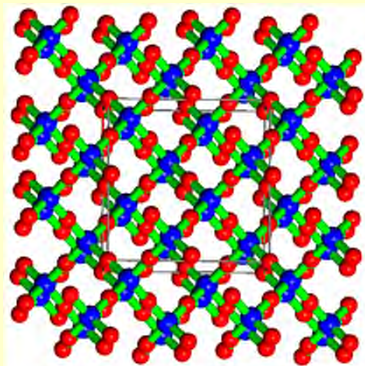
**hP3 structure**



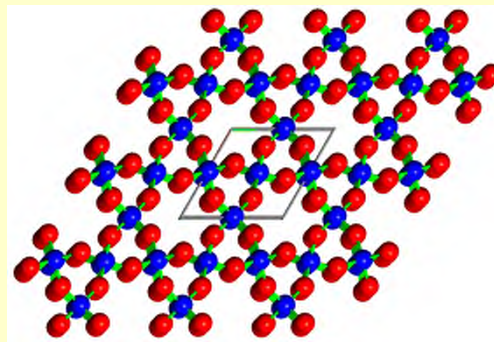
**tP12 structure**



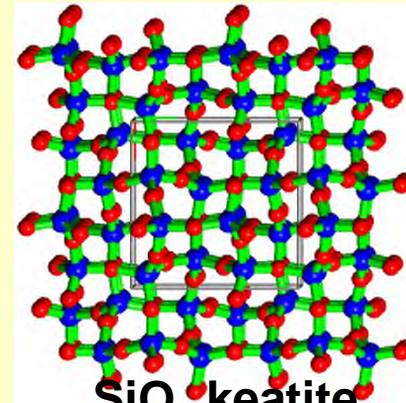
**tI12 structure**



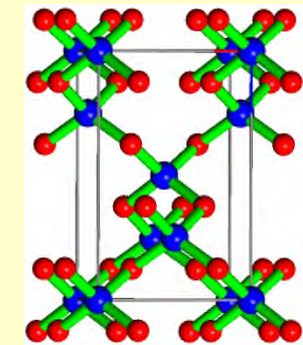
**SiO<sub>2</sub> cristobalite structure**



**SiO<sub>2</sub> quartz structure**



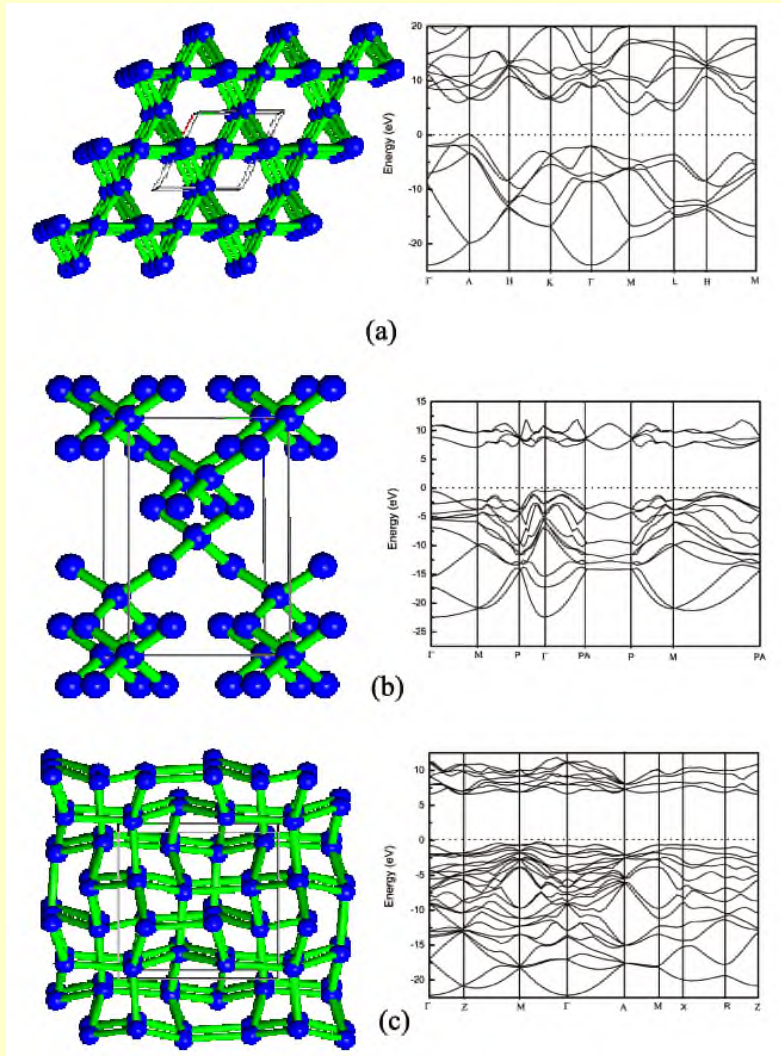
**SiO<sub>2</sub> keatite structure**



**High-pressure SiS<sub>2</sub> phase**



# Superdense carbon allotropes [Zhu, ARO, et al., PRB 2011]

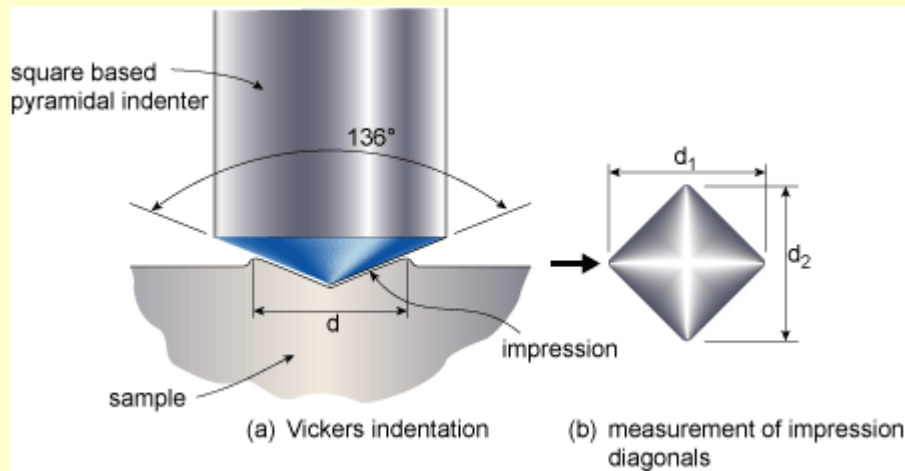


- (1)  $sp^3$  hybridization
- (2) superhard
- (3) huge refractive indices (up to 2.8!)
- (4) strong dispersion of light
- (6) tP12 is the widest-gap form of carbon (7.3 eV)

TABLE II. Energy relative to diamond ( $\delta E$ ), volume ( $V$ ), bulk modulus ( $B_0$ ), average bond length ( $d$ ), hardness ( $H$ ), static dielectric constant  $\epsilon_s$  for the investigated structures. Experimental data are in parentheses

Allotropes	$\delta E$ eV/atom	$V$ $\text{\AA}^3/\text{atom}$	$B_0$ GPa	$d$ $\text{\AA}$	$H$ GPa	Band gap eV	$\epsilon_s$
Diamond	0	5.70 (5.68)	431.1 (446)	1.545 (1.54)	94.3 (96 $\pm$ 5)	5.4 (5.5)	5.57 (5.68)
Lonsdaleite	0.024	5.71	437.3	1.548	93.2	5.0	5.52
M-carbon	0.159	5.97	392.7	1.551	89.8	5.0	5.52
bct-4	0.196	6.01	411.4	1.549	91.1	3.8	5.42
bc8	0.697	5.60	389.6	1.588	88.8	3.5	6.67
hP3	1.113	5.49	432.7	1.603	87.6	3.0	7.73
tI12	1.140	5.48	425.0	1.603	87.2	5.5	7.87
tP12	0.883	5.64	396.0	1.583	88.3	7.3	7.08

# Theory of hardness? Yes!



Material	Model of Li et al. (2009)	Lyakhov & ARO (2011)	Exp.
Diamond	91.2	89.7	90
<b>Graphite</b>	<b>57.4</b>	<b>0.17</b>	<b>0.14</b>
TiO <sub>2</sub> rutile	12.4	12.3	8-10
β-Si <sub>3</sub> N <sub>4</sub>	23.4	23.4	21
SiO <sub>2</sub> stishovite	31.8	30.8	33

Single-bond crystals:

$$H_k(\text{GPa}) = 423.8 N_v X_{ab} e^{-2.7f_i} - 3.4$$

Multibond crystals:

$$H_k(\text{GPa}) = \frac{423.8}{V} n \left[ \prod_{a,b=1}^n N_{ab} X_{ab} e^{-2.7f_{i(ab)}} \right]^{1/n} - 3.4$$

$N_v$  - bond density

$V$  - volume

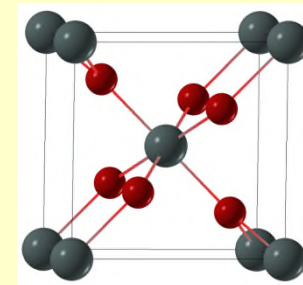
$N_{ab}$  - number of bonds of type  $ab$

$X_{ab}$  - "bond electronegativity"

$f_i$  - bond ionicity

$$X_k = \sqrt{\frac{\chi_i^k \chi_j^k}{CN_i^k CN_j^k}}$$

Lyakhov & ARO (2011) – augmented model of Li (2009) by bond valence model and graph theory.

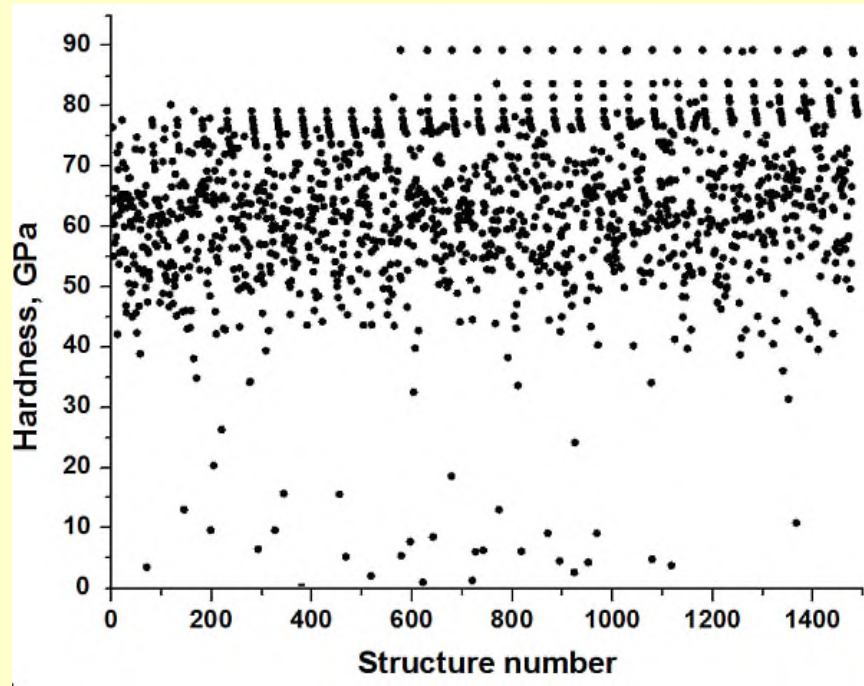


Special Issue "Theory of superhard materials" (editor – A.R. Oganov)

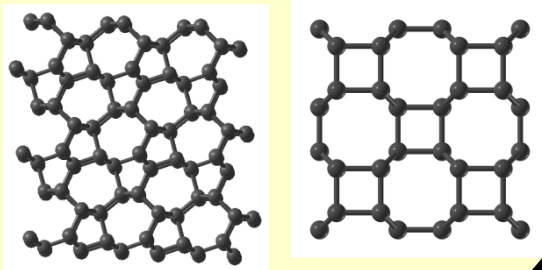
*Journal of Superhard Materials, 2010, Vol. 32, No. 3, © Allerton Press, Inc., 2010.*



# Is diamond the hardest structure for carbon? Yes [Lyakhov & ARO, PRB 2011].



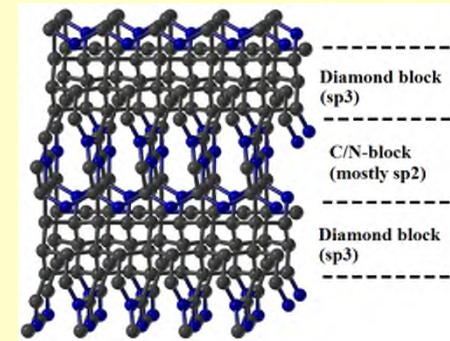
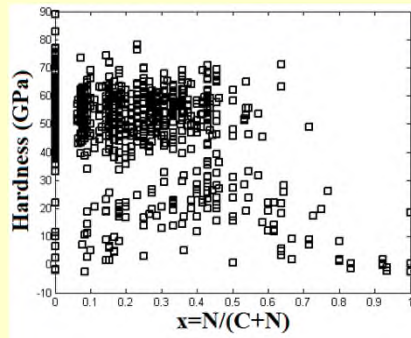
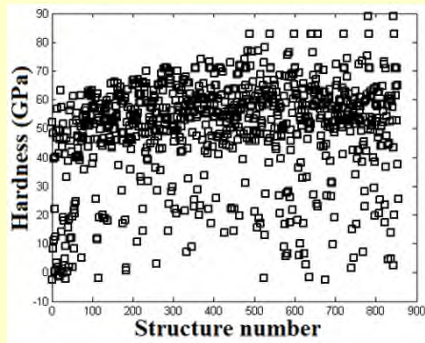
Simulation for carbon, 16 atoms/cell



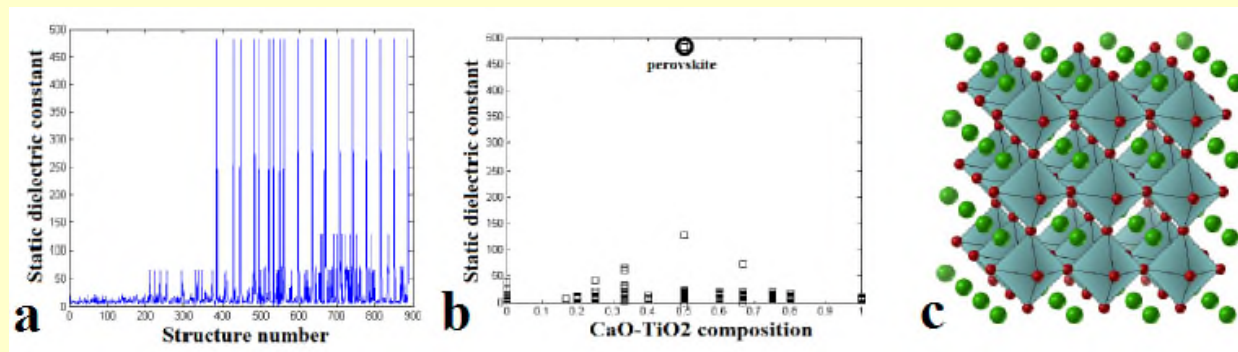
All of the hardest structures are  $sp^3$ -hybridized

Structure	Knoop hardness, GPa	Enthalpy, eV/atom
Diamond	89.7	0.000
Lonsdaleite	89.1	0.026
<i>C2/m</i>	84.3	0.163
<i>I4/mmm</i>	84.0	0.198
<i>Cmcm</i>	83.5	0.282
<i>P2/m</i>	83.4	0.166
<i>I2<sub>1</sub>2<sub>1</sub>2<sub>1</sub></i>	82.9	0.784
<i>Fmmm</i>	82.2	0.322
<i>Cmcm</i>	82.0	0.224
<i>P6<sub>5</sub>22</i>	81.3	0.111

# USPEX can optimize properties of variable-composition systems



Our calculations (ARO, 2012) clearly show that carbon nitrides cannot be harder than diamond, thus ending the old quest (Liu & Cohen, 1989; Teter & Hemley, 1995).



Our calculations (ARO, 2012) show that in the CaO-TiO<sub>2</sub> system CaTiO<sub>3</sub> perovskite has the highest dielectric constant.

# Looking for the hardest materials... What is the hardest oxide?

- Leger (Nature 1996) – SiO<sub>2</sub> stishovite (33 GPa).
- Dubrovinsky (Nature 2001) – TiO<sub>2</sub>-cotunnite (38 GPa).
- He (Appl.Phys.Lett. 2002) – B<sub>6</sub>O (45 GPa).

## Materials science

### The hardest known oxide

A material as hard as diamond or cubic boron nitride has yet to be identified<sup>1-6</sup>, but here we report the discovery of a cotunnite-structured titanium oxide which represents the hardest oxide known. This is a new polymorph of titanium dioxide, where titanium is nine-coordinated to oxygen in the cotunnite

Table 1 Hardness of polycrystalline materials

Material	Bulk modulus (GPa)	Hardness* (GPa)	Ref.
B <sub>4</sub> C	200	30 (30)	3
SiC	248	29 (29)	3
Al <sub>2</sub> O <sub>3</sub>	252	20 (19)	7
SiO <sub>2</sub> , stishovite	291	32 (33)	7
WC	421	30 (30)	8
Cubic BN	369	(32)	3
Cotunnite-type TiO <sub>2</sub> †	431	38	
Sintered diamond	444	(50)	3

\*Literature data are given in parentheses. The uncertainty in measured hardness is less than 3 GPa.  
†Measurements were made at 157 ± 2 K.

L. S. Dubrovinsky\*, N. A. Dubrovinskaia\*, V. Swamy†, J. Muscat†, N. M. Harrison‡, R. Ahuja§, B. Holm§, B. Johansson§||  
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†CSIRO Minerals, Box 312, Clayton South,

- Suggestion of TiO<sub>2</sub> is clearly incorrect – (1) it is unstable at 1 atm (!), (2) Experiments of Dubrovinsky were low-quality (bulk modulus is 43% overestimated – Al-Khatatbeh (2009), Hamane-Nishio (2010)) (3) No phase of TiO<sub>2</sub> can be harder than ~15 GPa.**



# Pseudo-hard $\text{TiO}_2$ : Dubrovinsky et al. (Nature 2001) overestimated bulk modulus by 43%, hardness by 140%

## Materials science

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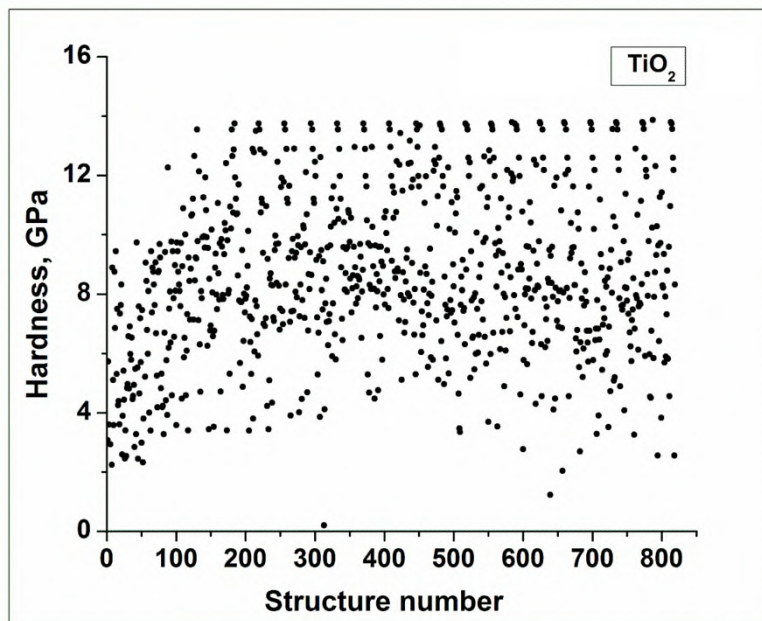
†Measurements were made at  $157 \pm 2$  K.

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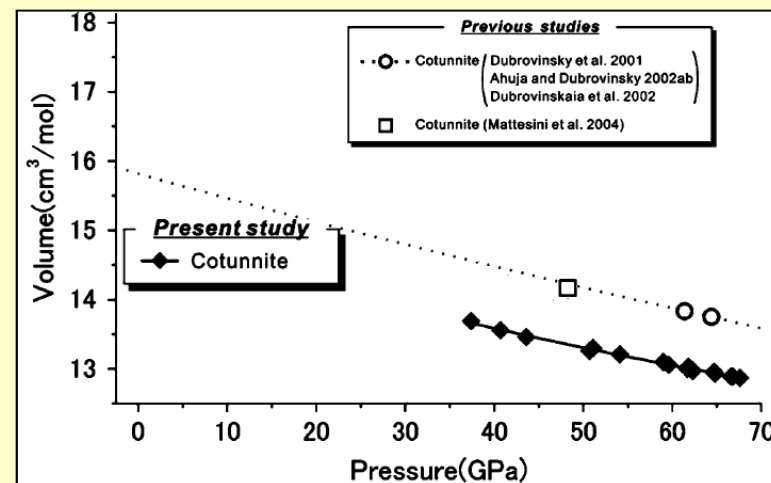
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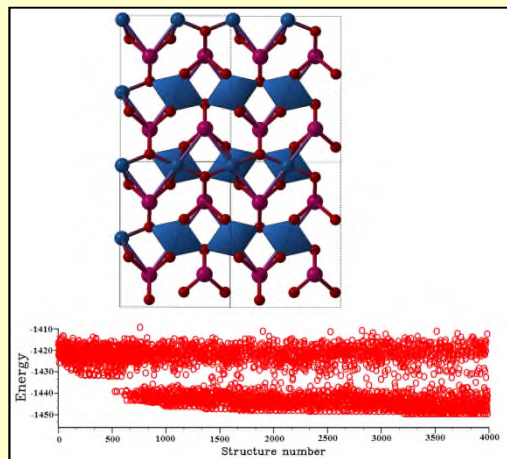


Lyakhov & ARO (2011)

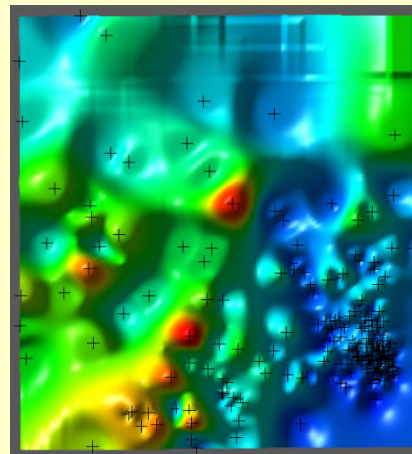


Nishio-Hamane (2010): bulk modulus is ~300 GPa, not 431 GPa.

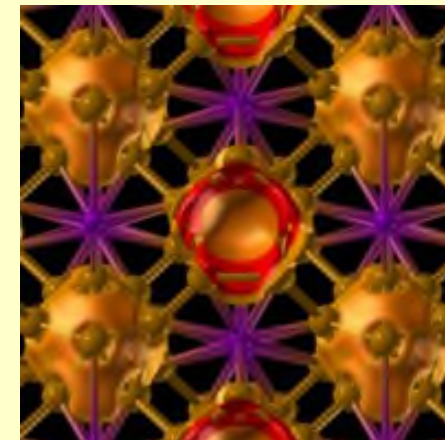
# New developments in crystal structure prediction extend the range of problems that can be solved



**Powerful methods for crystal structure prediction**



**Fingerprints - new language for crystallography**



**Predicted new materials and phenomena**